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AUTHORITY

AFWL ltr, 30 Nov 1971

NERVA MATERIALS IRRADIATION PROGRAM

Volume 2
GTR Test 16 — WANL Materials Test

Space Nuclear Propulsion Office
of the
National Auronauties and Space Administration
Cleveland, Ohio

Contract No. AF 29(601)-6643 Supplement 2



NUCLEAR AEROSPACE RESEARCH FACILITY

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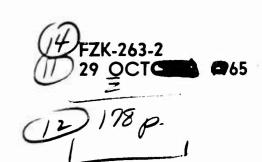
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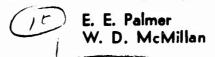
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NUCLEAR AEROSPACE RESEARCH FACILITY

NERVA MATERIALS IRRADIATION PROGRAM Volume 2 GTR Test 16 — WANL Materials Test



37/W202 Cemented Orfices

15/W401 O-Ring Seals

37/W401 Tensile Specimens Resistivity Specimens Steel Springs

Prepared for
Space Nuclear Propulsion Office
of the
National Aeronautics and Space Administration
Cleveland, Ohio



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FOREWORD

GTR Test 16 is the first of two NERVA materials tests to be performed by the Nuclear Aerospace Research Facility (NARF) at General Dynamics/Fort Worth, under Supplemental Agreement No. 2, Contract AF29(601)-6643, for the Space Nuclear Propulsion Office, Cleveland, Ohio (SNPO-C). The 3-Mw Ground Test Reactor (GTR) was used as the source of nuclear radiation in these tests.

The test was designed to measure the combined effects of nuclear radiation and liquid hydrogen or liquid nitrogen on metallic and graphite materials. This document describes those tests conducted at liquid-nitrogen temperatures for Westinghouse Astronuclear Laboratory. The tests conducted at liquid-hydrogen temperatures for Aerojet-General Corporation are described in GD/FW Report FZK-263-1.

Previous SNPO-C tests on NERVA components, conducted at NARF under Contract AF33(657)-7201, are reported in GD/FW Report FZK-170, Volumes 1 through 9. Other NERVA component tests, conducted under Supplement 1 to Contract AF29(601)-6213, are reported in GD/FW Report FZK-184, Volumes 1 through 6.

ACKNOWLEDGMENTS

The authors wish to express their gratitude to the following persons:

Mr. Jim Begley of Westinghouse Astronuclear Laboratory for his assistance in the conduction of the test program.

Mr. J. B. Wattier of General Dynamics/ Fort Worth for his statistical analysis of the data.

Mr. R. L. Trittipo and other General Dynamics personnel for their assistance in the experimental setup, conduction of the test, and reduction of data.

SUMMARY

The nuclear facility of GD/FW performed three tests for the Westinghouse Astronuclear Laboratory in accordance with test specifications described in WANL Reports TME-1037 and TME-1090.

These three tests were briefly.

- 37/W401(1) 1000 metal and graphite tensile specimens, 4 steel-wire resistivity specimens, and 8 stainless-steel spring specimens tested at LN2 and elevated temperatures. Of these, 700 tensile specimens, 4 resistivity specimens, and 4 spring specimens were irradiated at LN2 temperatures.
- 15/W401/(1) 2 0-ming seal test fixtures, irradiated in a gaseous-hydrogen environment.
- 37/W202(3) 14 unfueled fuel-element segments containing cemented orifices, irradiated at LN2 and ambient-air temperatures.

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The tensile specimens were divided into irradiation and control specimens. The irradiation specimens were irradiated at LN₂ temperatures to a maximum integrated neutron flux of 1.0 x 10¹⁸ n/cm² (E>1 Mev). After a storage period in LN₂ of approximately 30 days, the specimens were tested in tension to break under various test conditions ranging from a -320°F test temperature without an annealing treatment to a 1290°F test temperature after a 1-hr annealing treatment at some elevated temperature. All specimens were maintained at LN₂ temperatures without warmup from before the irradiation until the annealing cycles were begun (a period of approximately 60 days). Those specimens that were not annealed were maintained in LN₂ without warmup until after they were pulled in tension to break.

Ultimate tensile strength, tensile yield strength, notched tensile strength, notched-to-unnotched tensile-strength ratio, percent elongation, and percent reduction in area were determined from the test results. Significant changes in one or more properties were noted in all materials maintained in LN₂ without warmup, with appreciable to complete recovery evident after the annealing treatments.

Changes in ultimate tensile strength were generally slight except for a highly significant decrease of 70% in beryllium specimens maintained at ${\rm LN}_2$ temperature. Recovery was apparent in all materials after the annealing treatments.

Changes in tensile yield strength were generally significant, with increases of from 7% to 47% experienced by all materials maintained in LN₂ except beryllium, which showed a highly significant decrease of 70%. Appreciable recovery was experienced by all materials after the annealing treatments.

Changes in notched tensile strength were generally slight to significant, with increases of from 2% to 22% experienced by all materials maintained in LN_2 except titanium and beryllium, which showed decreases of 7% and 9%, respectively. Appreciable recovery was noted in all materials after annealing treatments.

Significant decreases in ductility were evident in all specimens maintained in LN_2 except beryllium, which had no measurable elongation for either the control or the irradiated specimens. Appreciable recovery was apparent after the annealing treatments.

In general, material properties experiencing apparent changes when maintained in LN2 without warmup indicated almost full

recovery after a room-temperature anneal.

The four Inconel wire resistivity specimens were irradiated at LN_2 temperature to a maximum integrated neutron flux of 4.5 x $10^{17} \, \text{n/cm}^2$ (E>1 MeV). During irradiation, the resistance of one of the Inconel 718 wire specimens was measured periodically. After a storage period of approximately three months at LN_2 temperature, the resistance of the specimens as a function of annealing temperature and time was measured. The specimens were maintained at LN_2 temperatures without warmup until the annealing cycles were begun.

During irradiation, resistance measurements of the Inconel 718 specimen indicated an increase of approximately 0.02 ohm. Postirradiation resistance measurements after the annealing treatments indicated a maximum resistance increase of approximately 0.013 ohm. This would correspond to an increase in resistance during irradiation of something greater than 13 milliohms.

Four of the eight stainless-steel springs were irradiated at LN_2 temperature to a maximum integrated neutron flux of $4.5 \times 10^{17} \, \text{n/cm}^2$ (E>1 Mev). The other four springs were used as control specimens. Several postirradiation measurements, including spring constant, were made on the specimens. Radiation-induced changes in the specimens, if any, were insignificant.

One of the two O-ring test fixtures irradiated was mounted in the gaseous-hydrogen space inside one of the LH₂ dewars used in the AGC tests being performed during GTR-16. The other specimen was placed inside a sealed aluminum container and mounted to the outside of the LH₂ dewar. A gaseous-hydrogen environment was maintained inside the aluminum container during the irradiation.

The test fixtures received an average integrated neutron flux of $4 \times 10^{16} \text{ n/cm}^2$ (E>2.9 MeV) and a gamma dose of 2.6 x 10^{10} ergs/gm(C). After the irradiation, the fixtures were disassembled and the 0-ring seals returned to Westinghouse for testing at their facilities. No results from these tests are available for this report.

The 14 unfueled fuel-element segments containing cemented orifices were divided into two groups of seven each. One group was irradiated at $\rm LN_2$ temperature; the other group was irradiated at ambient-air temperatures. The specimens irradiated at $\rm LN_2$ temperatures received a maximum integrated neutron flux of 2.5 x $\rm 10^{17}$ n/cm² (E>1 MeV); those irradiated under ambient conditions received a maximum integrated neutron flux of 4.0 x $\rm 10^{17}$ n/cm² (E>1 MeV). All testing of these specimens was performed at Westinghouse by Westinghouse personnel and no results are available for this report.

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I. INTRODUCTION

Under Supplement 2 of Contract AF29(601)-6643, the Nuclear Aerospace Research Facility (NARF) of General Dynamics/Fort Worth is conducting a series of tests to determine the effects of nuclear radiation, in combination with other environmental factors, on materials proposed for application in the NERVA engine. Aerojet-General Corporation (AGC) has prime responsibility for development of the NERVA engine; the nuclear reactor in the engine is being developed by Westinghouse Electric Corporation.

This document reports the procedures and results of tests performed on materials during irradiation test GTR-16 for Westinghouse Astronuclear Laboratory (WANL). Other NERVA tests performed during GTR-16 were sponsored by AGC and are reported separately in Reference 1. The purpose of the WANL tests was to determine the serviceability of metals and graphite under the extremes of temperature and nuclear radiation.

The tests were performed in accordance with specifications submitted by WANL. The test specimens were supplied by WANL; test fixtures and instrumentation were supplied by CD/FW.

Section II contains a discussion of the test setup and procedures and a description of the test specimens. Section III includes a presentation of data in tabular and graphical form, a statistical analysis of the data, and a discussion and analysis of the results.

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II. TEST PROGRAM

The GTR Radiation Effects Testing System at NARF is described briefly in Appendix A and in more detail in Reference 2. Figure A-2 shows the reactor, the reactor "closet" and the three test positions (east, north, and west) in the irradiation cell adjacent to the closet. For this test the reactor was moved to the "full-in" position in the closet (2 in. of water on the north face, 4 in. of water on the east and west faces) and operated at a power level of 3 Mw for 368.2 hr. The total megawatt hours was 1104.8. The plot of integrated power vs real time shown in Figure 2-1 depicts the radiation history of the GTR Test 16. GTR Test 16 was divided into two irradiation periods with a six-day shutdown in between. The first irradiation period was for 570 Mw-hr and the second was for 534.8 Mw-hr. The Westinghouse materials tests were run on the north irradiation position (Fig. 2-2) and remained in the test cell at LN2 temperatures without warmup for the total 1104.8 Mw-hr. The east and west irradiation positions were used for the AGC materials tests at LH2 temperature. The six-day shutdown between the two irradiation periods was required for a changeover of experiments being conducted on the east and west irradiation positions.

2.1 Test Description and Procedures

2.1.1 Tensile Tests

The tensile test was designed to determine the effects of nuclear radiation and LN₂ temperatures on the mechanical

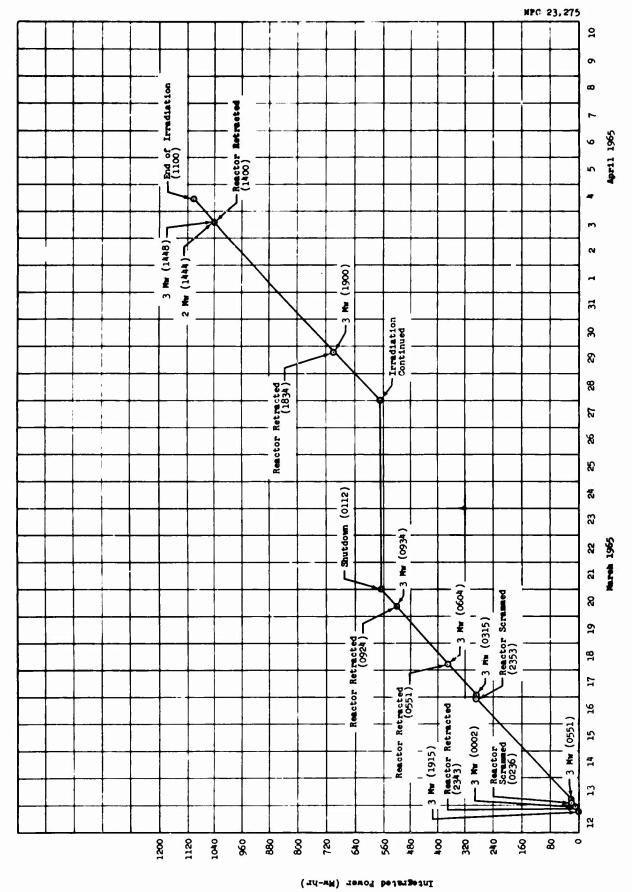


Figure 2-1 Reactor Profile for GTR Test 16

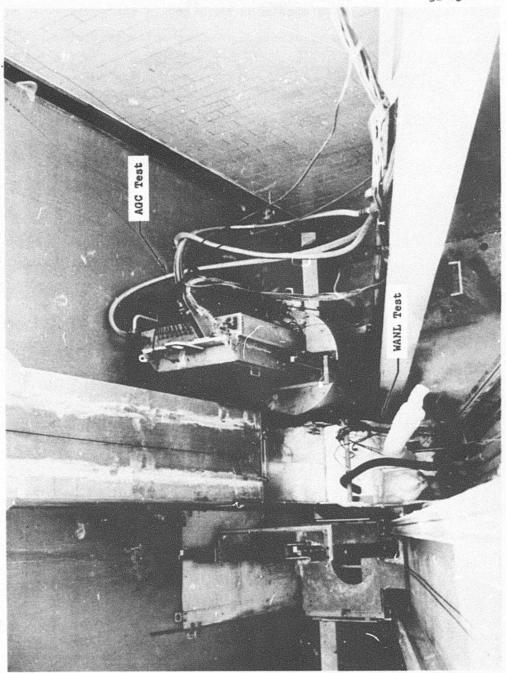


Figure 2-2 Test Hardware in Irradiation Position

properties of metallic and graphite specimens. Of the more than 1,000 metal and graphite specimens included in the program, approximately 700 were irradiated. The graphite specimens, 277 in number, were returned to WANL after the irradiation for postirradiation tests to be performed at their facilities. Tests on the other specimens were performed at GD/FW.

The test consisted of (1) irradiating the specimens at LN₂ temperatures, (2) storing the specimens at LN₂ temperatures for approximately four weeks for radioactivity decay, and (3) pulling the specimens in tension to break at LN₂ and elevated temperatures. Some of these specimens were held at LN₂ temperatures without warmup from the beginning of the irradiation until after they were pulled in tension to break. Others were held at LN₂ temperatures from the start of the irradiation until the beginning of tensile testing. Control specimens were stored in LN₂ during the time that the irradiated specimens were stored for radioactive decay.

Table 2-1 lists the test conditions imposed on the specimens, including the annealing temperatures and the symbols used in this document and related WANL documents to designate each test condition. Table 2-2 lists the number of specimens, control plus irradiated, that were tested under each test condition. Figures 2-3 through 2-5 illustrate the various specimen types used in the test.

After the storage period all specimens were pulled in tension to break in a Model TT Instron tensile testing machine.

All specimens were pulled at a crosshead speed of O.1 in./min with

Table 2-1 Description of Tensile-Specimen Test Conditions

Test Condition*	Description
	<u>Tensile Tests</u>
1A	Specimens were maintained in LN without warmup until after they were pulled in tension to break.
B, B1	Specimens were warmed up to room temperature, then tested at room temperature.
C, Ci	Specimens were warmed up to room temperature, then cooled to LN ₂ temperature (-320°F) and tested.
D _A , D _{Ai}	Specimens were warmed up to room temperature, then annealed for 1 hr at 540°F and tested at 540°F.
D _B , D _{Bi} , D _{Bi} ,	Specimens were warmed up to room temperature, then annealed for 1 hr at 790°F and tested at 790°F.
D _C , D _{C1} , D _{C1}	Specimens were warmed up to room temperature, then annealed for 1 hr at 1040°F and tested at 1040°F.
D _C , D _{Di} , D _{Di} ,	Specimens were warmed up to room temperature, then annealed for 1 hr at 1290°F and tested at 1290°F.
E _A , E _{A1}	Specimens were warmed up to room temperature, annealed for 1 hr at 540°F, then cooled to LN ₂ temperature (-320°F) and tested.
E _C , E _{C1}	Specimens were warmed up to room temperature, annealed for 1 hr at 1040°F, then cooled to LN ₂ temperature (-320°F) and tested.
	Strain-Rate Study
Fi, F _{li}	Specimens were warmed up to room temperature, then cooled to LN ₂ temperature (-320°F) and tested.
Gi, G _{li}	Specimens were warmed up to room temperature, then cooled to -110°F and tested.
Hi, H _{li} , H _{2i}	Specimens were warmed up to room temperature, then tested at room temperature.

^{*}Letter (i) - irradiated specimens.

Prime (') - specimens irradiated in gadlinium foil.

Subscript (1) - specimens pulled at a crosshead speed of

C.Ol in./min.

Subscript (2) - denotes specimens pulled at a variable crosshead speed of from 0.01 to 0.1 in./min.

All other specimens were pulled at a crosshead speed of O.1 in./min.

Table 2-2 Number and Type of Tenaile Specimens Tested Under Bock Condition

									Ā	Test Comdition	ditte			ĺ			'					2)	-115-	15	
Material	Specimen Type	A1 B	15	U	ថ	ď	DA1	Da1	Ho eo	Had H		ਸ਼ੂ ਮੂ	156	8	Ę,	. To	r,	EAL	E.C.	E E	Strate Rate Study	Checkout	Specimens	he turned to MANT.	Totals
Incomel 718, Type 3	3P & 3M	9 9	7	#	60	8	~	cv.	2	2	CV.	9	6 2	1	*	2			9	4		8	2	••	88
Incomel 718, Type 1	1P & 1M	5 7	80	9	œ							7 7							9	9	å				66
Incomel 718-WS Type 3	3P & 3K	7 9	7	6	-						'														36
Incomel X-750 Type 1	1P & 1M	9 9	1	7 7	ထ							9 9	9						80	7		1		Q.	2
Incomel X-750 Type 3	3P & 3K	19	6	€O	9							7 14		В					7	9		3			£
AISI 301-CW, Type 3	3P & 3M	8 9	80	80	8	8	9										60	80		H		2	~	7	1
AISI 303-Se. Type 1	IP & IN	9	7 10	80	6	80	7															1	2		ţ.
Al 2219-T6, Type 3	3P & 3M	9 9	80	*	7	9	8										9	80				5	2		38
Al 2219-T5-Transverse Type 2	2P & 2K	6 9	15	6	75																			£.	53
Al 2219-T6-Radial Type 2	2P & 2N	9	9	5	7												7	7				1			29
Beryllium, Type 4	No 4 9	8	80	13	13																				50
T1 A-110-AT-E11 Type 2	2P & 2N	9 9	9	9	9				1											-			2		35
Graphite	01 & 02																			_				112	222
Inconel 718	Threaded																							7	.,
TOTALS		74 84	%	46	8	24	23	~	N	cu .	~	26 33	3	~	*	CV	2	23	73	8	8	27	ជ	70€	E TO:

• P1 (4), P_{11} (3), Q1 (3), Q_{11} (3), R1 (2), H11 (3) and H21 (2). •• In addition to these, ten specimens were irradiated for AGC.

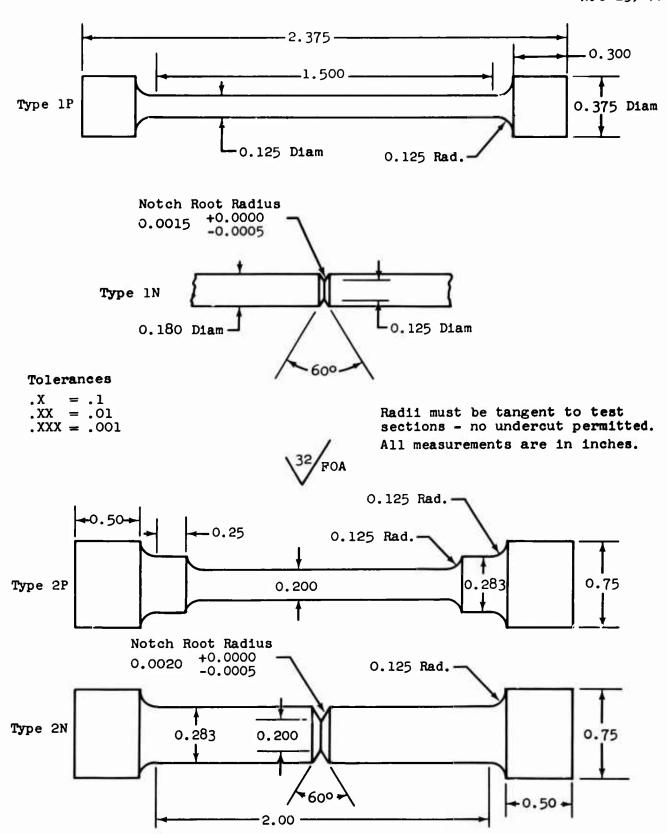


Figure 2-3 Metal Tensile Specimens - Types 1 and 2

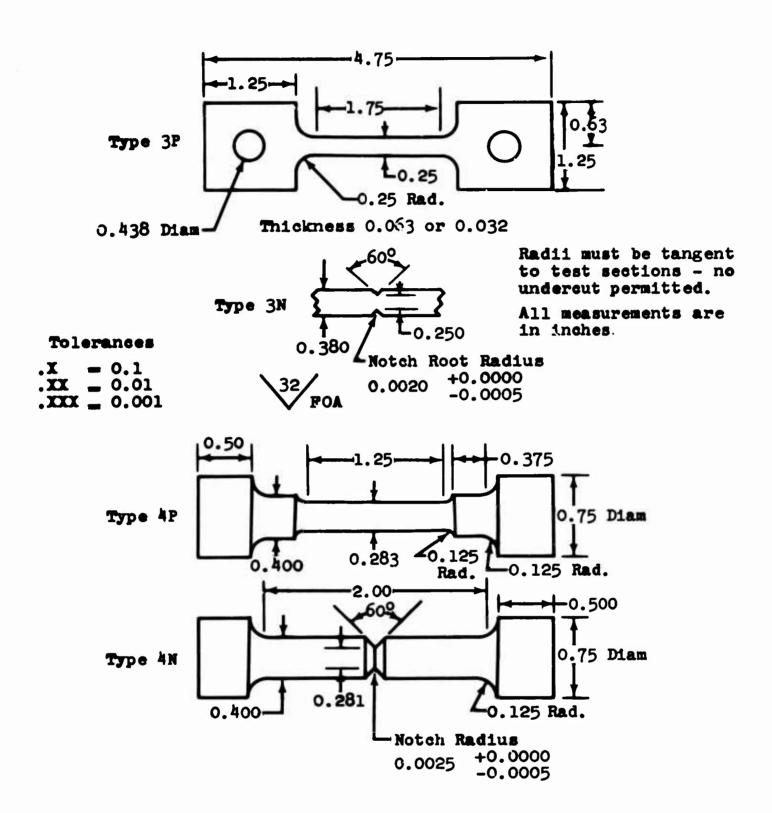
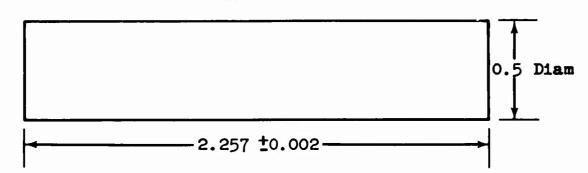


Figure 2-4 Metal Tensile Specimens - Types 3 and 4

Graphite Compression Specimen Type G-1



Diameter or Radius

B = 0.55 C = 0.125 D = 0.40

All radii are to be smooth blended

All measurements are in inches

Graphite Tensile Specimen Type G-2

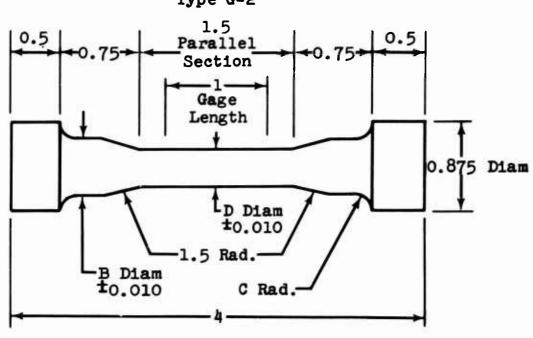


Figure 2-5 Graphite Specimens

the exception of five Incone; 718 specimens pulled at 0.01 in./
min as part of a strain-rate study. After the specimens were
pulled in tension to break, the two halves of each broken specimen
were fitted together and the dimensional measurements required for
the calculation of percent elongation and reduction in area were
made. These measurements were made with a special test jig and
micrometers supplied by Westinghouse. From the results of the
Instron data and the specimen dimension measurements, the
following tensile properties were determined: ultimate tensile
strength, 0.2% tensile yield strength, notched tensile strength,
notched-to-unnotched tensile-strength ratio, percent elongation,
and percent reduction in area.

To satisfy test condition Ai, specimens were transferred in LN_2 from the irradiation dewar to the Instron dewar where they were loaded into specimen grips while submerged in LN_2 . This was accomplished using dippers and tongs as required.

Specimens that had to be pulled at elevated temperatures were pulled inside a cylindrical oven (see Sec. 2.2.1). Before the actual test specimens were inserted, the oven was calibrated for each specimen type at each temperature desired. Specimens that had to be annealed were subjected to the specified temperature for 1 hr in a forced-air oven. Control of the annealing oven was better than $\pm 5^{\circ}$ F; control of the cylindrical Instron oven was $\pm 10^{\circ}$ F of the set point, from the top of the upper grip to the bottom of the lower grip. The oven was calibrated to

hold the middle of the specimens to within $\pm 5^{\circ}$ F of the desired set point.

A strain-rate study was performed on 18 Inconel 718 specimens. Specimens were tested at 80° , -110° , and -320° F at crosshead speeds of 0.01 and 0.1 in./min. The -110°F temperature was obtained with a mixture of crushed dry ice and ethyl alcohol.

The possibility of ozone crystals forming in the LN₂ dewar during irradiation and the subsequent storage period was lessened by two safety measures. The first was the sampling and testing of the LN₂ supply for oxygen content. The oxygen content was found to be always less than 20 ppm. The second precaution was the periodic dumping of a portion of the LN₂ from the bottom of the dewar, the flow of which would tend to carry off any ozone crystals which might form in the dewars. The necessary precautions were taken during dumping of the dewar to insure that the LN₂ level always remained above the test specimens.

2.1.2 Resistivity Tests

Electric resistivity is a sensitive measure of lattice irregularities in metals. Therefore, resistance measurements of several Inconel specimens were made to determine the extent of lattice irregularities caused by neutron bombardment.

Two specimens each of Inconel X-750 and Inconel 718 were irradiated. The specimens consisted of 10-15 ft of 0.020-in.-diam wire wound upon an aluminum mandrel. The wire was insulated from the mandrel by a coating of Bean H cement on the mandrel. The mandrel was placed inside a tubular aluminum vessel open at

one end. The vessel, mandrel, and wire can be seen in Figure 2-6. The aluminum vessels were placed in the north dewar where they remained submerged in LN2 until removed from the dewar.

During irradiation the resistance of one of the Inconel 718 specimens was measured periodically. After the irradiation and subsequent radioactivity decay period, the specimens were cryogenically transferred to a smaller dewar for testing. Postirradiation resistance measurements were made before and after annealing treatments at temperatures ranging from -270°F to room temperature. The temperature steps were achieved with a heater placed around the specimen and inside the aluminum vessel containing the specimen. When power was supplied to the heater, the IN2 in the aluminum vessel evaporated and the temperature was allowed to increase to the desired level. The vessel was then refilled with IN2 and the resistance measured. This procedure was repeated for each desired annealing treatment.

The desired annealing temperature was detected by two copperconstantan (Cu-Cn) thermocouples attached to the specimen heater. A calibration of heater temperature vs sample temperature was performed using a spare specimen with three Cu-Cn thermocouples attached to the specimen mandrel.

2.1.3 Steel-Spring Specimen Tests

Eight AISI 304 stainless-steel spring specimens were provided. One of the springs is shown in Figure 2-7. Four of the eight specimens were irradiated at LN₂ temperatures on the north irradiation position along with the tensile specimens. The other

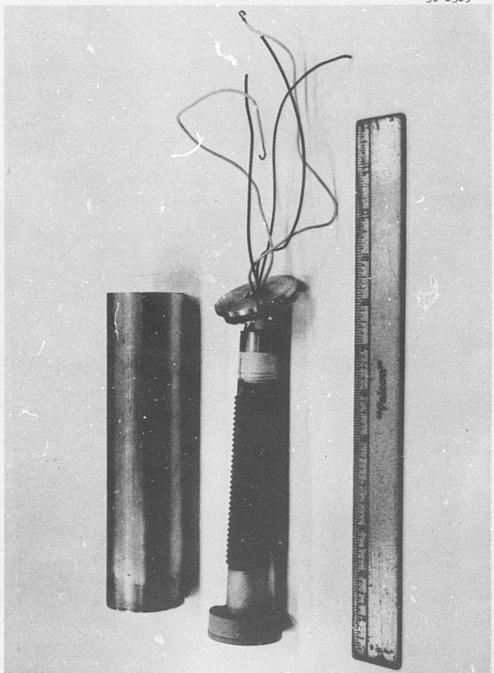
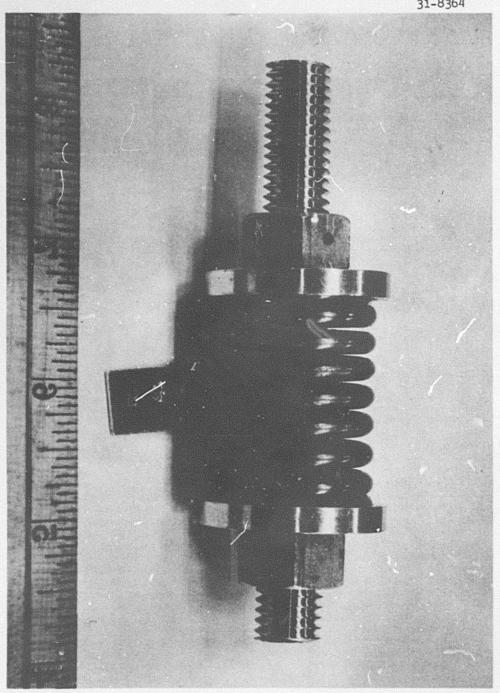


Figure 2-6 Resistivity Specimen and Container





four specimens were used for control measurements.

Before the irradiation the free length of all specimens, irradiation and control, was measured at room temperature. The springs were then compressed by means of a bolt, washer, and nut arrangement through the axial centerline of the springs. The compression ranged from 0.947 in. to 0.983 in. for the eight springs. After the nuts were tightened to the desired deflection, they were pinned to the bolt to insure that the compression remained constant during the irradiation. After the irradiation and subsequent storage period for radioactivity decay, the specimens were removed from the LN₂ and the load removed. The control specimens were stored in LN₂ under load during the time the irradiation specimens were stored for radioactivity decay. The following measurements were made on both the irradiated and the control specimens.

- 1. Specimen free length at room temperature.
- 2. Specimen free length of two control and two irradiated specimens after a 1-hr anneal at 1000 or.
- 3. Spring constant at room temperature (by means of load-deflection curves) of two control and two irradiated specimens after a room-temperature anneal.
- 4. Spring constant at room temperature of two control and two irradiated specimens after a 1-hr anneal at 1000 oR.
- 5. Free length of all specimens after the spring constant tests.

2.1.4 O-Ring Seal Tests

This test was conducted to evaluate 0-ring seals (PMP-6188A phenylmenthylvinyl silicone elastomers) at cryogenic-

and ambient-temperature conditions. Two test fixtures, each containing four 0-ring seal specimens, were supplied by WANL. One of the fixtures after risassembly is shown in Figure 2-8. One test fixture was captulated in an aluminum container leak-tested to 50 psi helium with a helium-leak detector. The other test fixture (without capsule) was located in the hydrogen-gas space inside the east LH₂ dewar used in the test (Ref. 1). The capsulated fixture was placed on the outside of the east LH₂ dewar and irradiated under ambient-temperature conditions. The capsulated fixture contained a Cu-Cn thermocouple, the output of which was monitored during the irradiation. Sufficient hydrogen gas was bled through the capsulated fixture to maintain it in a hydrogen environment during the irradiation. After the irradiation the test fixtures were dismantled and the eight 0-ring seals returned to WANL. All testing of the seals was done at Westinghouse by WANL personnel.

2.1.5 Cemented-Orifice Tests

Fourteen unfueled fuel-element segments containing cemented orifices were irradiated. Photographs of the test specimens are deleted from this report because of their security classification. The specimens were divided into two groups of seven each. One group was placed inside the north dewar in LN2; the other group was placed on the outside of the north dewar in ambient air. The specimen types located inside and outside the dewar are listed below.

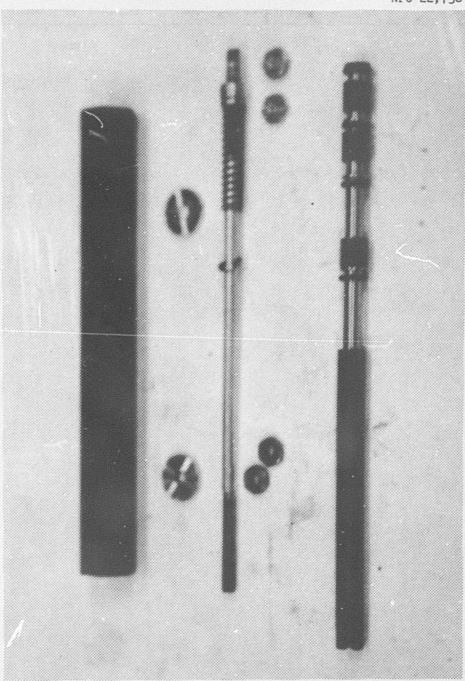


Figure 2-8 Dismantled O-Ring Test Fixture

Specimen	Number
Inside LN ₂ Dewar	Outside LN ₂ Dewar
1 and 2	3 and 4
1	2 and 3
1 and 2	3
1 and 2	3 and 4
	Inside LN ₂ Dewar 1 and 2 1 and 2

The temperature of the specimens located outside the dewar was monitored with a Cu-Cn thermocouple and a Minneapolis-Honeywell multipoint recorder. These items were returned to Westinghouse after the irradiation. All tests performed on these specimens were performed by Westinghouse personnel at their facilities.

2.2 Test Hardware and Instrumentation

2.2.1 Tensile Tests

The tensile specimens were irradiated in aluminum loading racks which were mounted in an aluminum framework (Figs. 2-9 through 2-13). The location of specimens by specimen number in each of the racks is depicted in Figure 2-14. The configuration of the racks and framework was such that the racks could be removed from the framework in sequence from top to bottom while submerged in LN2. This allowed for the systematic removal, with tongs and dippers, of individual specimens from the irradiation dewar to the Instron dewar, maintaining them in LN2 during transfer without interfering with the other specimens in the dewar.

NPC 22,753 31-8225

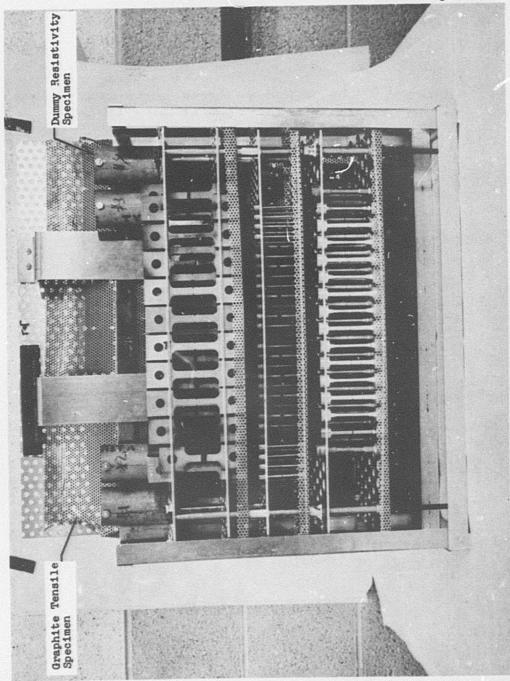


Figure 2-9 Tensile Specimen Loading for Irradiation Test - Front View

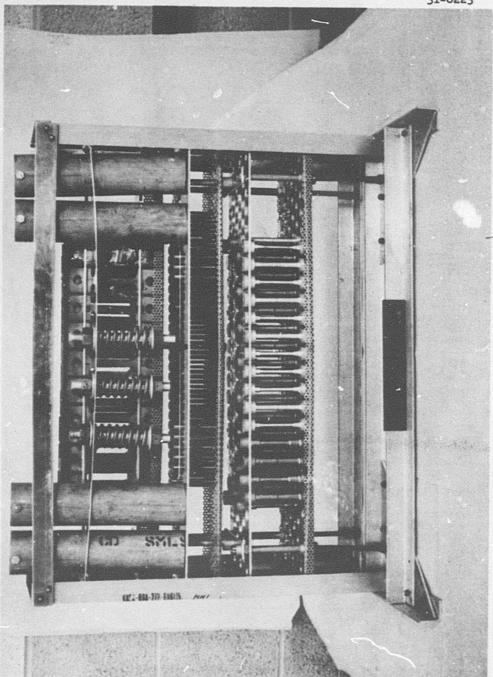


Figure 2-10 Tensile Specimen Loading for Irradiation Test - Rear View

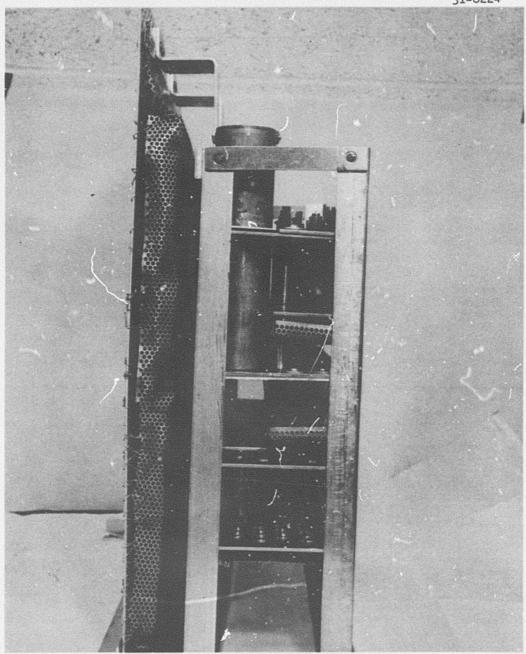
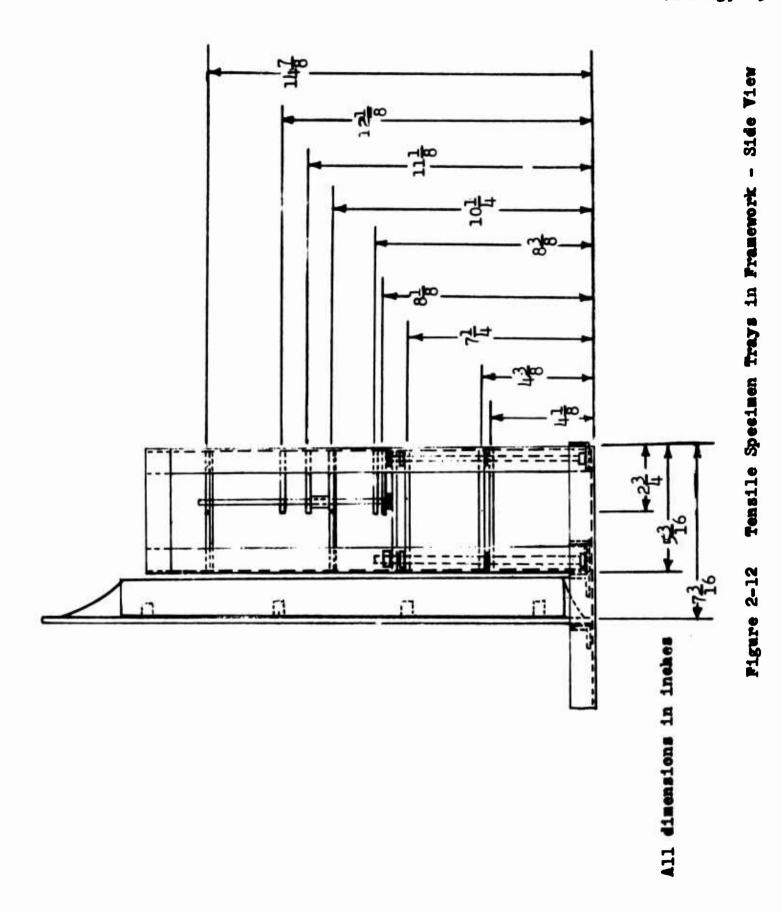
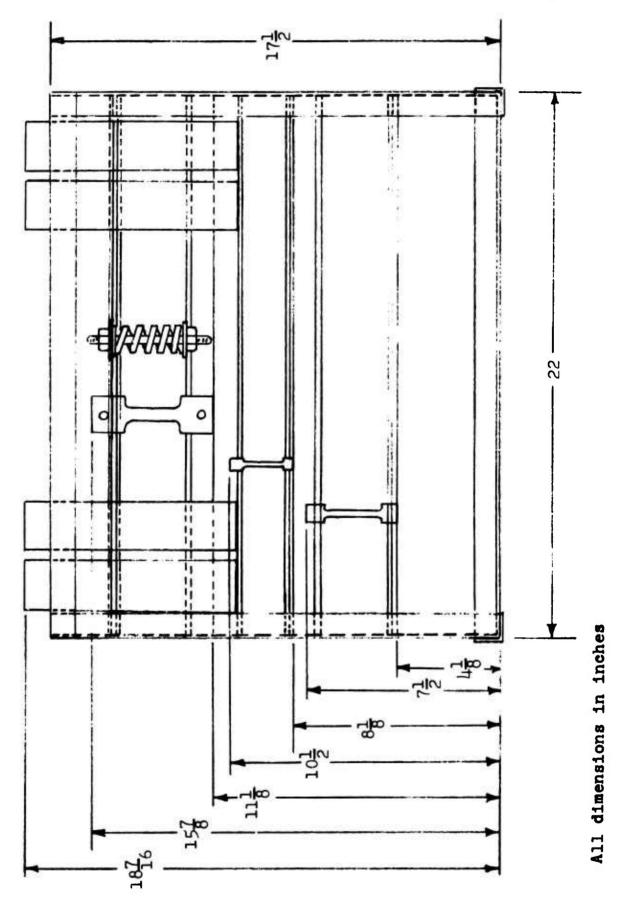
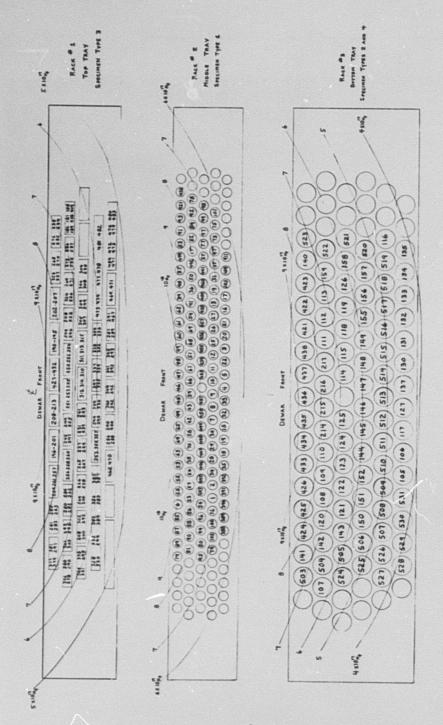


Figure 2-11 Tensile Specimen Loading for Irradiation Test - Side View





ure 2-13 Tensile Specimen Trays in Framework - Front View



Tensile Specimen Locations and Integrated Neutron Flux (E > 1 Mev) Profile Figure 2-14

The 277 graphite specimens were secured to a perforated aluminum tray and mounted to the back of the aluminum framework. The tray can be seen in Figure 2-11. The locations of specimens on the tray are shown in Figure 2-15. The graphite specimen tray was mounted in the framework in such a way that it could be lifted from the irradiation dewar (filled with LN_2) independent of other specimens in the dewar.

The irradiation dewar is shown in Figure 2-16. The loading racks, graphite specimen tray, and framework were placed in the dewar and the dewar filled with LN_2 . The LN_2 level was maintained above the specimens from before the irradiation, through the storage period for radioactivity decay, until the annealing cycles were begun during the postirradiation testing of the specimens, a period of approximately 60 days.

The instrumentation used to monitor the liquid level in the dewar during the irradiation is shown in Figures 2-17 and 2-18. This instrumentation, working in conjunction with a liquid-level probe mounted in the dewar, gave a visual and/or audible indication of liquid level.

The liquid-level probe consisted of seven 0.25-watt carbon resistors mounted in a rake 28-1/2, 11-3/4, 9-3/4, 7-3/4, 5-5/8, 4-1/8, and 1-3/4 in. below the bottom of the dewar flange and of three Cu-Cn thermocouples mounted 11, 9, and 7 in. below the flange. The level was maintained between the resistors at 5-5/8 and 4-1/8 in. Each resistor in the probe was excited to dissipate its rated power for maximum sensitivity and response. The changes

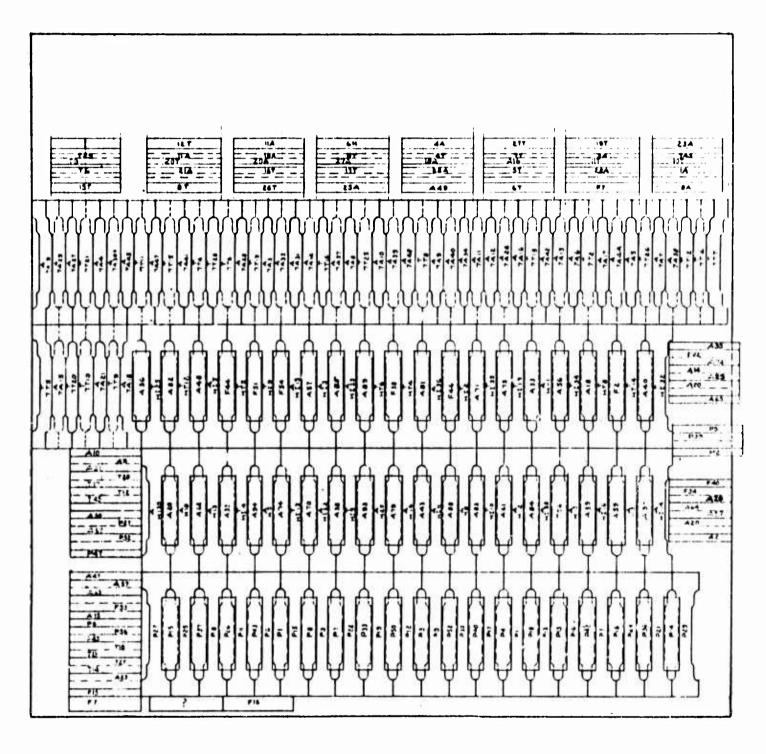
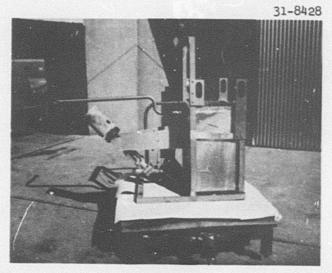
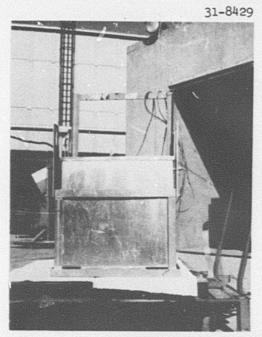


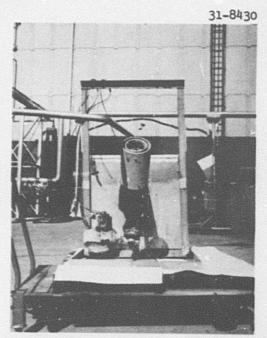
Figure 2-15 Location of Graphite Specimens on Rack



Side View



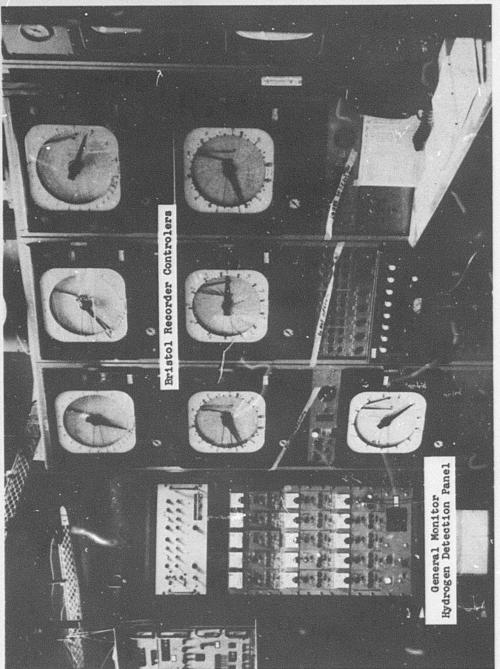
Front View



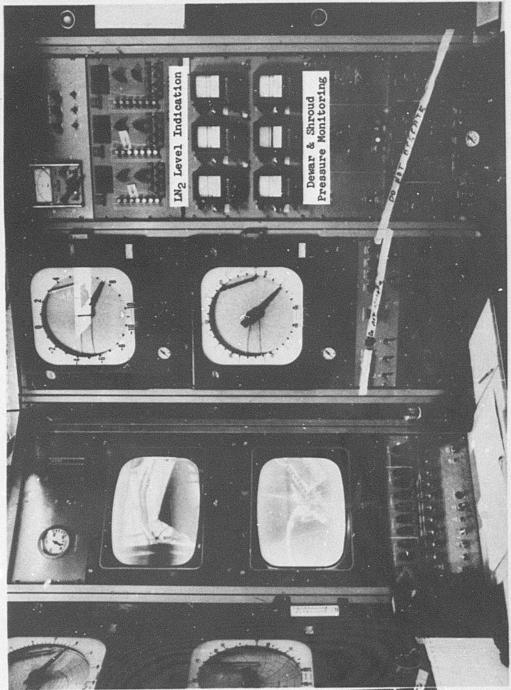
Rear View

Figure 2-16 WANL Liquid-Nitrogen Dewar

NPC 23,546 31-8343



Radiation Effects Console in Reactor Control Room-Laft Side Figure 2-17



Radiation Effects Console in Reactor Control Room-Fight Side Figure 2-18

in resistance as a function of temperature were used to trigger the alarm-system instrumentation, which indicated the liquid level by visual and/or audible means.

The level in the dewar was automatically controlled by Bristol recorders operating in conjunction with the thermocouples mounted in the liquid-level probe. The outputs of the thermocouples were converted by the Bristol controller to a pneumatic signal which was used to operate a Fischer Proportional Positioner mounted on the LN₂ cryogen supply valve.

The instrumentation used to monitor the level in the dewar after the irradiation, during the storage period, and during the subsequent pulling of the specimens is nown in Figure 2-19. This system also gives visual and audible indication of liquid level and automatically controls the liquid level in the dewar by electric signals to a solenoid valve in the LN₂ supply line. The same type of liquid-level probe was used during the storage period as was used during the irradiation, and the liquid level in the dewar was maintained at the same level.

Figure 2-20 depicts the instrumentation employed in the pulling of specimens in tension to break. A 1/2-in.-thick lead shield, not shown in the photograph, was installed on the front of the Instron machine to shield personnel during the pulling of irradiated specimens. A sketch of the oven control system is presented in Figure 2-21.

Figure 2-22 depicts representative calibration specimens, with thermocouples attached, and tools used in the transfer

NPC 23,281 31-8363

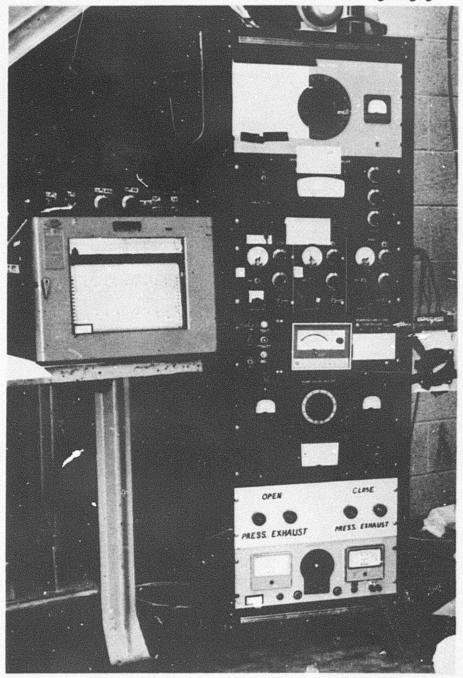


Figure 2-19 Liquid-Level Control System (IML)

NPC 23,280 31-8363

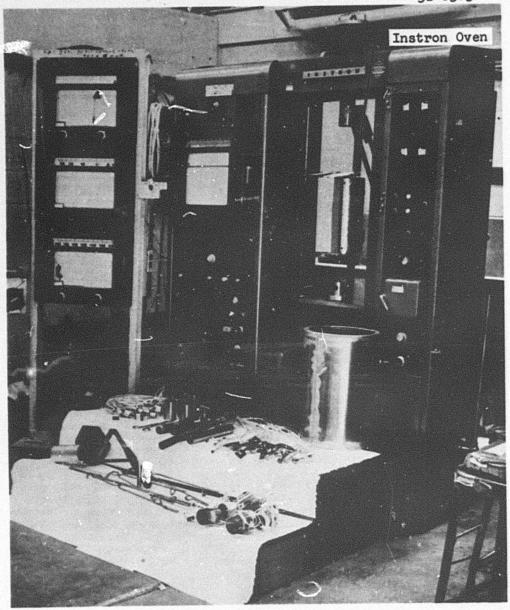


Figure 2-20 Tensile Test Instrumentation

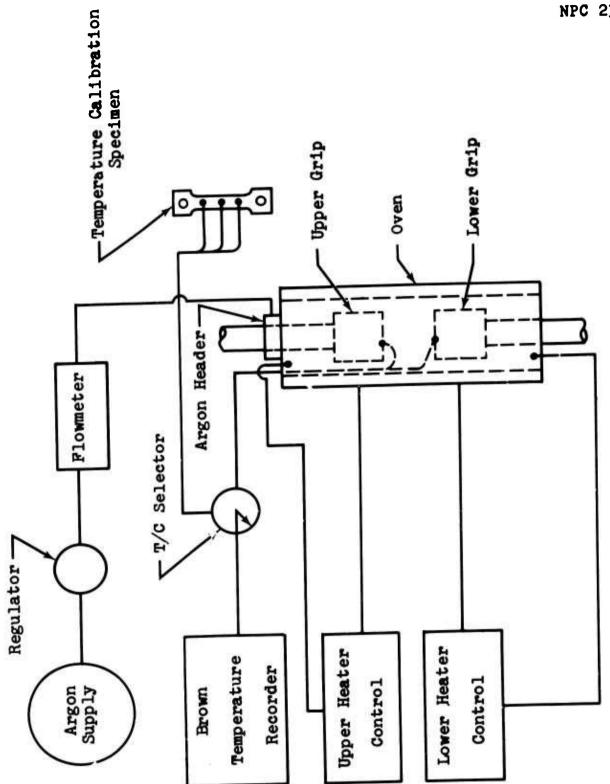


Figure 2-21 IML Oven Control Setup

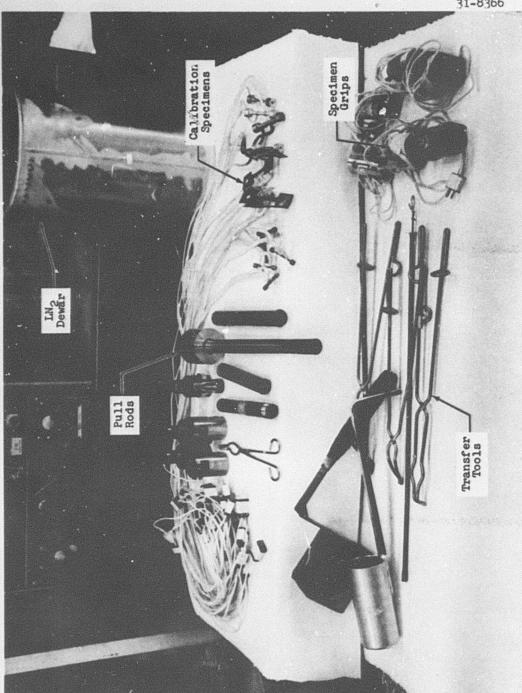


Figure 2-22 Accessory Hardware for Tensile Tests

of specimens at LN2 temperatures.

The irradiation dewar in its storage area can be seen in Figure 2-23. The irradiation dewar was placed in the IML hot cave and the cave shielding augmented with concrete blocks. In addition, 4 in. of lead shielding was placed in front of the dewar and on top of the steel dump valve at the rear of the dewar to further shield personnel when removing specimens from the dewar.

2.2.2 Resistivity Tests

11

Resistance measurements of one Inconel 718 wire specimen were made during irradiation. After irradiation, resistance measurements of two Inconel X-750 specimens and one Inconel 718 specimen were made at LN₂ temperatures before and after various annealing treatments.

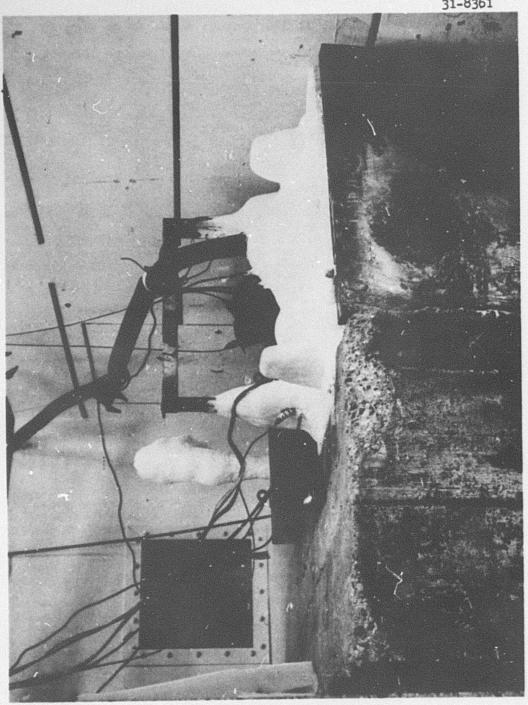
The resistance measurements made during and after irradiation were accomplished with the instrumentation setups represented in Figures 2-24 and 2-25, respectively. A schematic diagram of the resistivity bridge circuit is presented in Figure 2-26. The same resistance bridge circuit was used for measurements during and after irradiation. The test instrumentation and LN₂ dewar are shown in Figure 2-27.

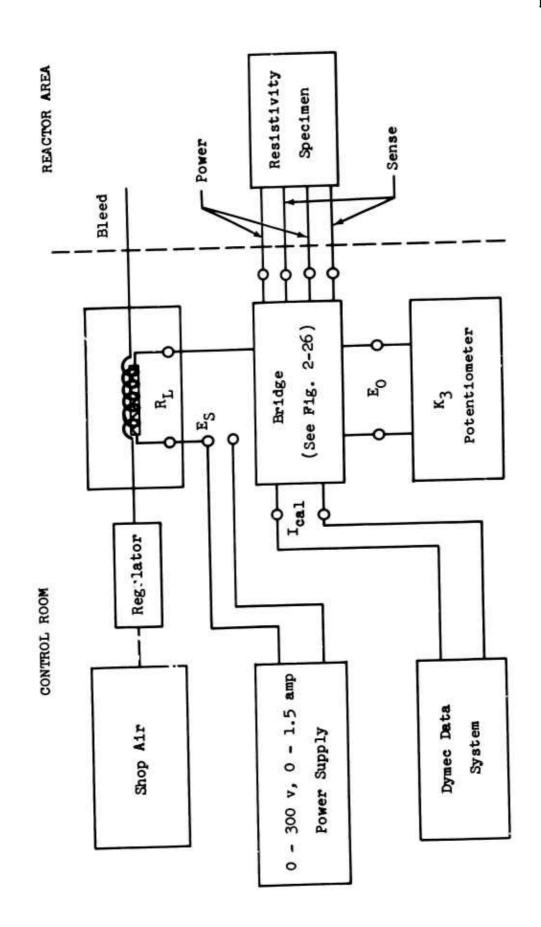
The location of the resistivity specimens in the aluminum framework of the north irradiation dewar can be seen in Figures 2-9, 2-10, and 2-11. The actual specimens were not available when the picture was taken and dummy specimens were substituted.

2.2.3 Steel-Spring Specimen Test

The location of the springs on the aluminum framework of

NPC 22,756 31-8361





100

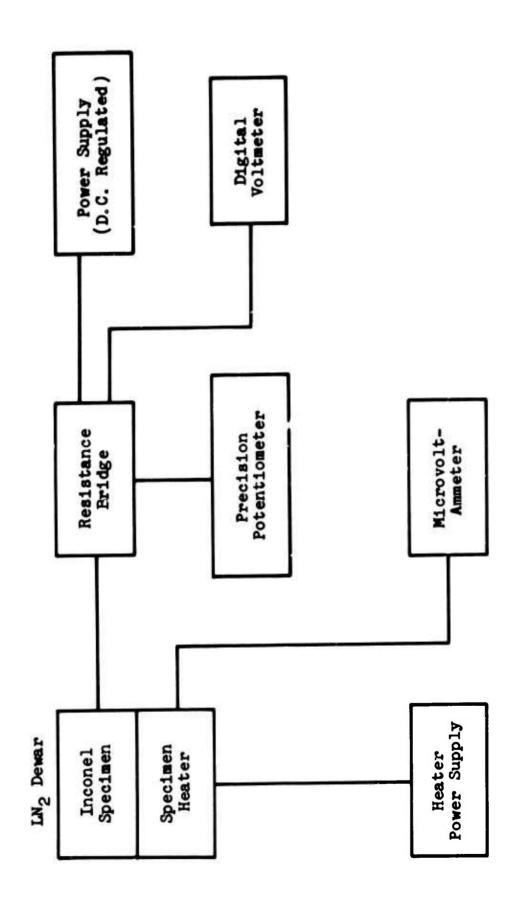


Diagram for Postirradiation Resistivity Test Figure 2-25

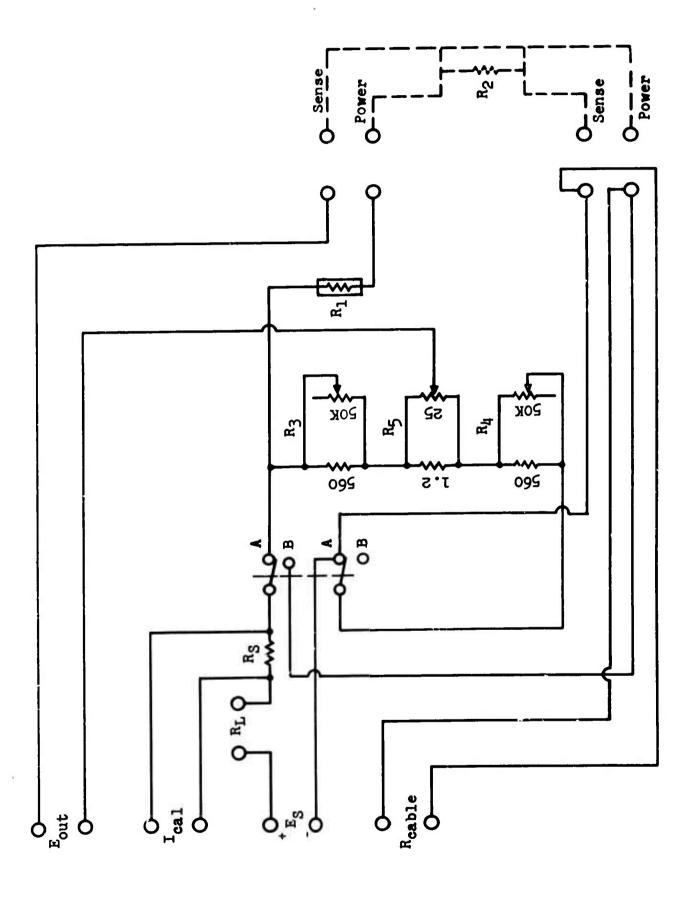


Figure 2-26 Schematic of Resistivity Bridge Circuit

NPC 23,282 31-8432

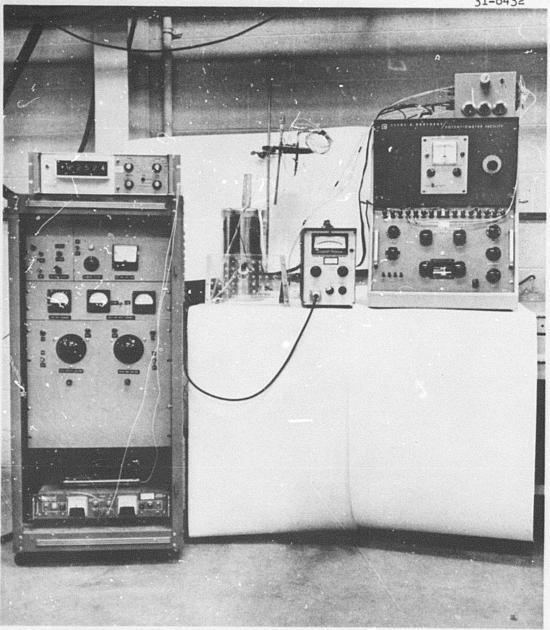


Figure 2-27 Resistivity Test Instrumentation

the north dewar can be seen in Figure 2-10. Only three specimens were available at the time this picture was taken; a fourth spring was added at a later date.

The Model TT Instron tensile testing machine used for the tensile tests was used in this test to determine the spring constant. Dimensional measurements on the springs were made with a Luckin micrometer mounted in a test jig.

2.2.4 O-Ring Seal Tests

Two test fixtures were supplied by WANL. One fixture was mounted on the front dosimeter rack inside the east LH₂ dewar in the gaseous hydrogen. The other fixture was enclosed in an aluminum capsule and mounted on the outside of the east LH₂ dewar (Fig. 2-28). The capsule after disassembly is shown in Figure 2-29.

The encapsulated fixture had one Cu-Cn thermocouple placed inside the capsule. The output of the thermocouple was monitored continuously with a Minneapolis-Honeywell multipoint recorder. Aluminum tubing, 1/4-in. in diameter, was run from a bottled helium and hydrogen supply to the capsule and from the capsule to the north hydrogen vent stack. A sufficient amount of hydrogen gas was bled through the capsule during irradiation to maintain the test specimen in a hydrogen environment. Helium was used to purge the system before and after hydrogen flow. A sketch of the experimental setup is shown in Figure 2-30.

2.2.5 Cemented-Orifice Tests

The cemented-orifice specimens were placed in two aluminum containers approximately 1 in. thick by 7 in. ep by 7 in. wide.

NPC 23,283 31-8351

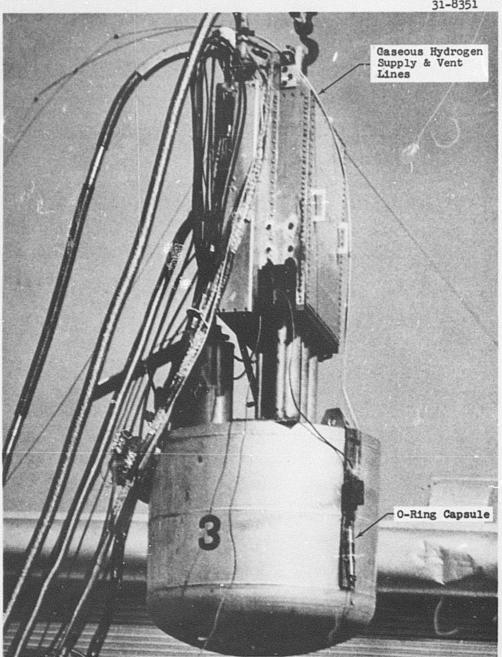


Figure 2-28 O-Ring Capsule Irradiation Configuration - East Dewar

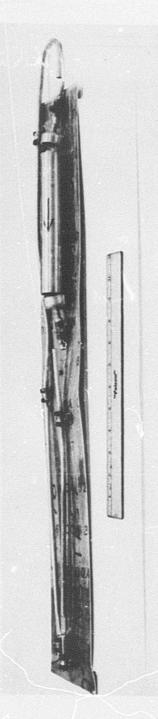


Figure 2-29 Dismantled O-Ring Test Capsule

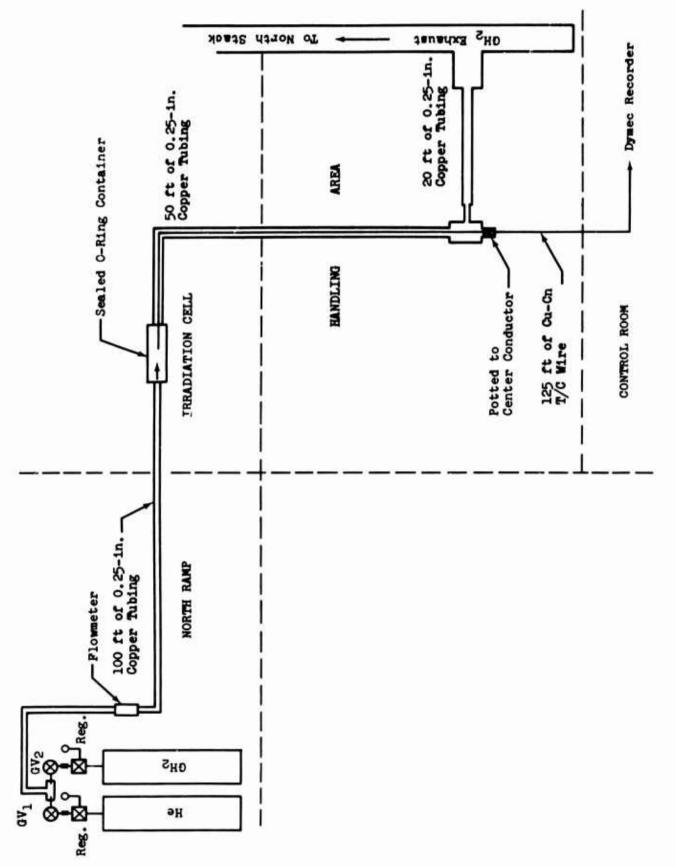


Figure 2-30 O-Ring Experimental Setup

One container was mounted inside the north dewar below the LN₂ liquid level. This container was mounted on the rear centerline of the graphite-tensile-specimen tray. The other container was mounted on the front center of the west side of the north dewar. One Cu-Cn thermocouple was installed in the container and its output monitored during the irradiation with a Minneapolis-Honeywell multipoint recorder.

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III. EXPERIMENTAL RESULTS

3.1 Tensile Tests

3.1.1 Data Presentation

The integrated neutron fluxes received by the tensile specimens are presented in Table 3-1. Detailed dosimetry data on these specimens are presented in Appendix B.

Three irradiation specimens and three control specimens of each material type were generally assigned for testing under each test condition. A number of specimens of each material type were assigned as spares. After all specimens except the spares had been pulled, a preliminary analysis of the data was made and the spares assigned to test conditions exhibiting questionable data. Table 2-2 lists the actual number of specimens, including spares, that were tested under each test condition. The results of all tensile tests are tabulated in Tables 3-2 through 3-8 and presented in bar graph form in Figures 3-1 through 3-7. Table 3-9 identifies the specimen numbers associated with each bit of data presented in Tables 3-2 through 3-8. An explanation of column headings is given in Table 2-1.

The tensile data on specimens pulled during the strain-rate study are tabulated in Table 3-10. The specimens are identified by number in Table 3-11. No analysis of the effect of strain rate on the properties of the specimens was made by GD/FW personnel. The original Instron charts for this test have been forwarded to Westinghouse for analysis by Wellinghouse personnel.

Table 3-1 Integrated Neutron Fluxes Received by Tensile Specimens (10^{17} n/cm², E >1 MeV)

Specimen		Ref.								Tes	Test Condition	ditio	ç									
			- 14	ij	Bi	DA1	DA1 DB1	- 18 ₀	Par,	D _{C1}	DC1.	ā	PDI,	E _{C1}	EA1	Ect F	r.	F11	GA	011	H	H11
Incomel 718,		3P 7.5		7.5	7.5-7.5 7.5-7.0	6.0	5.5	6.0 5	5.5	7.0-7.5	5.0	6.0 5.0		7.0-7.5		•	1.					
Inconel 718, Type 1		1P 10 10 NI		8.5	8.5-9.5					8.5				8.5-9.5								
Incomel 718-W, Type 3		3P 8.0		500	6.5-7.0																	
Inconel X-750, Type 1		11P 7.5		6.5-8.5	6.5-7.5					6.5-7.9						5.5						
Inconel X-750, Type 3		3P 6.0		6.0	6.0-7.5				5,5,	5.5-6.5		-	3141	5.5								
AISI 301-CW, Type 3		3P 6.5 3N 6.5		6.0-7.0	6.0-6.5	6.0-6.5 6.0-7.0								9.9	00							
S AISI 303-Se, Type 1		1P 8.5		6.5-8.5	6.5-8.5	6.5-7.5								7.0	6.5							
A1 2219-T6,		3P 8.5		6.5-8.5	6.5-8.5	8.80								80.0	8.0							
A1 2219-T6-T.		2P 7.0	7.5-7.5	4.5-6.5	4.5-6.5			***					-									
A1 2219-T6-R, Type 2		2P 8.5 2N 8.5		00.8	8.0-8.5																	
Beryllium, Type 4		4P 5.0	5.0-7.0	4.0-7.0	000											ind trace as						
Ti A-110-AT-E11, Type 2	original transfer	2P 6.0		5.5-6.5	5.5-6.5				-													
Incomel 718 Strain Rate Study		4									w a La	1				ω	1.0 7	8.0 7.5-8,0	0.0	8.0	9.0	8.0-8.5

. See Table 3-9, under same Ref. No. and Test Condition, for identification of specimen.

Tensile Test Results Table 3-2

									Inconel	1 718,	Type	3								
Property	Ref.									T	Test Con	Condition								
	2	0	AI	10	3	B1	PA	DA1	DA1'	DB	DBI	, 18c	D _C	10g	DC: 1	D,	Dri	Don'	101	102
UTS (ksi)	25635	270.2	270.6 269.8 268.2	269.3 265.03 268.0	218.7	214.7 211.1 213.6 215.3	200.0	200.8	200.7	195.1	194.3	1-0	940	QQUID.	4.60	154.7	Oω	152.7	273.0	272.4
Average		270.0	269.6	267.4	218.9	213.7	198.2	200.0	200.2	187.7	193.6	190.3	190.3	180 4	188 3	154 7	1000	161 7	0.000	2 000
TYS (ks1)	4384	213.1	233.1 234.6 228.5	219.6 216.4 216.8	184.7 184.1	189.4 185.4 186.7	162.7	166.6	168.7	162.6		10-		153.2	159.4	- 0	100	- 44	u cue:cu	221.7
Average		212.4	232.1	217.6	184.9	188.2	164.9	166.6	168.5	162.7	167.1	169.0	ובא ה	0 33	4 63	04		-		
NTS (kal)	24,43,613	243.5	257.9 261.0 260.8	257.4 247.3 254.1 251.7 251.5	220.5 219.1 218.0	218.5 217.6 219.9						+ and	77.77			2	83.2	1,55	2558.3 247.9 246.5	241.9 238.0 236.4 256.4
Average		242.6	P59.9	252,4	219.5	218.7						Γ	80.5.1	80.0		ľ	R 2 R	Γ	0 030	1 . 6 %
NTS/UTS	Max	0.90	0.97	0.97	1.01	1.04							000	00	$\overline{1}$		1.23		J. Uni	0.91
Average		0.90	0.96	0.94	1.00	1.02							1.00	0	T		1 00		0,00	0.07
Percent Elongation	254351	12.9	13.1	10.2	10.5	12.4 11.4 11.5	10.0	10.1	9.82	9.6	9.56	8.8 28.74	7.79 8.71 7.75	8.00.8	9.39	6.23	6.6	3.64		and and and and
Average		12.5	12.6	13.0	10.8	11.8	10.7	9.6	10.71	9.38	9.68	8.40	8.08	8.46	9.33	6.23	3.45	3.60	14 06	9
Percent Reduction in Area	18 M& B	19.0	20.3 22.7 21.5	11.2 17.6 13.9	26.6 20.6	19.5 24.1 22.0 28.9	22.4	18.8	18.8	22.4	22.2	22.97	12.8		22.3	8.36	5.93	29	17.0	15.46
Average		17.1	21.5	14.2	24.4	23.6	23.6	21.3	21.14	24.4	21.6	22.92 15.0	_	80.8	19.45	8.36	6.16	7.63	21.30	17.61

21.30 17.61

. See Table 3-5, under same Ref. No. and Test Condition, for identifiestion of specimen.

Table 3-3

Tensile Test Results

Inconel 718, Type 1

	ပ	207.3 205.7 213.3 205.6 199.8	206.3	196.4 196.2 199.8 191.7	195.7	159.3 153.8 168.8 166.2	162.0	0.85	0.79	0.98 0.89 1.20 1.02	96.0	9.89 9.36 10.63 10.14	• •
_			· · · · · ·										1
	ECI	270.8 276.3 272.4	273.2	210.0 211.6 203.8	210.2	307.3 327.2 312.7	315.8	1.21	1.16	12.3 12.9 15.7	13.7	21.1 31.0 25.4	25.8
	ក ប	267.7 274.9 273.0	271.9	203.1 212.9 205.0	207.0	324.3 318.8 306.6	316.5	1.21	1.16	15.0 14.2 13.8	14.4	26.6 23.9 26.8	25.8
	DCI	193.9 193.0	193.5	156.6 155.3	156.0	227.0 241.3 223.8 228.9 224.4	229.1	1.25	1.18	9.83 12.6	11.2	13.9	16.2
no	Dc	188.3 192.2 192.4 193.7	191.7	154.9 159.9 155.7 159.6	157.5	228.5 240.8 244.5	237.9	1.30	1.24	10.9 8.11 11.8 8.46	9.82	24.0 19.9 19.7 4.8	17.1
Condit1	B1	213.7 217.4 210.8 217.5	214.4	185.2 172.5 188.4 188.9	188.5	291.1 296.2 284.0	290.5	1.41	1.35	13.3 12.1 9.37 11.2	11.4	28.5 28.5 28.5 22.5 22.5	26.7
Test	В	216.6 221.6 215.3	217.8	177.6 183.2 177.8	179.5	276.3 279.2 282.3 285.7	280.9	1.33	1.29	10.5 10.6 11.6	10.9	25.4 24.3 29.5	3, 4
	C1	276.3 270.0 272.4 275.5	273.6	225.7 220.2 218.6 225.8	222.6	325.7 333.5 349.5 344.1	338.2	1.29	1.24	12.7 10.7 16.7 10.7	12.7	24.1 30.8 29.4 20.6	26.2
	A1	267.5 272.4 265.1	268.3	234.1 234.9 236.5	235.2	329.4 354.7	342.0	1.34	1.27	6.57 13.2 5.31	8.35	25.3 30.7 18.2	7 7
	၁	256.8 271.2 275.7	67.9	205.0 203.9 211.3	206.7	314.8 298.6 324.4	312.6	1.26	1.17	7.49 12.9 17.1	12.5	25.5 19.7 26.7	24.0
Ref.	No.	1P 2P 3P 4F 5P		11P 22P 33P 45P 57P		11N 22N 33N 54N		Max Min		1P 2P 3P 4P 5P		11P 32P 47P 57P	
Property	Measured	UTS (ks1)	Average	TYS (ksi)	Average	NTS (ksi)	Average	NTS/UTS	Average	Fercent Elongation	Average	Percent Reduction in Area	Average

See Table 3-9, under same Ref. No. and Test Condition, for identification of specimen.

1.15

9.80

96.0

9.89

12.47 8.28 6.13 11.07

6.68 8.50 8.73 8.73

85.18 36.18 36.18

3.57.08 8.03.08

67.6

7.90

5.47

5.15

0.89

9.8

0.77

0.78

11.16

0.89 0.62 0.84 0.71

0.8 1.85 1.02 1.02

9888

0.0

0.93

0.0 689 689

0.85

149.6

148.2

165.0

165.8

163.8 140.8 144.2

145.5 146.6 1148.6 152.2

149.2 185.1 160.6

154.9 172.5 169.9

167.8

162.6

205.8

209.4

168.9

173.3

214.2

211.6

170.6 169.3 170.3 165.3

179.7 178.9 171.8 164.5

216.5 211.6 209.1 219.4

206.2 202.7 220.4 216.9

Bi

Test Condition

A

Inconel 718-WS, Type

165.4 162.6 185.4 158.0

168.9 167.3 162.2 152.7

209.4 201.8 201.8 210.0

206.2 202.7 216.6 211.8

Table 3-4 Tensile Test Results

0.87 96.0 000 335.5 0.0.0 25.7 209 533 922 207 Eci 888 0.87 r- 10:00 9.84 0-0 40000 173.7 000 000 207. 122 123. 203 2728 3330 × S 0.83 0.90 138.3 07.1 0 4 -1 - OI 124.9 0000m 124.0. 127. 255335 0.93 0.89 146.0 142.7 400 127.4 Inconel X-750, Type 96.1 7.7.7 13.17 888 130.5 Test Condition Do 163.6 163.8 165.1 165.1 1.04 0 174.2 165.1 171.2 1.07 16.8 F04410 18.1 19883 171.3 322333 129. H 165.5 0.93 888 213.2 166.4 0,40 107.7 159.4 147.0 162.8 152.5 18.9 :8.8 37.4 33.2 33.0 35.8 4 00 288 155. • 213.8 213.0 212.9 0.97 96.0 153.6 יםיחמי 153.2 205.1 00 - m 205. 888 23 # mm 5 1.02 211.8 210.8 216.3 213.0 169.7 168.2 172.5 1.0 217.0 218.4 217.8 217.8 170.1 22.1 21.9 32.7 A 213.1 210.8 210.3 206.0 210.1 0.91 35.1 187.0 183.4 177.8 0.87 maau 125.1 82.7 27.3 27.0 35.55 2883 212.9 203.5 204.7 212.0 203.3 120.0 236.1 122.6 116.6 719.1 240.0 22.0 32.4 41.0 41.0 Jo Ect 22 1dentification 207.6 207.6 207.2 205.9 217.1 210.0 120.7 1118.7 1119.8 243.2 1.30 15.3 16.3 33.0 .05 202.3 22.5 4 4 50 50 150 25 Sagan BC 147.3 1040 Nam 1.22 mag 19.8 5,683 288 93. 180. 130 0000 6 DCI for 1.19 1.21 153.3 9. 800 93.3 200 111 000 11.1 Condition, 151 666 1837 184 27.6 DC Condition 169.3 166.8 1.51 450 240.9 16.35 0 8 33.3 31.7 Type 1 19.1 133 131 241 800 B1 Test 172.1 166.5 168.5 1.32 1.29 910 216.3 15.8 16.3 x-750, 217.6 Test 900 999 105. Sam m and 212.5 216.2 213.6 207.9 213.8 152.9 273.6 282.8 585.1 279.6 1.37 35.4 Inconi 280.3 183.55 21.1 4000 No. 151. 8888 5 Ref. 203.0 213.7 206.0 167.2 169.2 163.1 N in 000 299.3 17.1 33.4 19.1 209 166 301 5.0mc ¥ 1.19 33.5 under se 212.1 212.7 208.8 208.6 210.5 wa-4 7.611 000 244.3 25.0 54.9 3833.7 121 238 0 3-9, Ref. Max 50 50 50 B 0000000 SASSERSE 255555 255255 Table Percent Property Percent Reduction in Area Average (ksi) Average Average NTS/UTS Average Average See TYS (Kal) NTS KSI)

Table 3-5 Tensile Test Results

-	Measured No.	80000	Average 292.8	TYS 2P 163.8 (Kst) 3P 165.1 4P 165.1	Average 165.2	NTS 2N 208.7 (*s1) 3N 208.6 (*s1) 4N 209.2	Average 207.3	1. TS/UTS Max 0.72 Min 0.59	Average 0.71	Percent 2P 19.9 Elongation 3P 20.5 4P 19.5	Average 20.4	Percent 2P 35.7 Reduction 3P 30.7	
		0.00	290.4	177.8 173.4 177.1	176.1	214.4	214.3	0.75	0.74	20.1 21.0 19.6	20.2	35.4	STATISTICS NAMED IN
AISI 301	8	292.9 294.1 292.5 297.6	294.3	172.0 175.5 172.0 175.4	173.7	217.5 216.7 214.5 214.5	7.912	0.74	0.74	20.5 19.4 20.4 19.8	20.0	%## 36.5.5 36.5.5	140, 210,000,000,000,000
301-CW, Ty	a lear	4886	186.9	149.5 153.0 148.6 153.1	151.0	191.9 191.0 191.0	191.2	1.03	1.02	18.8 18.7 19.1 20.3	19.2	37.0 35.6 41.4 46.0	
Type 3	Bi Bi	189.8 1145.5 190.3	183.8	156.2 156.2 155.8 156.8	156.3	194.5 193.9 194.5 192.2	193.8	1.05	1.03	16.2 16.2 16.2 18.3	17.4	26.2 37.6 37.3 43.7	
8	Da	-1040	145.8	134.1 132.3 130.8 135.0	133.0	159.1 158.7 158.2 155.6	157.9	1.10	1.08	4.36 10.4 4.09	4.14	19.6 18.9 25.2	The second secon
	Dat	040	147.9	137.1 133.6 137.3	136.0	158.4 160.5 163.3	160.7	1.11	1.09	3.83	4,01	19.5	
	EA	2.0.0	291.4	192.0 190.2 189.5	9.061	228.3 226.2 226.3 228.3	227.4	0.784	0.780	22.7 22.4 23.7	22.9	333.8	
	EAT	300.4 289.7 291.7 292.8	293.7	194.5 194.5 195.2 195.2	193.3	230.2 231.3 232.6 233.7	232.0	0.80	0.79	20.6 21.6 21.6 21.9	21.3	70.8 38.5 36.0	
		165.6 157.4 173.3	165,4	83.6 93.0	85.4	192.7 190.6 185.8 194.9	191.0	1.24	1.15	4833.0 65.80	43.1	504.00	
	A1	168.8 175.5 174.6	173.0	112.6 116.2 116.2	115.0	230.1 217.7 235.5	228.2	1.40	1.32	29.0 37.0 37.0	34.3	8.044 8.094 8.09	
	10	172.1 176.2 166.4 171.9	172.4	106.0 112.2 99.6 102.7	104.4	208.2 209.0 207.1 209.9	208.5	1.26	1.21	433337.7 43533.7.7	38.7	45.8 38.8 41.3	
Test	В	92.2	92.7	49.8 49.8 47.7	49.1	124.1 121.1 119.1 121.9	121.6	1.35	1.31	30.9	31.2	43.2 41.0 50.4	
t Condition	B1	96.0 97.1 94.7	96.0	61.1 67.3 61.0 57.3	60.8	134.6 130.9 130.5 130.5	131.4	1.42	1.37	28.88.88 64.7.46	26.0	37.8 42.2 43.7 67.6	
13 be 1	V	72.6 74.2 72.7 70.9	72.6	3333	33.3	98.0 94.0	97.2	1.39	1.34	25.0 25.0 25.0	23.6	35.2 26.32 66.8	35 de
	DAL	73.6 73.1 72.3	72.2	2000 2000 2000 2000 2000 2000 2000 200	35.6	101.2 104.1 98.4	101.2	1.49	1.40	22.9	22.0	40.1 37.5 30.8 21.3	
	E _A	170.1 166.0 169.7	168.6	90.5	87.6	176.4 193.4 193.4	184.2	1.17	1.09	50.7	48.1	53.0 50.7 49.3	
	EAI	171.2	168.3	87.5	89.9	198.6 198.6 193.6 195.0	191.9	1.19	1.14	50.03	45.2	45.3	

Table 3-6 Tensile Test Results

					ur ccr3-10)					
Property	No.*		. 4	5	α	Test Cond	Condition	c	C:	L.
		,	W.	5	n	12	NA VA	D Ai	A A	A.
UTS (ks1)	1.2 2.2 4.7 4.7 4.7 4.7 4.7 4.7 4.7 4.7 4.7 4.7	75.1	74.7 75.0 75.5	74.1 74.8 73.7	57.8 60.2 61.0	58.8 59.4 59.6 59.0	21.8 21.6	21.5 21.8 22.1	67.1 68.8 58.3	68.1 58.0 55.7
Average		73.1	75.1	74.2	59.7	58.4	21.7	21.8	68.1	é7.1
TYS (ksi)	11 22 45 47 57	51.5 46.9	62.6 63.1 63.6	51.7 52.4 51.3	41.0 42.8 42.3	41.8 42.3 40.3 41.6	20.4	20.3 20.3 20.3	7.04 0.14 0.64	41.3 40.4 37.8 38.1
Average		49.2	63.1	51.8	42.0	41.5	20.4	20.5	L.04	39.4
NTS (ksi)	N K W N N	67.9 67.2	75.9 74.6 75.7	68.0 69.3 65.9 68.3	55.6 56.0 57.8	54.6 50.7 55.6 53.2	26.6 27.0 26.6	27.4 26.8 27.3	58.6 60.0 77.2	78.00 1.00 1.00 1.00
Average		67.5	75.4	67.9	5.95	53.5	26.7	27.2	58.€	57.4
STU/STN	Max	0.95	1.02	0.94 0.88	1.00 0.91	0.98 0.85	1.25	1.27	0.69 0.83	0.0
Average		0.92	1.00	0.91	0.95	0.92	1.23	1.24	0.86	0.86
Percent Elongation	11 32 44 54 47	9.52 6.22	8.80 7.82 7.28	10.4 9.43 10.4	6.54 7.64 6.80	6.97 6.84 6.09 7.11	7.60	7.91 6.84 7.82	13.5 13.5 13.8	11.0 10.2 10.8 13.4
Average		9.1	7.97	10.1	6.99	6.75	7.60	7.52	13.0	11.3
Percent Reduction in Area	1 4 4 4 4 4 4 4 4 4 4 4 4 4 4 4 4 4 4 4	18.5	23.6 23.6 4.6	18.0 21.7 17.7	13.0 10.5 10.4	16.4 22.7 19.4 26.9	23.7 20.6	37.6 31.5 38.5	22.5 19.2 18.9	23.0 23.6 24.8
Average		14.8	23.5	19.1	11.3	21.3	22.2	25.0	20.2	22.9

Table 3-7

Tensile Test Results

N 9.40 1.29 1.25 62.4 61.9 61.1 46.2 47.5 45.4 12.7 ω £6.3 75.9 77.3 10.9 10.7 12.1 15.4 Type 61. m Test Condition 2219-TG-Radial, $\frac{1.29}{1.19}$ 7 9.8 73.7 52.8 54.3 10.7 74.7 92.3 W 4 13. 94. 89. 13. ಚ 10.7 7.83 6.89 2004 605 400 605 5.62 8.47 1.39 3 30.5 111.5 105.9 98.7 78.3 404 045 888 25.8 105. 7 ¥ 12.98 1.32 1.26 16.7 52 93.0 73.6 11.7 75.1 72.1 51.7 mm 00 4.8 95. 52. ပ 8.80 7.84 5.92 7.83 7.00.00 0.00.00 0.00.00 1.22 8.02 9.03 61.5 60.3 61.5 61.5 61.5 1.08.7.4 1.08.0.7.4 1.09.0.7. 24.53.50 74.54.57 74.59.57 72.2 4 61. 47. H 9.35 1.28 10.33 8.13 7.27 8.07 7.53 15.36 7.86 7.84 7.83 22 27 75.5 62.7 63.1 62.1 61.4 62.1 46466 **ω**α του 78933 27000 ω m Condition Ø 6.40 6.27 6.87 6.87 7.73 4.20 22 67 71.6 72.0 74.0 72.1 73.2 69.7 25.55.55 25.55.55 25.55 92.0 91.1 91.5 82.3 84.5 88.1 72. 54. Ø \ddot{c} 2219 -T6-Transverse, 2.14 1.95 2.71 1.21 4.27 27 888 80.7 85.0 92.6 96.5 91.4 500 0 26.6 79. שיטים તં 388 Ai 1.35 7.00 8.27 7.00 7.00 9. 5.73 22 2.5.4 2.82 2.92 53.5 48,088,08 0000 m Q a 72. 52.27 53. 88 1 ပ A Max Min No Se 438446 64466 97449P SON FABRIA Percent Elongation Percent Reduction in Area Property Measured NTS/UTS Average Average Average Average Average Average NTS KB1) WTS (kst) TYS (ks1)

1.27

16.3

78.1

61.7

450

61. 61.

Bi

47.3 47.0 46.2 જ

in m m

125.

10.7 14.4 7.84

11.0

specime of for identification Condition, Test and . 80. Ref. ваше under 3-9, Table See

Table 3-8

Tensile Test Results

	Property Ref.		UTS 2P 2P 3P (ksi) 4P 5P 6P	Average	TYS 2P 2P 3P 4P 4P 5P 6P 6P 7P	Average	NTS 2N 2N (ksi) 4N 5N 6N	Average	NTS/UTS Max	Average	Percent 3P 4P 4P 5P 5P 6P 6P	Average	Percent 3P Reduction 4P In Area 5P	Average	* See Table 3-9.
		U	23.28 29.66 38.7 47.0	36.2	23.2 31.7 29.6 38.7 46.7	36.2	6140 8014 8008 8014 8008	6.9	0.30	0.19	0000000	0.0	0000000	0.0	under
Beryllium		Ai	12.7 9.76 10.3 10.3	10.8	12.7 9.76 10.3 10.3	10.8	6.45 4.32 7.62	6.29	0.78	1 0.58	00000	0.0	00000	0.0	Ĵ.
1, Type 4	Test Co		40.2 43.1 26.5 28.0 36.4	37.3	450.2 45.1 26.5 28.0 41.8	37.3	8.3 10.1 9.7 7.9	8.5	0.38	0.23	0000000		0000000	4 4	No. and T
	Condition	ш	49.4 51.9 50.7 52.4	51.1	41.4 42.1 41.0 41.1	41.4	36.1 29.6 31.2 27.7	31.2	0.73	0.61	0.00 0.36 0.44	0.39	1.45 2.17 1.44 2.16	1.30	Test Condition.
		Bī	49.4 51.1 50.9 48.2	6.64	43.7 42.8 42.5 42.5	42.9	29.6 31.0 32.6 32.4	31.4	0.68	0.63	0.43 0.68 0.68 0.68	0.62	0.0 0.72 0.72		for
		υ	184 152 163	183	178 177 176	177	248	245			122	18	131	72	identificati
I		A1	.1 .3 .197 .1 .195	.3 197	.3 7 194 .8 193	.6 195	.3 283	.0 227	336 1	.34	κ.ε.ι. 1444	۳.		75 27	5
Ti A-110-	Test		5.9	7.5	0 m 0	0	∞04	۲.	1.20	1.15	יםיםי	4.6	ma 6	7.8	
A-110-AT-E11,	Conditio	i	191.4 190.1 193.7	.61.3	87.9 187.3 191.1	188.3	225.5 235.1 224.3	228.5	1.24	1.19	9.3 9.3	12.0	233.7 2.1.4.7	29.7	
Type 2	u	В	121.6 120.7 120.7	121.1	119.4 119.1 119.1	119.2	188.3 188.0 191.5	139.3	1.59	1.56	15.9 17.3 16.4	16.5	35.2 35.2	37.1	
		ŭ	1283	129.0	128.1 128.1 129.0	125.ć	1000 5000 5000 5000 5000	193.2	1.52	1.50	133. 12.9 9	12.8	6.04 6.04 7.54	37.6	1

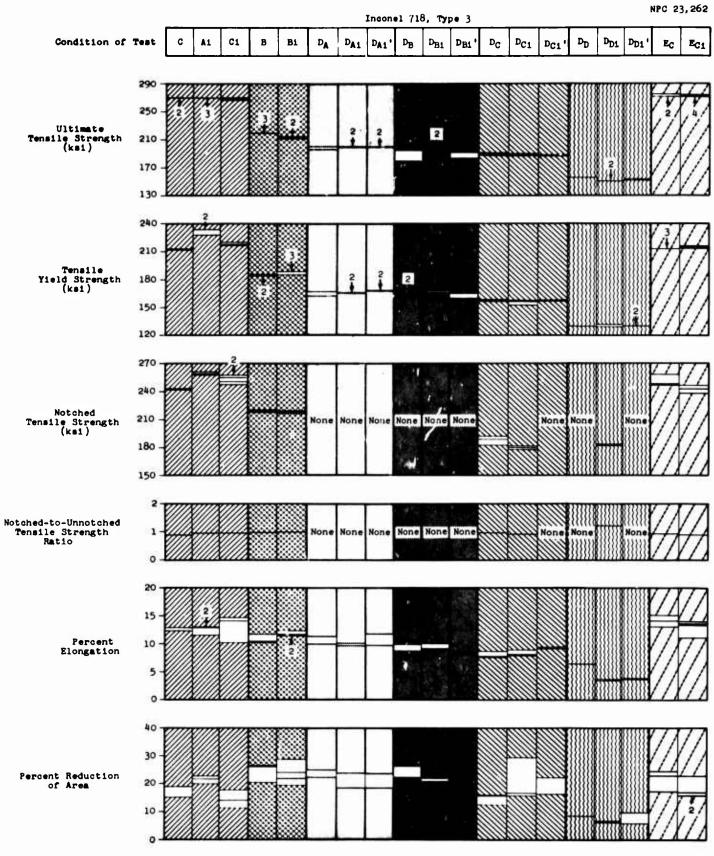


Figure 3-1 Summary of Tensile Test Results: Inconel 718, Type 3

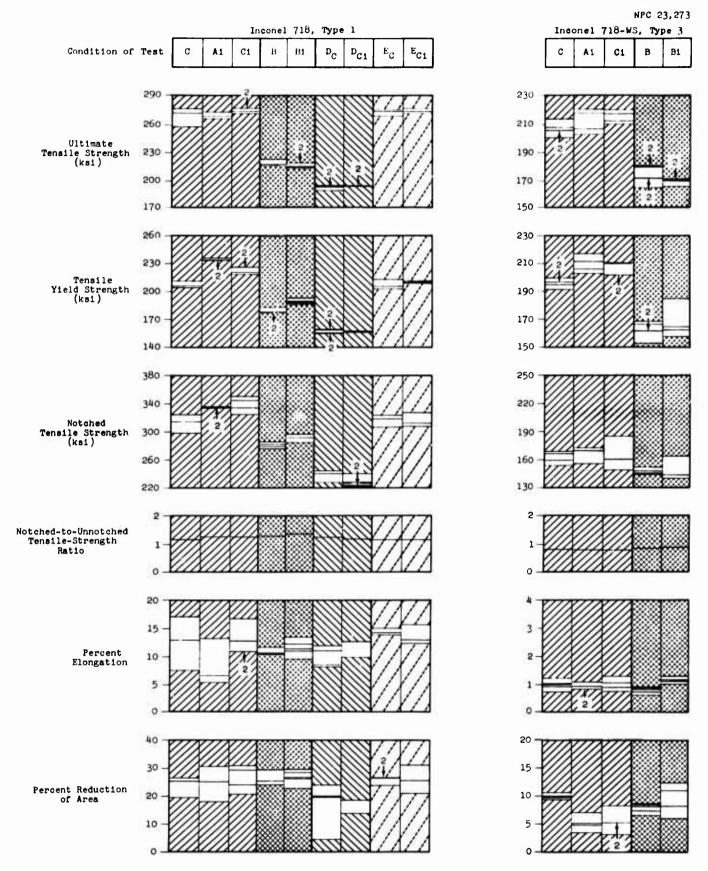


Figure 3-2 Summary of Tensile Test Results: Inconel 718, Type 1, and Inconel 718-WS, Type 3

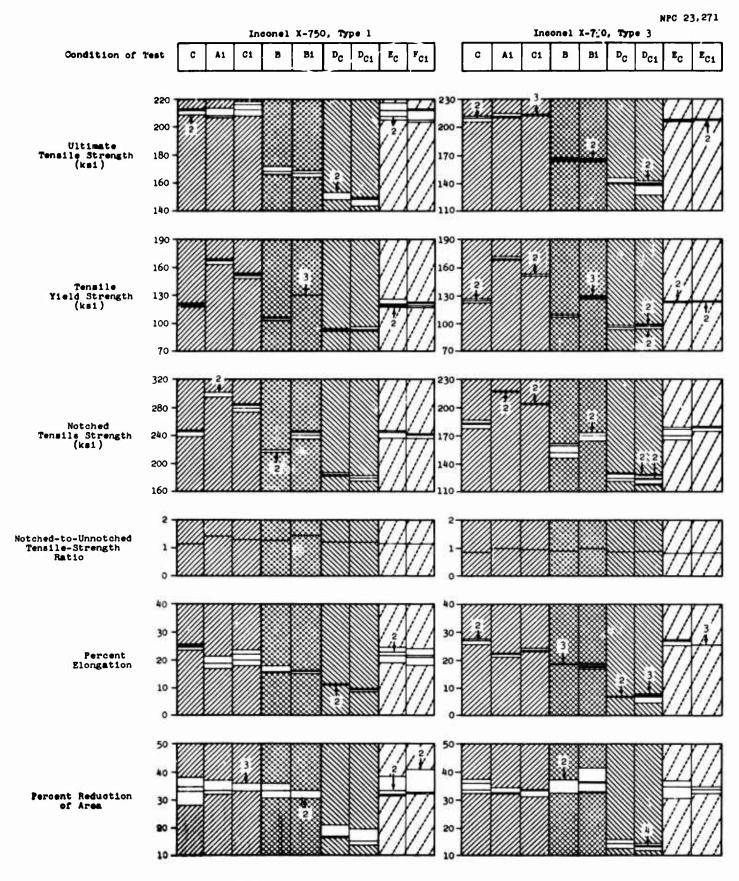


Figure 3-3 Summary of Tensile Test Results: Inconel X-750, Types 1 and 3

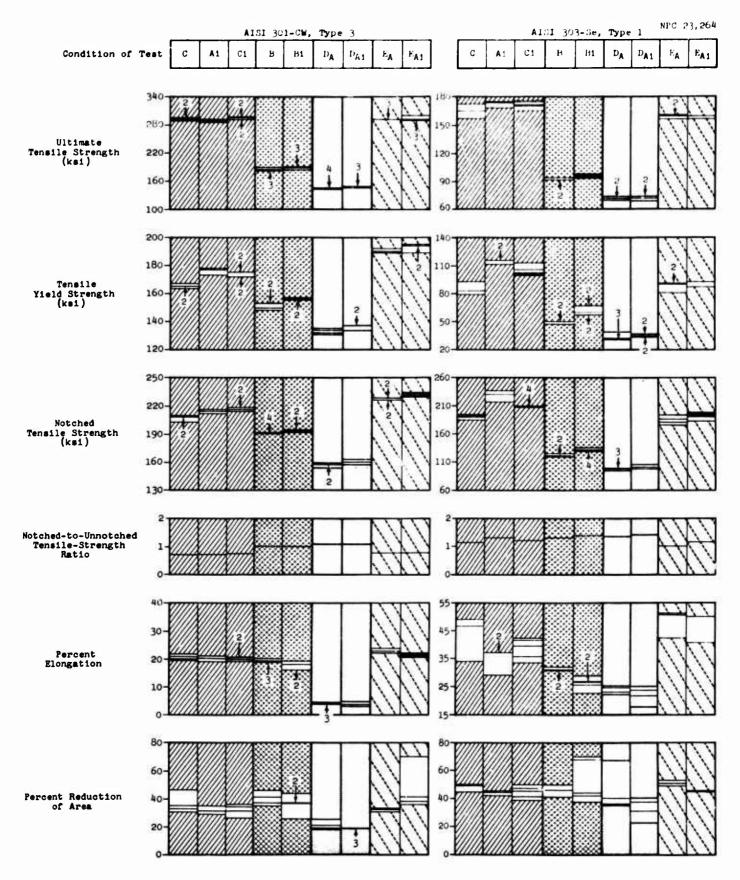


Figure 3-4 Summary of Tensile Test Results: AISI 301-CW, Type 3, and AISI 303-Se, Type 1

NPC 23,266

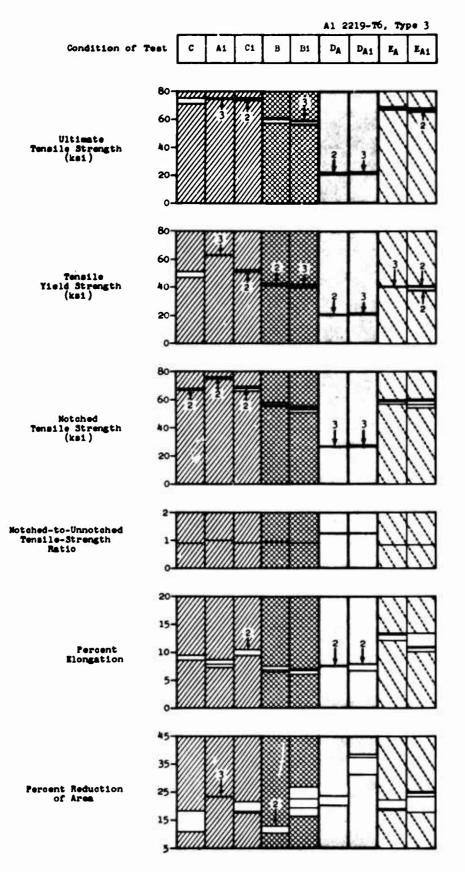


Figure 3-5 Summary of Tensile Test Results: Al 2219-T6, Type 3

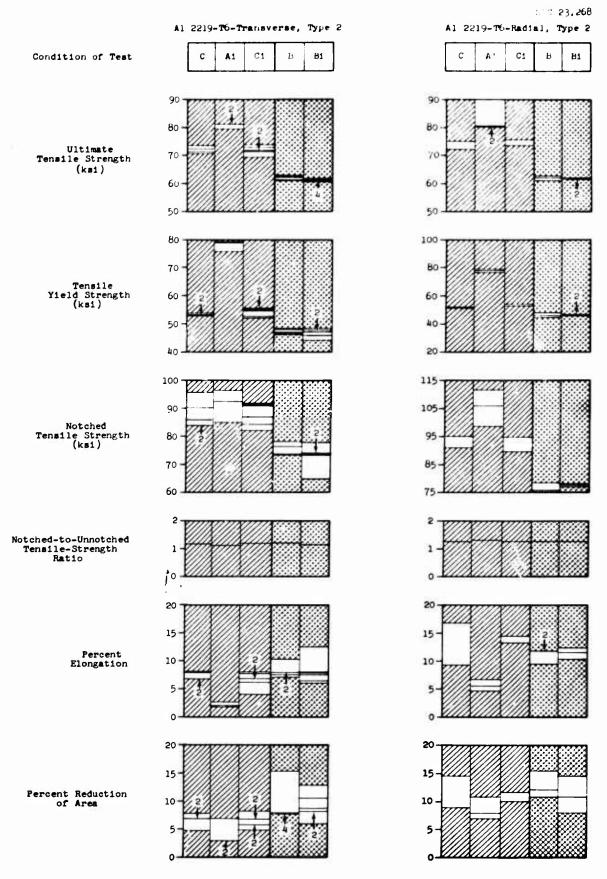


Figure 3-6 Summary of Tensile Test Results: Aluminum 2219-T6, Type 2, Transverse and Radial

NPC 23,269

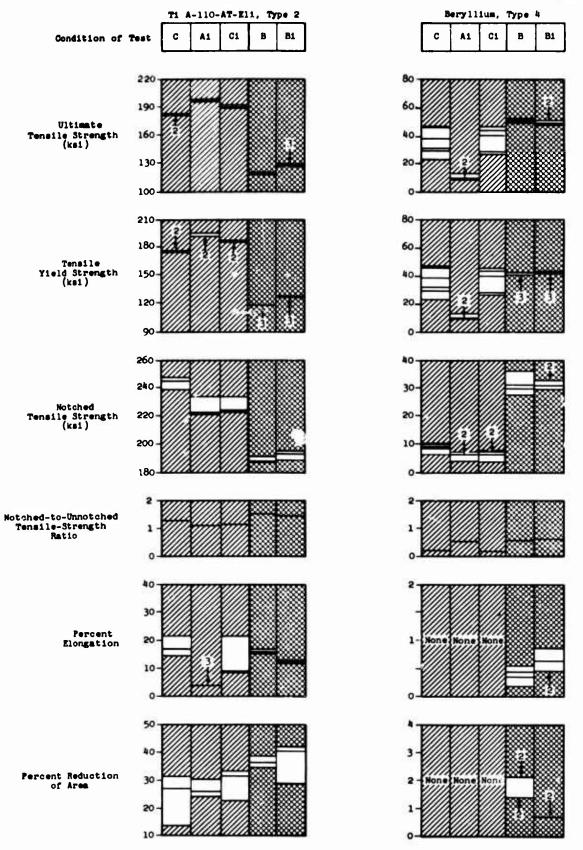


Figure 3-7 Summary of Tensile Test Results: Titanium A-110-AT-Eli, Type 2, and Beryllium, Type 4

Table 3-9
Identification of Individual Tenaile Specimens as to Material
Type, and Test Condition

Specimen Material	No.	-	XI	et	я	nr .	5 _A	T _{A1}	Pat	D _p ,	The C	B _{P1}	D _C	F _{C1}	PC1	r _D	T _{D1}	Da.	F.	FAI	e _c	Ect
	1F 2F 3F 5F	610 630	255 251 253	121	490 510 530	245 247 249 237	1250 1250	273		126e 3270	477 478	175 176	730 750 770	269 271 273	483 484	1890	481 462	*73			890 970 950	281 283 285 285 286 280
Inconel 718 Type 3	28 28 38 48 58	620 640	250 252 254	324 326 232 233	500 520 540	217 244 246 246							750 760 780	268 270 272			234 236				860 900	280 280
	1P 2P 3P 4P	1750 1770 1790	89 87 85	43 22 67 51	1690 1710 1730	61 63 65 57							1810 1830 1850 4650	69 37							1890 1910	39 41 53
Income1 718 Type 1	1P 2P 3P 4P 5P 1N 2N 4N 5N	1766 1786 1800	46 46	42 66 50 36	1700 1720 1720 1720 4680	5323							1820 1840 1860	70 38 68 58							1900 1920	10 52 56
Incone1 716-45	10 20 40 50 50 50 50 50 50 50 50 50 50 50 50 50	670 690 710 810	261 259 257 263	227 229 231 241	550 570 590 790 910	265 267 243 239																
Type 3	SN SN SN	670 690 710 810 830 700 720 820	268 258 256	226 228 230	58c 58c 60c 80c	264 264 266																
Income1	19 20 20 20	1510 1530 1550 3180	7 9 11	13 31 23 21	1450 1470 1490	25 27 29							1570 1590 1610	1 19 35							1630 1650 1670 1150 1200 1648 1660 1680	17 33 5
Inconel X-750 Type I	1N 2N 3N 4N	150d 154d 156d	8 12 10	14 6 24 16	1460 1480 1500	26 30 20							1586 1600 1620	18 34							1646 1660 1680	35
	14 4 4 4 4	1090 1110 1130 3130	183 185 387	133	970 990 1010	177 179 181 295 297							1210 1230 1250	341 347 367 367 367 367							133c 135c 137c	207
Income1 X-750 Type 3	19代表の大学の大学の大学の大学の大学の大学の大学の大学の大学の大学の大学の大学の大学の	1100 1120 1140 1420	382 384 386	330 332	980 1000 1000 1400	376 378 380 294							1220 1240 1260 1440	341 341 347 367 469 342 466 296 470							1340 1360 1380 3100	286 286 290
AISI 301-CW Type 3	11° 27° 31° 44° 110° 110° 110° 110° 110° 110° 110	2650 2670 2670 2670 2680 2660 2700 2680	311 313 315 316 312 314	299 301 303 355 298 300 302 156	2590 2610 2630 2630 2600 2600 2640 2640 2840	305 307 309 351 306 306 308 154	2710 2730 2750 2890 2720 2720 2720 2740 2760 2900	291 295 297 294 294 296											2770 2810 2730 2800 2780 2820 2920	157 197 14 15 18 19 19 19 19 19 19 19 19 19 19 19 19 19		
		1990 2010 2030	89 87 85	U-Alberta	1930 1950 1970	83 91 93 103	2900 2050 2050 2090 1190	97 99 77 75X											211C 213C 215C	73 101		_
AISI 303-Se Type 1	17 29 39 59 59 18 28 38 48 58	2000 2020 2040 1180	88 88	79 81 95 911 389 30 94 96 388	1946 1966 1986 1286	103 391 62 64 32 104 390	2068 2080 2100 2100 2100	75X 78 98 392											2150 2140 2120 1300	74 76 100 102 921		
A1 2219-76		30 10	427 429 431	209 211 213	130 150 170	territorio meterativo	250 270	197 199 201											370 390 410			
A1 2219-T6 Type 3	IP SP SP IN SN SN SN	50	438 430 432	208 210 212 188	146 160 180	191 193 195 189 190 192 194 186	300 260 260	196 198 200											400 420	203 205 207 187 202 204 206 418		
	17 27 37 57 57	223C 225C 227C 235C	109 111 113	107 115 119 105 137	2290 2190 2210 2310 2330	121 123 125 127																
Al 2219-T6 Transverse Type 2	6P 1R 2N 3N 4N 5R 68	2246 2260 2280 2300 2360	108 110 112	133 113 116 126 130 132 134	2326 2206 2226 2346	121 123 125 127 131 135 120 122 124 106 116																
Al 2219-76 Radial Type 2	1P 2P 3P 1N 2N 3N	210 230 210 210	433 435 437 426 436 436	215 217 218 216	70 90 110 80 100	421 423 425 426 424 438																
Beryllium Type a	1P 2P 3P 5P 5P 5P 7P 1N 2N 3N 5N 5N 6H	4050 4070 4090 4110 4130 4150	505 509 503 507	519 521 523 525 527 529	3970 3990 4010 4030	511 513 515 517																
	1000000	\$040 \$060 \$100 \$100 \$140 \$160	504 506 508 510	526 522 524 526 528 530	3966 3980 4000 4000	512 514 516 518																
T1 A-110- AT-E11 Type 2	1P 2P 1P 1B 2B 3B	2470 2490 2510 2480 2500 2500	189 185 187 188 186 188	155 157 159 156 158 140	2410 2430 2450 2450 2460 2460	141 143 151 142 150 152				,								**				

^{*}P - plain specimens; N - notched specimens

Table 3-10

Tensile Test Results:
Inconel 718 Strain-Rate Study

Property	Ref.		Test	Conditio	on			
Measured	No.*	F1	F _{li}	G1.	Gli	H1	H _{li}	H21**
UTS (ksi)	1P 2P 3P	276.1 277.9	273.0 268.9 269.7	236.3 233.1 238.0	232.3 233.9 229.8	219.2 219.5	211.9 212.7 208.6	
Average		277.0	270.5	235.8	232.0	219.4	211.1	
TYS (ksi)	1P 2P 3P	218.9 222.5	216.0 214.3 214.3	198.9 198.9 200.5	194.8 194.8 189.5	189.9 189.7	182.6 183.4 183.4	
Average		220.7	214.9	199.4	193.0	189.8	183.1	
Percent Elongation	1P 2P 3P	10.9 14.2	13.9 12.5 16.3	13.8 10.9 11.6	13.7 13.4 13.2	13.3 13.5	14.3 11.2 7.26	
Average		12.6	14.2	12.1	13.4	13.4	10.9	
Percent Reduction in Area	1P 2P 3P	14.9 29.5	29.4 25.1 33.8	29.4 32.8 31.5	29.4 32.8 36.0	26.1 22.1	30.8 28.8 18.3	
Average		22.2	29.4	31.2	32.7	24.4	26.0	

^{*} See table below.

Table 3-11
Identification of Specimens Used in Strain-Rate Study

Reference				Test Cond	ition		
Number	F1	F _{li}	G1	Gli	H1	H _{li}	H ₂₁
1P 2P 3P	448 451	444 445 446	455 456 457	453 454 458	449 450	441 442 443	459 460

^{*}Raw data on these specimen sent to WANL for analysis

^{**} Raw data on these specimen sent to WANL for analysis.

3.1.2 Statistical Analysis of Data (by J. B. Wattier)

3.1.2.1 Methods

Analysis of variance, combined with "t" and "F" tests, has been used to evaluate, on a probability basis, the observed effects of radiation, "annealing temperatures," and test temperatures on the measurements of ultimate tensile strength (UTS), tensile yield strength (TYS), notched tensile strength (NTS), percent elongation, and percent reduction in area for various metals.

As in other situations in which mathematics is used as a tool, assumptions are required. The statistical significance tests used in the analysis of the tensile-specimen data are valid only within the framework of the assumed structure of the variation present in the observations. The assumption of random and normally distributed errors has been made. Any unknown biases introduced into the experiment would invalidate the conclusions, because the standard methods of statistical analysis give no warning of the presence of bias. These techniques assume, in fact, that no bias is present.

In a few instances, when seemingly extreme observations were noted, the question arose whether he observation should be considered discrepant and, therefore, be rejected. One discrepant value in a group might give pertinent information with regard to difficulties in the testing method, but it could also bias the analytical results if included in the calculations. Therefore, an observation that was found to lie a very long way from its

fellows in a series of replicate observations was subjected to a ratio test for extreme values and rejected if the ratio exceeded the tabulated critical value. In some instances the analysis was performed with and without the rejected "outlier."

The analysis methods used as the basis for making inferences about the observed effects determine only the statistical significance of the observed variations or differences in the data; an effect may be statistically significant and yet so small as to be of no engineering importance. In making the statistical tests of significance, probability levels of $\alpha = 0.10$, 0.05, and 0.01 were used $\left[\alpha \text{ is type I error}, \text{ or what is perhaps more commonly known as a } (1-\alpha)\% \text{ test}\right]$. When an observed difference in the averages being compared is determined to be not-significant (probability <0.90) it does not necessarily mean that there is no effect; it might be that the experiment was not sensitive enough to detect an effect when in fact such an effect does exist.

3.1.2.2 Statistical Results

Tables 3-12 through 3-23 summarize the results of the statistical analysis. The main body of the tables (arranged in 2 x 2 and 2 x 3 arrays) contains the average values, standard deviations (by range method), and the number, n, of specimens tested for the conditions in each category. The observed differences and the statistical significance of the observed differences between the averages being compared are listed in the margins of the arrays. An observed difference between the row averages within a given column is a measure of the radiation effects after the "annealing temperature" treatment specified and at the test temperature

specified. An observed difference between the column averages within a given row, for the left-hand arrays, is a measure of the "annealing temperature" effects at the -320°F test temperature and the radiation conditions specified. In the right-hand arrays, the observed difference between the column averages within a given row is a measure of the test temperature effects after an 80°F "annealing temperature" treatment and the radiation conditions specified.

3.1.2.3 Discussion of Results

The interpretation of these data is based on the significance level of the observed differences between the averages being analyzed as compared with the sampling error and the number of specimens tested. All data are not listed in the tables because they do not fit into the pattern of the 2 x 2 and 2 x 3 arrays; however, pertinent statistical aspects of these data are discussed later in this section.

The materials were subjected to various temperature conditions before testing to determine the extent that radiation-induced changes could be annealed out as a result of these "annealing temperature" treatments. Analysis of test results indicate that these "annealing temperature" treatments resulted in both an annealing-out of radiation-induced changes and, for some materials, a permanent change in the measured properties - even in the absence of radiation (control specimens). In addition, test temperature effects are quite evident, apart from any statistical analysis, so that these effects are not discussed except when

Statistical Analysis of Test Date: Income! /18, Type 5

Significance Probability

0.90 6 8 < 0.90 0.95 6 0 < 0.95 0.95 6 0 < 0.99 0.99 6 d Nets Summery

Test Condition Avg of n Velues (e/n)

			,						
	Test Temper	reture, -32	001	Annee 1	ing Tempere	ture, 60°F			
Annee1	ing Tempera	ture	Difrerence	Test To	mpereture	Difference			
-380°7	80°7	1040°P		-320 ⁰ P	80°F	nik			
		Ult	imate Teneile Str	ength (kei)					
.6.		r _C	te - t	C	•	0 - C			
"270.0"	270.0 (0.3/2)	272.2 (1.4/3)	+ 2.24	270.0	218.9 (0.9/3)	-51.1 ^d			
Al	Çi	Eci	tei - ci	Ci	01	01 - CI			
(1.4/3)	267.4 (2.5/3)	272.0 (0.9/4)	+ 5.2 ^d	207.4	213./ (2.0/4)	-53.7 ^d			
AI - 6 - 0.4ª	-2.68	€e1 - €c +0.4*	CI - AI - 2.20		01 - 0 - 5⋅2 ^d				
		To	neile Yield Streng	th (kai)					
-6.	c	tc	tc - c	C	8	8 - C			
"212.4"	212.4 (1.2/2)	212.2 (0.0/3)	- 0.2ª	212.4	184.9 (1.0/3)	-27.5 ^d			
Al	CI	fei	t _{ci} - ci	Ci	81	01 - CI			
(3.6/3)	(1.9/j)	213.5 (1.0/4)	- 4.1°	217.6	188.2 (1.9/5)	-29.4d			
AI - 6 +19.7 ^d	ei - e +5.2 ^d	€¢1 - €¢ +1.3°	CI - AI -14,5 ^d		# 3.3°				
		Note	shed Tensile Stren	ngth (kai)					
°¢•	C	E _C	Ec - C	C	•	8 - C			
"242.6"	242.6 (1.6/2)	250.9 (7.0/3)	- 8.3°	242.6	219.2 (1.5/3)	-23.4d			
Al	¢i .	Eci	Eci - Ci	Çi	0 1	Bi - Ci			
(1.8/3)	252.4 (4.3/5)	242.1 (5.0/3)	-10.3°	252.4	218.7 (1.4/3)	-33.7 ^d			
41 - ¢	€1 - € +9.8°	€c1 - €c +8.8°	61 - A1 - 7.5 ^b		01 - 0 - 0.5°				
			Persent Elongs	ion					
•6•	C	r _c	Ec - C	C		8 - C			
"12.6"	12.6 (0.4/2)	14.1 (1.2/3)	+ 1.5	12.6	10.8 (0.9/3)	- 1.6ª			
Al	Ci	€c1	tei - ei	Ci	Bi	DI - CI			
12.6 (0.9/3)	13.0 (2.7/3)	13.0 (1.3/4)	O	13.0	11.8 (0.5/4)	- 1.2			
A1 - C	CI - C +0.48	€c1 - €c -0.9*	CI - AI - 0.4ª		01 - 0 + 1.0*	e			
	·	7	ercent Reduction	In Area					
.c.	Č	£¢.	Ec - C	C	•	8 - C			
"17.1"	17.1	21.3	+ 4.2	17.1	24.4 (3.5/3)	7.3°			
Al	CI	Eci	Eci - Ci	CI	01	Bi - Ci			
21.5 (1.4/4)	14.2 (3.8/3)	17.5 (3.5/4)	+ 3.4m	14.2	23.0 (4.6/4)	9.4d			
A1 - C	CI - C -2.9 ⁸	Eci - Ec -3./*	Ci - Ai - 7.3°		81 - 8 - υ.8 ^a				
		L		<u> </u>	<u>-</u> -				

Table 5-15

Statistical Analysis of Test Data: Inconel /18, Type 1

Significance Probability

0.90 6 6 0.95 0.95 6 6 0.99 0.99 6 6 Date Summery

Test Condition Avg of n Values (e/n)

			-0-	Annealing Temperature, 80°F					
······	Test Tempe	reture, -32	1			ture, 50°P			
Anneal:	ing Temper	ture	Difference	Test Te	mpereture	Difference			
-320°F	80°F	1040 ⁰ P		-320°F	80°F				
		Ult	imate Tensile Str	ngth (kal)					
-c-	E	€¢.	fc - c	¢	•	0 - C			
"267.9"	267.9 (11.2/3)	271.9 (4.3/3)	+ 4,00	267.9	217.8 (3.7/3)	-50.1 ^d			
Al	CI	Eci	Eci - Ci	CI	81	Bi - Ci			
268.3 (4.3/3)	273.6 (3.1/4)	(3.2/3)	- 0.4	273.0	214.4 (2.9/5)	-59.6d			
AI - C - 0.4 ⁸	CI - C + 5.78	fc1 - fc +1.3ª	CI - AI + 5.3 ^d		Bi - B -4,40				
	L	Te	neile Yield Streng	th (ksi)	L	L			
.c.	C	Ec	Ec - C	C	•	9 - C			
"206.7"	206.7 (4.4/3)	207.0 (5.8/3)	+ 0.3	206.7	179.5 (3.3/3)	-27.2 ^d			
Al	CI	Eci	Ect - Cl	CI	DI	Di - Ci			
235.2 (1.4/3)	(3.5/4)	210.2 (1.7/3)	-12.3°	222.5	188.5 (3.1/5)	+34.0 ^d			
Ai - C +28.5 ^d	ci - c +15.8 ^d	€c1 - €c +3.2"	CI - AI -12.7 ^d		8i - 8 +9.0 ^d				
, , , , , , , , , , , , , , , , , , , 		Not	ched Tensile Stree	igth (ksi)					
•c•	С	Ec	Ec - C	С		8 - C			
"312.6"	312.6 (15.2/3)	316.5 (10.5/3)	+ 3.9ª	312.6	280.9 (4.6/4)	-31.7 ^d			
Al	Ci	ECI	Eci - CI	CI	Bi	01 - Ci			
342 (22.4/2)	338.2 (11.6/4)	315.8 (11.8/3)	-22.4 ^c	338.2	290.5 (7.2/3)	-47.7 ^d			
AI - C	CI - C	Eci - Ec	CI - AI		81 - 8				
+29.4°	+25.6°	-0.7	- 3.8ª		+9.6				
			Percent Elongs	ion					
•c•	C	Ec.	Ec - C	C	D	8 - C			
"12.5"	12.5 (5.7/3)	14.4 (0.7/3)	+ 1.9ª	12.5	10.9 (0.6/3)	- 1.6ª			
Al	CI	ECI	Ect - C1	CI	81	Bi - Ci			
8.4 (4.7/3)	12.7 (3.3/4)	13.7 (2.0/3)	+ 1.0 ⁸	12.7	11.4 (1.7/5)	- 1.3ª			
Ai - C - 4.1ª	ci - c	Ec: - Ec -0.7ª	CI - AI + 4.3ª		#0.5 ⁸				
		Po	ercent Reduction 1	n Area					
•c•	C	Ec	Ec - C	С	0	8 - C			
24.0	24.0	25.8	1.88	24.0	26.4 (3.1/3)	+ 2.4			
AI 24.7 (7.4/3)	61 26.2 (6.0/4)	E¢1 25.8 (5.8/3)	Eci - Ci - 0.48	26.2 C1	26.7 (3.0/5)	81 - Ci + 1.5 ⁸			
AI - 6 + 0.7°	+ 5.5 ₉	Eci - Ec	Ci - Ai + 1.5 ⁸		+0.3				
									

Table 5-14

Statistical Analysis of Test Date: Indone1 [18-WS, Type 1

Significance Probability

0.90 6 b < 0.95 0.95 6 c < 0.99 0.99 6 d

Data Summary
Test Condition
Avg of n Values
(ø/n)

	5.99 G								
Test	Temperature	, -320°F		ing Temperat	ure, 80°F				
Annesling To	empereture	Difference	Test Tem	perature	Difference				
-320°F	80°P		-320°F	80°F					
		Ultimate Tens	le Strength (kei)					
.e.	C		C	•	8 - C				
"206.3"	206.3		206.3	,173.3	-33.0 ^d				
	(5.8/5)			(6.5/5)					
ĀĪ	Ci	CI - AI	Ci	01	81 - 61				
211.6 (8.6/4)	(5.0/4)	+2.6	214.2	168.9	-45.30				
(0.0/4)	(5.0/4)			(2.0/4)					
AI - C	CI - Ç			81 - 8					
+5.3°	+7.9 ^b			-4.4					
	L	1	Strength (ke	 					
6	E	tenette tiete	C C	., •	1 - 6				
"195.7"	195.7		1	162.6	-33.1 ^d				
*****	(3.5/5)		195.7	(7.0/5)	-55.1				
	ei ei	CI - AI	ci ci	DI DI	Bi - Ci				
211.6	205.8	-5.8ª	205.8	167.8	-38.0d				
(8.6/4)	(4.0/4)	7.5	1	(13.3/4)	-,0.0				
AI - E	61 - 6		 	01 - 0					
+15.9°	+10.1b			+5.2					
		1	<u> </u>						
		Notched Tensil	e Strength (k	o1)					
.c.	E		c	T • T	8 - C				
"162.0"	162.0		162.0	148.2	-13.8 ^b				
	(7.3/4)			(3.3/4)					
Al	CI	CI - AI	CI	••	Bi - Ci				
165.8 (10.4/3)	165.0 (21.2/3)	-0.8°	165.0	149.6 (11.6/3)	-15.4 ^b				
A1 - 8	č1 - č		Ì	01 - 0					
+3.8	3.0]	+0.0					
		Percent	Elongation						
·6·	· ·	····		1	0 - C				
"0.96"	0.96		0.90	0.80	- 0.16°				
	(0.21/5)			(0.13/5)					
Al	CI	CI - AI	Ci Ci	61	Bi - Ci				
v.89	0.96	+0.07	0.96	1.15	+ 0.19 ⁸				
(0.12/4)	(0.26/4)		1	(0.13/4)					
Al - C	CI - C		<u> </u>	81 - 8					
-0.07	0			+0.35°					
		Percent he tuo	tion in Assa	L					
6		tatoan, ve (no	C C	1 1	8 - C				
"10.0"	10.0		10.0	7.9	- 2.1				
20.0			1	(0.9/5)	- 6.1				
<u> </u>	C1	CI - AI	- ci	I	Bi - Ci				
A1 5.2	5.5	0.3	5.5	9.5	+ 4.0d				
(1.6/4)	~	,	,,,	(3.1/4)	7 410				
Al - C	C1 - C			81 - 0					
_4.8d	-4.5d			+1.00					
			L	L					

Tet(1e (-1))

Statistical Analysis of Test Data: Incomel X-750, Type 1

Significance Probability

8 < 0.90 0.90 4 b < 0.95 0.95 4 c < 0.99 0.99 4 d Data Summary

Test Condition
Avg of n Values
(\sigma/n)

				Annealing Temperature, 80°F					
	Test Tempe	reture, -32	00	Annes 1	ing Tempere	ture, 80°F			
	ing Temper	ture	Difference	Test Te	mperature	Difference			
-320°F	80°₽	1040°F	ļ	-320°P	80°F	L			
		Ult	imate Tensile Str		•				
" c" "210.5"	215 (2.0/4)	210.0 (4.8/5)	€c - C - 0.5 ⁸	210.5	169.1 (3.3/3)	-41.4d			
Al	Ci	Eci	Eci - Ci	Ci	81	Bi - Ci			
209.2 (4.5/3)	213.8 (5.2/4)	208.3 (4.6/4)	- 5.5 ⁸	213.8	166.8 (3.2/3)	-47.0 ^d			
Ai - C	CI - C	Eci - Ec	CI - AI		B1 - B				
- 1.3ª	+ 3.3ª	-1./8	+ 4.6ª	li	- 2.3ª				
		To	neile Yield Stren	gth (ksi)					
c	C	ξc	€c - C	C	•	- c			
"119.7"	(1.4/4)	120.9 (3.3/5)	+ 1.28	119.7	105.4 (2.0/3)	-14.3 ^d			
Ai	CI	Eci	Eci - Ci	CI	01	81 - C1			
166.5 (3.6/3)	151.6 (2.4/4)	120 (2.9/4)	-31.6 ^d	151.6	131.0 (0.5/3)	-20.6 ^d			
AI - C	ci - c	ECI - EC	Ci - Ai		Bi - B				
+46.8d	+31.9 ^d	-0.9 ⁸	-1 ¹ 1.9 ^d		+25.6 ^d				
		Not	ched Tensile Stre	ngth (ksi)					
"C"	С	EC	Ec - C	C	•	8 - C			
"244.3"	244.3 (4.9/3)	242.3 (6.6/3)	- 1 ⁸	244.3	217.6 (2.3/3)	-26.7 ^d			
ΑĬ	CI	ECI	Eci - Ci	CI	81	BI - CI			
299.3 (4.1/3)	280.3 (5.6/4)	240.0 (4.0/3)	-40.3 ^d	280,3	241.8 (5.4/4)	-38.5 ^d			
Al - C	Ci - C	Eci - Ec	Ci - Ai		B1 - B	'			
+55.0 ^d	+36.0d	-2.3	-19.0 ^d		+24.2 ^d	į.			
			Percent Elongat	ton					
,c.	C	ŧc	EC - C	C	•	9 - C			
"24.9"	24.9 (1.0/4)	22.0 (2.3/5)	- 2.9b	24.9	16.3 (1.3/3)	- 8.6 ^d			
AI 19.1 (2.5/3)	21.1 (2.8/4)	21.4 (2.8/4)	+ 0.3ª	21.1	●i 16.1 (0.6/3)	5.0 ^d			
AI - C - 5.8d	ci - c - 3.8°	Eci - Ec - 0.6ª	CI - AI + 2.0 ⁸		0.28	c			
		Po	rcent Reduction	ln Area					
c	3	€c	E _C - C	C	•	0 - C			
"33•5"	33.5	33.8	+ 0.3 ⁸	33.5	33.3 (3.2/3)	- 0.2ª			
A1 34.2	CI 35.4	Ec. 36.8	€ _C - C + 1.48	CI 35.4	31.7	01 - C1 - 3.7°			
(3.0/3)	(1.5/4) Ci - C	(4.2/4) Ect - Ec	CI - AI		(3-2/3)				
+ 0.7	+ 1.9	+ 3.08	1.28		- 1.6 ⁸				

Table 3-16

Statistical Analysis of Test Date: ______ Inconel X-750, Type 3

Significance Probability

0.90 4 b < 0.90 0.95 4 e < 0.99 0.99 4 d

Dete Summery

Test Condition Avg of n Values (s/n)

				Anneeling Temperature, 80°P				
	Test Tempe:	reture, -32	001	Annes 1	ing Tempere	ture, 80°F		
	ing Temper	ture	Difference	Test To	mperature	Difference		
-320°P	80°7	1040°F		-320°F	80°p			
		U1 %	imate Tensile Str	ength (ksi)				
.6.		€¢.	Ec - C	C	Ü	9 - 6		
"210.1"	210.1 (3.4/4)	207.8 (1.2/3)	- 2.3ª	210.1	166.4 (1.1/3)	-43.7 ^d		
ĀI	CI	tei	Ect - Ci	CI	81	81 - CI		
213.0 (3.2/3)	213.2 (0.5/3	208.1 (1.1/3)	- 5.1 ^d	213.2	164.9 (1.4/5)	-48.3 ^d		
AI - C + 2.9 ^b	CI - C + 3.1 ^b	€¢1 - €¢ +0.3*	61 - A1 + 0.2ª		81 - 8 - 1.5			
		To	neile Yield Stren	th (ksi)		<u> </u>		
-64	6	ŧc	€c - €	C	•	8 - C		
"125.1"	125.1 (2.3/4)	123.6 (0.9/3)	- 1.5	125.1	107.7 (1.8/3)	-17.4 ^d		
Al	CI	Eci	Eci - Ci	CI	91	Bi - Ci		
170.1 (2.5/3)	153.2 (0.9/3)	123.7 (1.0/3)	-29.5 ^d	153.2	129.4 (1.3/5)	-23.8 ^d		
AI - C +45.0 ^d	61 - 6 +28.1 ^d	€cı - €c +0.1°	CI - AI -16.9 ^d		91 - 8 +21.7 ^d			
		Not	ched Tensile Stree	igth (ks1)				
6	C	Ec	Ec - C	C	ı	8 - C		
"182.7"	182.7 (4.5/4)	173.4 (6.2/4)	- 9.3°	182.7	155.4 (7.7/4)	-27.3 ^d		
Al	Çi	Eci	Ect - Ci	CI	81	SI - CI		
(0.8/3)	205.1 (0.5/3)	178.6 (3.0/3)	-26.5 ^d	205.1	171.3 (4.6/4)	-33.8 ^d		
AI - C	CI - C	Eci - Ec	Ci - Ai		81 - 8			
+35.1 ^d	+22.4 ^d	+5.2ª	-12.7 ^d		+15.9 ^d			
			Percent Elongat	ion				
.6.	C	ξç	Ec - C	C	ı	8 - C		
"27.0"	27.0 (0.9/4)	26.7 (1.2/3)	- 0.3	27.0	18.8 (0.1/3)	- 8.2 ^d		
Al	CI	ECI	Eci - CI	CI	1	Bi - Ci		
21.9 (0.8/3)	23.9 (0.6/3)	25.7 (0.1/3)	+ 1.8°	23.9	18.1 (0.9/5)	- 5.8 ^d		
Al - C	CI - C	ECI - EC	CI - AI		81 - 8	c		
- 5.1 ^d	- 3.1 ^d	-1.0	+ 2.0d		- 0.7			
			ercent Reduction					
C		Ec	Ec - C	C	35.0	8 - C		
"35.1"	35.1 (2.5/4)	34.3 (3.8/3)	- 0.8ª	35.1	35.8 (2.8/3)	0.7		
Ai	CI	Eci	Eci - Ci	C1	Bi	BI - CI		
33.2 (1.5/3)	33.0 (1.5/3)	33.9 (1.5/3)	+ 0.9ª	33.0	36.0 (3.9/5)	3.0		
Al - C	CI - C	Eci - Ec	CI - AI		81 - 8			
- 1.9ª	- 2.1	-0.4	- 0.2		0.5			

Statistical Analysis of Test Data: AISI 301-CW, Type 3

Significance Probability

0.90 6 b 0.95 0.95 6 c 0.99 0.99 6 4

Deta Summary

Test Condition
Avg of n Values
(\(\sigmu/n\))

	4,33		Annealing Temperature, 80°F						
	Test Tempe	reture, -32	004	Annea 1	ing Tempers	ture, 80 ⁰ F			
Annea 1	ing Temper	ture	Difference	Test Te	mpereture	Difference			
-320°F	80°F	540°F		-320 ⁰ ₽	80°F				
		Ult	imate Tensile Str	ength (ksi)					
"2 92.8"	292.8 (2.4/4)	291.4 (0.2/3)	Ε _Α - C - 1.4°	292. 8	186.9 (3.5/4)	● - c - 5.9°			
Al 290.4 (2.6/3)	294.3 (2.5/4)	EAI 293.7 (5.2/4)	EA) - Ci - 1.6 ⁸	CI 294.3	188.8 (2.3/4)	81 - C1 - 5∗5°			
AI - C - 2.48	Ci - C +1.5 ⁸	+2.38	CI - AI 3.9*		81 - 8 +1.9 ⁴	i			
	 	Te	nsile Yield Stren	gth (kal)	L				
-c-	C	EA	EA - C	C	ı	9 - C			
"165.2"	165.2 (1.6/4)	190.6 (1.5/3)	+25.4 ^d	165.2	151.0 (2.2/4)	-14.2 ^c			
Ai	CI	EAI	EAI - CI	CI	84	Di - Ci			
176.1 (2.6/3)	173.7	193.3 (5.0/4)	+19.6 ^d	173.7	156.3 (0.5/4)	-17.4°			
Al - C	ci - c	EAI - EA	CI - AI		01 - 0				
+10.9 ^d	+8.5 ^d	+2.7 ^b	- 2.48		+5•3 ^d				
			ched Tensile Stren	igth (kai)					
"207.3"	207.3 (3.3/4)	227.4 (1.1/4)	+20.1 ^d	207.3	191.2 (0.7/4)	-15.3 ^d			
Al 214.3	CI 216.7 (1.5/4)	EAI 232 (1.7/4)	EAI - CI +15.3 ^d	Ci 216.7	193.8	Bi - Ci -22.9 ^d			
(2.5/3) Ai - C 7.0d	ci - c 9.4d	EAI - EA 4.6d	CI - AI 2.48		(1.1/4) 81 - 8 2.6 ^b				
7.0	9.4	4.0			2.3				
			Percent Elonget	ion					
-C-	С	EA	EA - C	C	•	B - C			
"20.4"	20.4 (1.1/4)	(0.8/3)	+ 2.5 ^d	20.4	(0.8/4)	- 1.2 ^b			
Ai	CI	EAI	EAI - CI	Çi	8i	Bi - Ci			
20.2 (0.3/3)	20.0 (0.5/4)	21.3 (0.9/4)	+ 1.3 ^b	20.0	17.4 (1.3/4)	- 2.6 ^d			
Ai - C	CI - C	EAI - EA	CI - AI		81 - 8	C			
- 0.2ª	-0.4 ⁸	-1.6 ^c	- 0.2 ⁸	<u> </u>	-1.8 ^c				
		Pe	ercent Reduction i	n Ares					
c	c	EA	E _A - C	С		● - C			
36.3	36.3 (6.7/4)	(1.8/3)	- 4.1 ⁸	36.3	40.0 (5.0/4)	+ 3•7ª			
Ai	Ci	EAI	EAI - Ci	CI	Bi	B1 - C1			
32.1 (3.8/3)	32.3 (5.6/4)	46.7 (3.2/3)	+14.4 ⁸	32.3	36.2 (8.5/4)	+ 3.9ª			
Ai - C - 4.2ª	CI - C oa	EAI - EA -14.5 ⁸	0.2ª		-3.8ª				
				·····					

Table | (-12

Statistical Analysis of Test Date: AISI 101-30, Type 1

Significance Probability

0.90 6 b < 0.95 0.95 6 e < 0.99 0.99 6 d Deta Summery

Test Condition.
Avg of n Values
(ø/n)

	7.5650		-0-	Annealing Temperature, 80°P					
	rest Temper	reture, -32	0~1	Annes 1	ing Tempere	ture, 80°F			
	ing Tempere		Difference		mperature	Difference			
-320°F	80°p	540 0p		-320°F	80°7				
		Ult	imate Tensile Str	ength (kel)					
,¢,	,	EA	ta - c	C	В	0 - C			
"165.4"	105.4 (9.4/3)	169.0 (2.4/3)	+ 3.2	105.4	92./ (0.9/3)	-12.1 ^d			
Al	CI	EAI	EAI - CI	CI	81	Bi - Ci			
173.0 (4.0/3)	172.4 (4.2/5)	158.8 (4.3/2)	- 3.0ª	172.4	96.0 (1.0/5)	-76.4 ^d			
AI - C + 7.6°	CI - C + 7.0°	+0.2	CI - AI - 0,68		01 - 0 + 3.3°				
		To	neile Yield Streng	th (kei)					
,6,	C	EA	EA - C	C	•	0 - C			
"85.4"	85.4 (8.0/3)	87.0 (6.2/3)	+ 2.2	85.4	49.1 (1.2/3)	-36.3ª			
Al	CI	EAI	EAI - CI	CI	81	81 - CI			
115.0 (2.1/3)	104.h (5.4/5)	(4.6/2)	-14.5 ^d	104.4	60.8 (4.3/5)	-43.6 ^d			
AI - C	Ci - C	EAI - EA	CI - AI	1	81 - 8				
+29.6 ^d	+19.0 ^d	+2.1	-10.0 ^d	.	+11.7 ^d				
		Note	ched Tensile Street	ngth (ksi)					
* C*	C	EA	EA - C	C	•	8 - C			
"191.0"	191.0	184.2 (8.3/4)	- 6.8ª	191.0	121.6	-69.4 ^d			
Al	Ci	EAI	EAI - CI	CI	91	BI - CI			
228.2 (11.2/3)	208.5 (1.4/4)	191.9 (6.9/5)	-16.6 ^d	208.5	131.4 (2.4/5)	-77.1 ^d			
Al - C	CI - C	EAI - EA	Ci - Aı		B1 - 0				
+37.2 ^d	+17.5 ^d	+ 7.7 ^b	-19.7 ^d		+ 9.8°				
			Percent Elonga	ion		•			
.c.	C	EA	EA - C	C	•	0 - C			
"43.1"	43.1 (9.0/3)	48.1 (5.3/3)	+ 5.0	43.1 (9.0/3)	31.2 (0.5/3)	-11.9°			
Al	CI	EAI	EAI - CI	CI	91	Bi - Ci			
34.3 (4.7/3)	38.7 (4.3/5)	45.2 (8.0/2)	+ 6,5	38.7 (4.3/5)	26.0 (2.6/5)	-12.7 ^d			
Al - C	Ci - C	EAI - EA	CI - AI		81 - 8	e			
- 8.8°	- 4.4	- 2.9ª	+ 4.48		- 5.2				
			ercent Reduction						
.c.	Ç.	EA	EA - C	C	•	0 - C			
48.2	48.2 (3.2/3)	51.0 (2.2/3)	+ 2.8ª	48.2	44.9 (5.6/3)	- 3·3°			
Al	CI	EAI	EAI - CI	CI	01	BI - CI			
44.3 (2.1/3)	44.5 (4.0/5)	45.1 (0.4/2)	+ 0.08	44.5	52.2	+ 7.7ª			
Ai - C	CI - C	EAI - EA	CI - AI		81 - 9				
- 3.9ª	- 3.78	- 5.9ª	+ 0.2 ⁸		+ 7.3				
		Ll			l				

Statistical Analysis of Test Data: _

A Clark Tree

Significance Probability

0.90 6 b 6 0.95 0.95 6 a 6 0.99 0.99 6 d Deta Summary

Test Condition Avg of n Values (a/n)

				<u> </u>				
Test Temperature, -320 ⁰ F				Annealing Temperature, 80°F				
Annealing Temperature			Difference	Test Te	mperature	Difference		
-320°F	80°p	540°P		-320°P 80°P				
		Ult	imate Tensile Str	ength (kel)				
.c.	6	EA	EA - C	C		8 - C		
"/5.1"	(3.1 (3.5/2)	68.1 (1.0/3)	- 9.0 ^d	/1.1	59.7 (1.9/3)	-13.49		
Al	Ei	EAI	EAI - CI	CI	81	81 - CI		
/5.1 (0.5/3)	(0.11/1)	(1.2/4)	- /.1 ^d	14.2	58.4 (1.4/4)	-15.8 ^d		
4 - C + 2.0ª	Ci - C +1.18	EAI - EA -1.0"	CI - AI - 0,9 ^a		01 - 0 -1.5			
		To	neile Yield Stren	gth (ks1)	-			
.c.	C	EA	EA - C	e	•	8 - C		
"49.2"	49.2 (4.1/2)	40.7 (0.4/3)	- 8.5 ^d	44.2	112.0 (1.1/3)	- 1.2ª		
Ai	CI	EAI	EAI - CI	Ç1	81	BI - CI		
63.1 (0.6/3)	51.8 (0.5/3)	39.4 (1.7/4)	-12.4 ^d	51.8	41.5 (1.0/4)	-10.3ª		
Al - C +13.9 ^d	Ci - C +2.0	EAI - EA -1.3 ⁰	Ci - Ai -11. i ^d		01 - 0 -0.5ª			
•		Not	ched Tensile Stre	ngth (kei)				
•e•	C	EA	EA - C	C	•	8 - C		
"07.5"	0.6/2)	58.5 (1.7/3)	- 8.9 ^d	97.5	50.5 (1.3/3)	-11.0°		
Ai	CI	EAI	EAI - CI	CI	01	Di - Ci		
75.4 (0.8/3)	67.9 (1.7/4)	57.4 (2.9/4)	•10.5 ^d	. 21.9	53.5 (2.4/4)	-14.4d		
Al - C	CI - C	EAT - EA	CI - At		01 - 0			
+ 7.9 ^d	+0.48	-1.2	- 1.5ª		-3.0°			
			Percent Elongs	ion				
·c-	C	EA	EA - C	C	•	8 - C		
"9.1"	9.1 (0.5/2)	13.0 (0.8/3)	+ 3.y ^d	9.1	7.0 (0.0/3)	- 2.1°		
Ai	CI	EAI	EAI - CI	Ci	91	BI - CI		
8.0 (0.9/1)	10.1 (0.6/3)	11.3 (1.6/4)	+ 1.2	10.1	0.5/4)	- 3.3 ^d		
Ai - C - 1.10	Ci - C +1.0ª	EAI - EA -1.7b	CI - AI + 2.1°		-0.2ª	6		
	Percent Reduction in Area							
c	•	E _A	€A - C	c	• 1	D - C		
14.8	14.8 (0.6/2)	20.2 (2.1/3)	+ 5.44	14.8	11.3 (1.5/3)	- 3.5		
Ai	Ci	EAI	EAI - CI	Ci	91	Bi - Ci		
23.5 (0.1/3)	19.1 (2.4/3)	22.9 (3.4/4)	+ 3.8 ^a	19.1	21.3 (5.1/4)	+ 2.2		
Ai - C + 8.7 ^b	Ci - C +4.3°	EAI - EA +2.78	Ci - Ai - 14.11 ⁸		+10.0 ^d			
				<u> </u>				

Table 3-20

Statistical Analysis of Test Date: Al 2219-To-Transverse, Type 2

Significance Probability

0.90 4 b < 0.95 0.95 4 c < 0.95 0.95 4 c < 0.99 Data Summary
Test Condition
Avg of n Values
(e/n)

0.99 6 4								
	Teape re ture	, -320 ⁰ F	Annealing Temperature, 80°F					
Anneeling Temperature		Difference	Test Tem		Difference			
-320°F	80°7		-320°F	80°F				
Ultimate Tensile Strength (ksi)								
*** "72.3"	(1.4/4)		6 72.3	62.1 (0.8/5)	-10.2 ^d			
80.7 (0.8/3)	(1.7/6)	CI - AI - 8,6 ^d	ti 72.1	61.4 (0.5/6)	81 - 61 -10.7 ^d			
A1 - 6 + 8.4 ^d	-0.2ª			01 - 0 -0.7ª				
		Tensile Yield	Strength (ke	1)				
6 *53.6*	53.6 (0.5/4)		53.0	47.3 (0.9/5)	- 6.3 ^d			
79.0 (0.9/3)	54.6 (1.1/6)	-24.4 ^d	61 54.6	(1.5/6)	81 - 61 - 7.4 ^d			
AI - 6 +25,4 ^d	€1 - € +1.0ª			-0.1°				
		Notshed Tensi	le Strength (k					
"88.2"	88.2 (5.2/5)		88.2	75.5 (2.4/4)	-12.1 ^d			
Al 91,4 (6.8/5)	68.1 (3.8/6)	CI - AI - 3.3 ^d	68.1	72.2 (4.1/5)	8i - 6i -15.9 ^d			
AI - 6 + 3.2°	€1 - € - 0.1°			-3.2ª				
		Percent	Elongation					
"6" "7.6"	7.6 (0.6/4)		7.6	8.27 (1.3/5)	0 - C + 0.7 ⁸			
2.27 (0.4/3)	6.67 (1.5/6)	CI - AI + 4.4 ^d	CI 6.67	8.02 (2.5/6)	81 - 61 + 1,40			
Ai - ¢ - 5.3 ^d	CI - C -0.9ª			01 - 0 -0.025				
Percent Reduction in Area								
6. 7	6.7 (1.5/4)		6.7	9.4 (3·3/5)	0 - G + 2.7°			
AI (2.3/3)	6.4 (1.1/6)	CI - AI + 2.1ª	C 1 6.4	9,0 (2.8/6)	91 - Ci + 2.6 ^b			
Al - ¢ - 2.40	¢1 - ¢ +0.3°			81 - 8 -0,48				

Table 3-21

Statistical Analysis of Test Date: Al 2219-To-Radial, Type 2

Significance Probability

0.90 6 b < 0.95 0.95 6 c < 0.99 0.99 6 d

Date Summery
Test Condition
Avg of n Values
(ø/n)

Test Temperature		, -320°P	Annee1	ture, 80°F				
Annealing Temperature		Difference	Test Temperature		D1 [ference			
-320°F 80°F			-320°F	80°p				
Ultimate Tensile Strength (xei)								
	С		C	1	1 - 6			
"73.6"	73.6		73.6	61.8	-11.8 ^d			
1 '3."	(2.7/2)		13.0	(0.8/3)	-11.0-			
				ļ				
Al	C1	CI - AI	CI	81	01 - CI			
80.5 (0.1/3)	74.7 (1.7/2)	- 5.8 ^d	74.4	61.7	-12.7 ^d			
(0.1/3)	(1.1/2)			(0.4/3)				
Al - C	CI - C			81 - 0				
+ 6.9ª	+0.8			-0.1				
			<u> </u>		· · · · · · · · · · · · · · · · · · ·			
		Tensile Yield	Strength (ke	1)				
.c.	С		C		8 - 6			
"51.7"	51.7		51.7	46.3	- 5.40			
,	(0.9/2)		1	(1.2/3)	,,,			
ÁI	CI	CI - AI	Ci Ci	81	BI - CI			
78.3		-24.8d		46.8	- 6.7ª			
(0.9/3)	53.5 (1.3/2)	-24.0-	53.5	(3.6/3)	- 6.7			
AI - C	CI - C			81 - 0				
+26.6 ^d	+1.8			+0.5				
			327		· · · ·			
		Notched Tensil						
.c.	C		c	•	0 - C			
"93.0"	,93.0		93.0	77.3 (2.4/2)	-15.7 ^d			
	(3.5/2)			(2.4/2)				
Al	CI	CI - AI	CI	01	01 - CI			
105.4	92.3	-13.10	92.3	77.7	-14.6 ^d			
(7.6/3)	92.3 (4.6/2)		, , , ,	(0.5/3)	4,7,10			
AI - C	ci - c			81 - 8				
+12.40	-0.7			+0.40				
746.7				¥0.4				
		Percent	Elongation					
c	c		C		9 - 6			
"13.0"	13.0		13.0	10.9	- 2.1 ⁸			
	(6.6/2)			(1.4/3)				
Al	CI	CI - AI	CI	81	81 - CI			
	- 1	+ 8.24	-					
5.62 (1.2/3)	13.8 (1.1/2)	+ 0.2-	13.8	11.4 (1.2/3)	- 2.40			
				1				
Al - C	CI - C			Bi - B				
- 7.4°	+0.8			+0.5				
Pagacat Reduction to Area								
Percent Reduction in Area								
·	C		C .	•	8 - C			
11.7	11. (5.0/2)		11.7	12.7 (2.8/3)	+ 1.0°			
Al	CI	Ci - Ai	CI	81	Bi - Ci			
8.5	10.7	+ 2.21	10.7	11.0	+ 0.3			
(2.2/3)	(1.6/2)			(3.9/3)				
Al - C	CI - C			91 - 9				
- 3.2ª	-1.0°			-1.7				

Table 3-22

Statistical Analysis of Test Date: Beryllium, Type 4

Significance Probability

0.90 6 b < 0.95 0.95 6 e < 0.99 0.99 6 d

Date Summery

Test Condition Avg of n Velues (#/n)

			<u></u>				
Test Temperature, -320°F				Annealing Temperature, 80°F			
Annealing Temperature		Difference	Test Ten	4	D1 ffe rence		
-320°F	80°F		-320°F	80°7			
Ultimate Tensile Strongth (kei)							
.6.	E		C	•	8 - G		
" 36.2	(9.4/6)		36.2	51.1 (1.5/4)	+14.9 ^d		
Al	EI	CI - AI	CI	81	81 - CI		
10.8	(6.9/1)	+26.5 ^d	37+3	49.9 (1.4/4)	+12.6 ^d		
A1 - C	C1 - C			Bi - B			
-25.4 ^d	+1.14			-1.2ª			
		Tensile Yield	Strength (ke	1)			
·c• ·	C		C	1 • 1	8 - 6		
"36.2"	36.2 (9.4/6)		36.2 (9.4/6)	(0.5/4)	+ 5.2°		
Ai	ci	CI - AI	CI	1 01	81 - CI		
10.8 (1.4/4)	(6.9/7)	+26.5 ^d	37.3 (6.9/7)	42.9 (0.6/4)	+ 5.6°		
Al - C -25.4 ^d	CI - C +1.1°		+1.1*	81 - 8 +1.5 ⁸			
	L	Motched Tensi	le Strength (k	a 1)			
.c.	¢		C	1 1	0 - C		
"6.9"	(1.7/6)		6.9	(4.1/h)	+24.3 ^d		
Al	¢1	CI - AI	CI	81	91 - Cl		
(1.6/4)	8.5 (2.5/6)	+ 2.2	8.5	31.4 (1.5/4)	+22.9 ^d		
N - C	e1 - e			01 - 8			
	<u> </u>	Percent	Elongstion	1			
.c.	c		C	0.39 (0.17/4)	B - Č		
ĀI	CI	CI - AI	CI	81	81 - CI		
		Insufficient Deta		0.62			
AI - C	ci - c			81 - 8			
insufficient Deta	Insufficier Dets	•		+0.23	· · · · · · · · · · · · · · · · · · ·		
Percent Reduction in Area							
•6•	C		c	•	B - C		
Ai	Çi	Ci - Ai Insufficient Data	¢1	81	BI - CI		
41 - C	CI - C		1	01 - 0			
nsufficient Dete	Insufficien			Insufficient Deta	·		

Table 3-21

Statistical Analysis of Test Date: Ti A-110-AT-Eli, Type 2

Significance Probability

0.90 & b · 0.95 0.95 & 0 < 0.19 0.99 & d

Data Summery

Test Condition Avg of n Values (\sigma/n)

0.99 % 4								
Test Temperature, -320°F			Annealing Temperature, 80°F					
Anneeling Temperature		Difference	Test Tem	pereture	Difference			
-320°P	80°P		-320 ⁰ F	80°7				
Ultimate Tensile Strength (kei)								
.c.	C		c	•	0 - C			
"183.3"	(0.8/3)		183.3	121.1 (0.5/3)	-05.5 _q			
Al	CI	CI - AI	CI	81	BI - CI			
$\binom{197.5}{(2.1/3)}$	191.8 (2.1/3)	-5.7 ^d	191.8	129.0 (0.4/3)	-62.8 ^d			
AI - C +14.2 ^d	ci - c + 8.5 ^d			•1 - • +7.9 ^d				
		Tensile Yiel	d Strength (ke	1)	 			
.c.	C		C	• 1	B - C			
"177.6"	177.6 (0.9/3)		177.6	119.2 (0.2/3)	-58.4 ^d			
At	CI	CI - AI	CI	Bi	91 - CI			
195.0 (1.9/3)	188.8 (2.2/3)	-6.2 ^d	188.8	128.6 (0.5/3)	-60.2 ^d			
AI - C +17.4 ^d	CI - C +11.2 ^d			#9.4d				
		Notched Tensi:	le Strength (k	1				
"C"	C		E	1 • 1	0 - ¢			
"245.0"	245.0 (5.0/3)		245.0	189.3 (2.1/3)	-55.7ª			
Äl	CI	CI - AI	CI	01	Bi - Ci			
(7.4/3)	228.5 (6.1/3)	+1.4*	228.5	193.2 (3.4/3)	-35•3 ^d			
AI - C	CI - C		i – — —	81 - 8				
-17.9 ^d	-16.5 ^d			+3.9				
		Percent	Elongation					
.c.	C		С	•	1 - C			
"18.3"	18.3 (4.3/3)		18.3	16.5 (0.8/3)	- 1.8°			
Ai	CI	CI - AI	ÇI	81	81 - CI			
4.6 (0.07/3)	12.0 (4.5/3)	+7.4°	12.0	12.8 (0.4/3)	+ 0.8ª			
AI - C -13.7 ^d	cı - c - 6.3°			+3.7				
Percent Reduction in Area								
c	C		C		8 - C			
24.7	24.7 (10.3/3)		24.7	37.1 (2.4/3)	+12.4 ⁰			
27.8 (3.6/3)	29.7 (6.1/3)	CI - Ai +1.9 ⁸	¢i 29.7	37.6 (11.7/3)	0i - Ci + 7.9 ⁰			
Al - C + 3.1ª	CI - C + 5.0ª			81 - 8 +0.5 ⁸				

radiation. Without going into detail, an estimate of the "annealing temperature" x radiation interaction and of the test temperature x radiation interaction terms is, respectively,

$$1/2$$
 [(C + Ei) - (E + Ci)]

and

$$1/2$$
 [(C + Bi) - (B + Ci)]

The latter interaction term, evaluated for NTS of Inconel 718, Type 3 (Table 3-12), is

$$1/2$$
 [(242.6 + 218.7) - (219.2 + 252.4)] = -5.2 ksi

This interaction term is significant. It means that there is an interaction between the test temperature and radiation such that the effects are not additive (independent). This can be observed by noting the different response to the test temperature for the radiation condition (Bi - Ci) as compared with the no-radiation condition (B - C). In other words, the -33.7-ksi value is a significantly larger change than the -23.4-ksi value.

The tensile properties of materials tested at the same temperature after being subjected to different "annealing temperature" treatments (e.g., E_{Ai} , Ci, and E_{A} , C) at times experienced changes resulting from an interaction between permanent changes in the material due to a temperature effect and recovery due to annealing-out of radiation-induced changes. This is referred to above

as an "annealing temperature" x radiation interaction. If the difference between the control specimen averages $(E_A - C)$ is insignificant (no temperature effects), then a difference in the irradiated specimen averages $(E_{A1} - C1)$ can generally be interpreted as an annealing-out of radiation-induced changes in the property. On the other hand, if a significant change (temperature effect) is evident between the control specimen averages, then the interpretation of changes between the irradiated specimen averages is not clear. Part of the observed change could be attributed to a permanent change in the property due to a temperature effect and part could be attributed to annealing as a result of the "annealing temperature" treatments.

To supplement Tables 3-12 through 3-23 and the previous general statements, the following interpretation of the statistical results for each material type is presented.

Inconel 718, Type 3 (Ref. Table 3-12)

Ultimate Tensile Strength

The significant +5.2 and -5.2 ksi values for $(E_{C1}-Ci)$ and (Bi-B), respectively, are difficult to interpret.

Tensile Yield Strength

There are significant radiation and annealing effects, with the radiation effects annealed out at $1040^{\circ}F$.

Notched Tensile Strength

Except for a suspected "outlier" in the data at $E_{\rm C}$, the interpretation is similar to that for TYS. Analyzing the data as is, there is a significant interaction between "annealing temperature" and radiation. If the "outlier" is deleted from the analysis, the "annealing temperature" x radiation interaction is not significant.

Percent Elongation

There are no significant radiation or annealing effects.

Percent Reduction in Area

The (C1 - A1) difference is significant.

D, D_1 , D_1 Series

There are no significant radiation effects. Although some of the observed differences are as large as some of the significant results in the tables, the discrimination of the observed differences is less sensitive because the sample size, n, was smaller for this D series.

Inconel 718, Type 1 (Ref. Table 3-13)

Ultimate Tensile Strength

There are no significant radiation or annealing effects.

Tensile Yield Strength

There are significant radiation and annealing effects, with the radiation effects still evident at 80°F and annealed out at 1040°F.

Notched Tensile Strength

The interpretation is similar to that for TYS except that there is no significant annealing effect between Ci and Ai and the significance level is lower because of the larger variability in the data.

Percent Elongation

There are no significant radiation or annealing effects.

Percent Reduction in Area

There are no significant radiation or annealing effects.

D_C, D_{C1} Series

There are no significant radiation effects.

Inconel 718-WS, Type 3 (Ref. Table 3-14)

Ultimate Tensile Strength

There are significant radiation effects.

Tensile Yield Strength

There are significant radiation effects.

Notched Tensile Strength

There are no significant radiation or annealing effects.

Percent Elongation

There are no significant radiation or annealing effects.

Percent Reduction in Area

There are significant radiation effects as well as a significant test temperature x radiation interaction.

Inconel X-750, Type 1 (Ref. Table 3-15)

Ultimate Tensile Strength

There are no radiation or annealing effects.

Tensile Yield Strength

There are radiation and annealing effects, with the radiation effects annealed out at 1040°F.

Notched Tensile Strength

There are radiation and annealing effects, with the radiation effects annealed out at 1040° F. The test temperature x radiation interaction is significant.

Percent Elongation

There are significant radiation effects.

Percent Reduction in Area

There are no significant radiation effects.

D_C, D_{Ci} Series

There are no significant radiation effects.

Inconel X-750, Type 3 (Ref. Table 3-16)

Ultimate Tensile Strength

There are significant radiation and annealing effects, with the radiation effects annealed out at 1040°F.

Tensile Yield Strength

There are significant radiation and annealing effects, with the radiation effects annealed out at 1040°F.

Notched Tensile Strength

There are significant radiation and annealing effects. A significant "annealing temperature" effect of -9.3 ksl for condition E_C - C and a significant "annealing temperature" x radiation reaction of -26.5 ksl for $(E_{C1}$ - C1) are apparent.

Percent Elongation

There are significant radiation and annealing effects, with the radiation effects annealed out at 1040°F.

Percent Reduction in Area

There are no significant radiation or annealing effects.

D_C, D_{Ci} Series

There are no significant radiation effects.

AISI 301-CW, Type 3 (Ref. Table 3-17)

Ultimate Tensile Strength

There are no significant radiation or annealing effects.

Tensile Yield Strength

There are significant radiation effects. The 540°F "annealing temperature" resulted in a significant increase in the TYS of irradiated specimens, indicating a significant interaction between "annealing temperature" and radiation. The radiation effects are not completely annealed out, as evidenced by the significant radiation effects still remaining after the 540°F anneal.

Notched Tensile Strength

The interpretation is similar to that given for TYS.

Percent Elongation

There are significant radiation and "annealing temperature" effects.

Percent Reduction in Area

There are no significant radiation or annealing effects. A 70.8-ksi datum point was deleted from the analysis.

DA, DA1 Series

There are significant radiation effects for TYS (3.0b ks1) and NTS (2.8b ks1).

AISI 303-Se, Type 1 (F.f. Table 3-18)

Ultimate Tensile Strength

There are significant radiation effects and an "annealing temperature" x radiation interaction. The radiation effects are annealed out at $540^{\circ}F$.

Tensile Yield Strength

There are significant radiation and annealing effects, with the radiation effects annealed out at 540°F.

Notched Tensile Strength

There are significant radiation and annealing effects, with an "annealing temperature" x radiation interaction indicated. The radiation effects are still evident after the 540°F anneal.

Percent Elongation

There are significant radiation effects for the (Ai - C) condition. Generally, annealing effects are not significant; however, radiation effects decrease as the annealing temperature increases.

Percent Reduction in Area

The apparent grouping of the data at Bi suggests that some sort of bias may have been introduced into the data. No conclusions are made because the grouping is difficult to interpret.

DA, DAi Series

There are no significant radiation effects. The 66.8-ksi value was deleted from the analysis.

Aluminum 2219-T6, Type 3 (Ref. Table 3-19)

Ultimate Tensile Strength

There are significant "annealing temperature" effects.

Tensile 'ield Strength

There are significant radiation and annealing effects with the radiation effects annealed out at 80°F. "Annealing temperature" effects occur between 80°F and 540°F.

Notched Tensile Strength

Same interpretation as TYS.

Percent Elongation

The significant effects observed are such that interpretation is difficult.

Percent Reduction in Area

There are significant radiation effects.

D_A, D_{Ai} Series

There are significant radiation effects at the 540°F (13.7d) test temperature for percent reduction in area and a significant test temperature x radiation interaction.

Aluminum 2219-T6-Transverse, Type 2 (Ref. Table 3-20)

Ultimate Tensile Strength

There are significant radiation and annealing effects, with the radiation effects annealed out at 80°F.

Tensile Yield Strength

There are significant radiation and annealing effects, with the radiation effects annealed out at 80°F.

Notched Tensile Strength

There are no significant radiation or annealing effects.

Percent Elongation

There are no significant radiation or annealing effects.

Percent Reduction in Area

There are no significant radiation or annealing effects.

Aluminum 2219-T6-Radial, Type 2 (Ref. Table 3-21)

Ultimate Tensile Strength

There are significant radiation and annealing effects, with the radiation effects annealed out at 80°F.

Tensile Yield Strength

There are significant radiation and annealing effects, with the radiation effects annealed out at 80°F.

Notched Tensile Strength

There are significant radiation and annealing effects, with the radiation effects annealed out at 80°F.

Percent Elongation

There are significant radiation and annealing effects, with the radiation effects annealed out at 800F.

Percent Reduction in Area

There are no significant radiation or annealing effects.

Beryllium, Type 4 (Ref. Table 3-22)

Ultimate Tensile Strength

There are significant radiation and annealing effects, with the radiation effects annealed out at 80°F. Test temperature effects are significant.

Tensile Yield Strength

Interpretation similar to UTS except the observed test temperature effects are not significant.

Notched Tensile Strength

There are no significant radiation or annealing effects.

Percent Elongation

Limited data (not significant).

Percent Reduction in Area

Insufficient data for any comparisons.

Titanium A-110-AT-Eli, Type 2 (Ref. Table 3-23)

Ultimate Tensile Strength

There are significant radiation and annealing effects. The radiation effect is partially annealed out at 80°F .

Tensile Yield Strength

There are significant radiation and annealing effects. The radiation effects are partially annealed cut at 80°F .

Notched lensile Strength

There are significant radiation effects. No annealing is evident at 80°F.

Percent Elongation

There are significant radiation and annealing effects. The radiation effects are partially annealed out at 80°F.

Percent Reduction in Area

There are no significant radiation or annealing effects.

3.1.3 Discussion and Analysis of Results

Tables 3-24 through 3-29 summarize the test results by giving the percent change, from control to irradiated, in the six types of physical-property measurements: UTS, TYS, NTS, NTS/UTS, percent elongation, and percent reduction in area. In each table, the percent change in the property, as determined from the average control and irradiation values, is presented for each material and test condition. The statistical significance probability of the data was taken from the statistical analysis of Section 3.1.2.

The amount of change that should be considered significant or insignificant is arbitrary; however, for the purpose of this discussion, which is to indicate the magnitude of change in property values experienced by materials as a result of this irradiation, the following assignments of significance are used. Percent changes of less than 5% will be considered insignificant, even though the observed differences between the averages of the control and irradiated values are statistically significant. Percent changes of greater than 5% but less than 10% that are statistically significant will be considered as of slight significance. Percent changes of greater than 10% that are statistically significant will be considered as significant changes.

No effort has been made to analyze the results of this test for the purpose of determining the mechanisms of material damage associated with the apparent changes due to the environmental conditions imposed. The analysis of results appearing in

Table 3-2 $^{\rm A}$ Percent Change: Ultimate Tensile Strength

			ļ			Test Cond	Condition (Control	1.	Irradiated)				
Material	C - A1	C - C1	B - Bt	DA - DA1	DA - DAI	D _B - D _{B1}	DB - DBi	13 ₀ - 3 ₀	Dc - Dc1	10 ₀ - 0 ₀	Po - Poi	EA - EA1	EC - EC1
Inconel 718 Type 3	- 0.15ª	- 0.90	- 2.384	+ 0.91	+ 1.0	+ 3.14	+ 1.38	- 0.42	- 1.05	- 3.10	- 1.94	•	• 0.15
Inconel 718 Type 1	+ 0.15	+ 2.12ª	- 1.50	•	•	•	•	* 0.94	•	•		•	+ :
Inconel 718-WS Type 3	+ 2.5/	+ 3.83b	- 2.54	ī	1	1	1	, i	•	•	ŀ	•	1
Inconel X-750 Type 1	عرن. ٠ -	+ 1.578	- 1.3%	•	•	1	•	- 2.84	,	•	•	•	4 (B.O.
Inconel X-750 Type 3	+ 1.3db	+ 1.476	e06.0 -	•	•	,	•	- 3.08	ī		,	,	+
AISI 301-CW Type 3	- 0.82	+ 0.51	+ 1.02	+ 1.44	•	•	•	,	,	,	1	+ 3.79	,
AISI 303-Se Type 1	- 4.59°	+ 4.236	+ 3.50	- 0.55ª	1	ı	•	,	ı	•	11	• 0.12	•
Al 2219-T6 Type 3	+ 2.73	+ 1.50	- 2.18	+ 0.46	,	•	1	ı	,	1	•	- 1.47	•
Al 2219-To Type 2, Trans.	+111.0d	O.	- 1.13	,	•		•	•	•	•	•	•	•
Al 2219-To Type 2, Radial	+ 9.374	+ 1.49	Q.	•	•	ı	•	•	٠	•	•	•	•
Beryllium Type 4	-70.164	+ 3.04	- 2.35	•	,	ı	•	•	ı	1	•	•	•
Ti A-110-AT-E11 Type 2	+ 7.754	+ pac.4 +	+ 5.524	•	,	•	•	•	•	•	•	•	•
Statistical Significance Probability:	11flcance	Probab11		o 06:0,≢¢	0.90 € № € 0.95	1	0.954040.99	0.9944					

Table 3-25

Percent Change: Tensile Yield Strength

						Test Cond	Condition (Control	١.	Irradiated)				
Material	C - A1	C - C1	B - B1	DA - DA1	DA - AL	DB - DB1	Da - Dai	Dc - Dc1	Dc - Dci	10 _G - 0 _G	Po - Poi	EA - EA1	EC - E3
Inconel 718 Type 3	+ 9.27		+ 2.45d + 1.78c	+ 1.03ª	+ 2.188	+ 2.70	+ 0.12	- 1.7.9	- 0.13ª	855	- 1.0ca	•	÷ 5. €1æ
Inconel 718 Type 1	+13.84	+ 7.994	+ 7.69d + 5.01 ^d	1	ı	•	-1	- 0.95	1	,	1	•	+
Inconel 718-WS Type 3	+ 7.30°		+ 5.16b + 3.20a		n	•	ı	ı	,	,	ı	•	ı
Inconel X-750 Type 1	+39.14	+26.0 ⁴	+24.3 ^d	ı	1	•	ı	+ 0.21	•	•	•	1	ी, ं ं
Inconel X-750 Type 3	+35.94	+22.54	+20.1d	•	1		,	+ 1.04	,	ı	•	1	2
AISI 301-CW Type 3	+ ٥. فيط	+ 5.144	+ 5.144 + 3.514	+ 2.26b	ı	ı	,	•	•	'	ı	+ 1.42	T.
AISI 303-Se Type 1	+34.79	+22.24	+23.84	+ 5.918	•	•	,	,	,	ı	•	+ 2.02	ı
A1 2219-T6 Type 3	+28.34	+ 5.28	- 1.19	+ 0.49ª	•	•	ı	ı	•	1	•	- 3.20ª	•
Al 2219-To Type 2, Trans.	447.4d	+ 1.86	S.	•		•	•	,	•	•	•	1	• 1
Al 2219-Tó Type 2, Radial	+51.54	+ 3.48	+ 3.48 + 1.08	•	a	•	•	,	•	ı	ı	ı	1
Beryllium Type 4	-70.24	+ 3.048	+ 3.02	•	•	1	ï	ı	ı	ı	١	•	ı
Ti A-110-AT-E11 Type 2	+ 9.804	+ 6.31 ^d	+ 6.314 + 7.894	•	'	7	•	•	•	•	•	•	•
Statistical Significance Probability:	11flcance	Probe bil		0 06.03	0.90		0.95 &c £ 0.99	0.9944	ם				

Table 3-26

Percent Change: Motched Tenaile Strength

						Test Cond	Test Condition (Control	•	Irredisted)				
Marcertal	C - A1	c - c1	B - Bt	DA - DA1	DA - DAL	nad - ad	D _B - D _{B1}	Dc - Dc1	De - Dei	100 - 001	Po - Poi	EA - EAL	Lc - Lc1
Inconel 718 Type 3	+ 7.13 ^d	9 TO T +	- 0.23	•	•	•	-	**5** -	•	•	•	•	- 3.51
Inconel 718 Type 1	- 9.40°		+ 8.194 + 3.42	•	•	•	ı	- 3.70	ı	•	•	•	- 0.22
Inconel 718-WS Type 3	+ 2.34		+ 1.85 + 0.94	,	ı	1	•	ı,	1	•	•	1	•
Inconel X-750 Type 1	+22.5d	+14.70	+11.14	i	•	1	•	- 2.22	,	,	,	•	- 0.95
Inconel X-750 Type 3	+19.24	+12.3 ^d	+10.24	•	1	•	•	- 1.96	•	•	1	•	+ 2.82
A131 301-CV Type 3	+ 3.384	+ 4.534	+ 4.53 ^d + 1.36 ^b	+ 1.77 ^b	•	,	•	ı	,	•	1	+ 2.024	•
AISI 303-3e Type 1	+19.5 ^d	+ 9.164	+ 9.164 + 8.066	+ 4.114	1	1	ı	,	1	ı	ı	+ 4.18b	ı,
Al 2219-T6 Type 3	+11.7ª	+ 0.59	- 5.31°	+ 1.87	•	ı	,	ı	1	•	1	-2.05	•
Al 2219-T6 Type 2, Trens.	+ 3.63	2	- 4.37	,	,	1	,	ı	1	1	•	,	,
Al 2219-T6 Type 2, Redial	+13.3°	- 0.75	- 0.75 + 0.52	ı	ļ	•	ı	1	,	1	ı	ı	ı
Beryllium Type 4	8.8°	+23.2	+ 0.64	•	•	•	•	•	ı	ı	,	•	í
Ti A-110-AT-E11 Type 2	- 7.31 ^d		ó.73 ^d + 2.06 ^e	•	•	•	ı	•	•	1	•	•	
Statistical Significance Probability:	1f1cance	Probeb11.	1	.0 06.0=	0.90£ b£ 0.95		0.956 c6 0.99	0.99£d					

Table 3-27

Percent Change: Notched-to-Unnotched Tensile Strength Ratio*

						Test Cond	Test Condition (Control	1.	Irradiated)				
Mater1a]	C - A1	ro - c	в - в	DA - DA1	DA - DAI	18 ₀ - 8 ₀	DB - DB1	Dc - Dc1	Dc - Dci	Pp - Pp1	D _D - D _D	EA - EA1	Ec - Ec1
Inconel 718 Type 3	+ 5.67	ηη·η +	+ 2.00	ı	•	1	1	- 5.0	•	ı	•	•	- 4.35
Inconel 718 Type 1	+ 8.55	+ 5.98	+ 4.65	1	ı	•	1	4.84	1	•	,	•	
Inconel 718-WS Type 3	- 1.26	- 2.53	+ 3.49	ı	•	•		11	1.4	ı	1	ı	•
Inconel X-750 Type 1	+23.3	+12.9	+12.4	•	ı	,	ı	+ 0.83	•	ı	1	1	•
Inconel X-750 Type 3	+17.2	+10.3	+11.8	,	ı	ı	1	+ 1.12	ı	ı	•	•	+ 2.35
AISI 301-CW Type 3	+ 4.23	+ 4.23	+ 0.98	+ 0.93	ı	1	ı	1	•	,	•	+ 1.28	•
AISI 303-Se Type 1	+14.80	+ 5.21	+ 4.58	87.4 +	1	ı	•	•	1	•	ı	+ 4.59	•
Al 2219-Tó Type 3	+ 8.70	- 1.09	- 3.16	+ 0.81	•	•	ı	ı	ı	•	ı	•	•
Al 2219-Tó Type 2, Trans.	- 7.38	ı	- 3.28	١.	ı	ı	•	1	ı	1	1	•	•
Al 2219-T6 Type 2, Radial	+ 3.97	- 1.59	+ 0.80	1	•	•	,	ı	,	1	•	,	•
Beryllium Type 4	+205.0	+21.05	+ 3.28	•	t	1	ı	,	•	•	,	•	•
T1 A-110-AT-E11 Type 2	-14.2	-11.9	- 3.85	•	1	•	ı	•	•	•	•	•	•

*No statistical analysis was performed on these ratios.

Table 3-86

ACTUAL TO COMPANY THE SAME SAME

Percent Change: Percent Elongation

						Sec. Cond	That Condition (Contact	1					
Material	C - M	0 - C	B - 31	DA - DA1	DA - DAL	н ₀ - ₆	140 - 60	1 5	Pc - Pci	10g - Qg	- Pa - Pa	IA - BA	Fc - Fc1
Inconel 718 Type 3	S S	+ 3.17	+ 9.26	-7.	+0.09	+3.20	-10.40	+ #.70	+15.5	-44.6	-40.80ª	•	- 7.82
Inconel 718 Type 1	-33.12	+ 1.60	+ 1.60 + 4.59	•	1	ı	1	+14.05	•	•	,	•	.86
Incomel 718-WS Type 3	- 7.29	NC	+43.7°	,	•	•	•	•	•	ı	•	•	•
Inconel X-750 Type 1	-23.3 ^d	-15.3°	-1.22	•	1	ı	1	-16.6	•	•	ı	٠	- 3.17
Inconel X-750 Type 3	-18.9 ^d	-11.5 ^d	-3.72	•	1	ı	•	+ 1.37	•	•	•	•	- 3.74
AISI 301-CN Type 3	- 0.98ª	- 1.96	- 9.37°	-3.14	•	1		•	•	•	•	- 6.99	•
AISI 303-Se Type 1	-20.4c	-10.2	-16.7	-6.78	•		,	1	•	1	,	- 6.03	•
A1 2219-T6 Type 3	-12.48	+11.0	- 3.43	-1.05	ı	•	1	1	•	•	ı	-13.08 ^b	•
Al 2219-T6 Type 2, frans.	-70.1 ^d	-12.2	- 3.02	•	ı	ı	ı	ı	:	•	1	1	•
Al 2219-T6 Type 2, Radial	-56.7°	+ 6.32	+ 4.59	•	1	1	•	,	•	ı	ı	•	•
Beryllium Type 4	٠	•	+59.0₽•	•	ı	1	•	,	•	•	•	•	•
Ti A-110-AT-E11 Type 2	-74.9 ^d	-34.4c	-22.4₽	•	-	1	•	•	•	•	i	•	,
Statistical Significance Probability:	1f1cance	Probab11	1	0 06.03	0.90 4b 4 0.95		0.95 &c & 0.99	0.99£d					

*See specimen data in Table 3-8

Table 3 29

Percent Change: Reduction 1. Area

						Test Cond	Test Condition (Control	1	Irradiated)				
Material	C - A1	C - C1	B - B1	DA - DA1	DA - DA!	я ₀ - е ₀	¹⁸ О - ва	Dc - Dc1	Dc - Dc1	τα _α - α _α	Po - Poi	EA - EA1	Ec - Ec1
Inconel 718 Type 3	+25.73	-16.96	- 3.28	- 9.75ª	-10.42ª	-11.48	-10.45	+38.67	+15.5	-26.3	-40.77	•	-17.3ª
Inconel 718 Type 1	+ 2.92		+ 9.178 + 1.148	•	•	1	,	- 5.26	1	11	•	•	ON.
Inconel 718-WS Type 3	-48.34	-45.1 ^d	+20.1	ı	•	•	4	1	1	1.	•		1
Inconel X-750 Type 1	+ 2.09	+ 5.67	- 4.80		1	•	•	-10.9ª	1	,	•	1	+ 8.87
Inconel X-750 Type 3	- 5.409	- 6.0	+ 0.56	ı	•	1	1	- 7.53ª	ı	ı	•	•	- 1.17ª
AISI 301-CW Type 3	-11.0ª	-11.0	- 9.50	- 9.80	ı	,	•	•	•	•	+45.03	•	•
AISI 303-Se Type 1	8.09	- 7.68	+16.3	-27.3	ı	ı	•	•	,	•	-11.º	ı	1
A1 2219-T6 Type 3	+58.8b	+29.19	+29.1° +88.5°	+01.74	•	1	1	ı	ı	ı	+13-409	ı	•
Al 2219-T6 Type 2, Trans.	-36.6	- 4.90	- 3.42	ı	•	t	ī	•	1	1	,	'	'
Al 2219-T6 Type 2, Radial	-27.6	- 8.55	-13.4	•	,	1	•	1	1	1		'	,
Beryllium Type 4	•	•	٠	,	•	ı	•	,	,	ŧ	,	•	•
T1 A-110-AT-E11 Type 2	+12.6	+20.2ª	+ 1.35	•	·	•	•	•	1	ı	•	•	•
Statistical Significance Probability:	nificance	Probabil.	1 1	0 06.03	0.90 6 6 6 0.95		0.95 4c £0.99	0.99 £d					
*See specimen data in Table 3-8	ets in Tal	3-8											

assume that all specimens received the same incident radiation, when in reality they did not. The integrated neutron flux received by the specimens, as tabulated in Table 3-1, ranged from 4 x 10¹⁷ to 10 x 10¹⁷ n/cm² (E>1 MeV) for all materials, with a worst-case condition for any material group being a factor of 2 between the lowest flux received by any specimen and the highest flux received by any specimen of the same material type. It is probable that some of the scatter evident in the material data is a result of the difference in incident radiation received by the specimens; however, it is not believed to have a significant effect on the interpretation of the results.

In the following discussion, the apparent changes in each of the tensile properties measured are discussed for each material irradiated. No effort was made to analyze the results on a basis of specimen type, that is, 1, 2, or 3. The purpose of the discussion is to indicate the general trends established by changes in the tensile properties of each material, regardless of specimen type. When a statement is made concerning some material, it will be the general trend experienced by all specimens as a group, regardless of type, unless specifically called out otherwise.

3.1.3.1 Ultimate Tensile Strength

Apparent changes in ultimate tensile strength, presented in tabular form in Table 3-24, were generally insignificant

(<5%). Tests performed in LN_2 without warmup indicated that slight (5-10%) to significant (>10%) changes had occurred in this property for titanium, aluminum, and beryllium; however, appreciable to complete recovery was evident after a room-temperature anneal.

For test condition Ai, slight to significant increases were noted for titanium ($\sim 8.0\%$ and for both transverse (12%) and radial (9%) aluminum specimens. A significant decrease ($\approx 70\%$) was noted for beryllium. Tests performed on specimens after a room-temperature anneal indicated complete recovery in this property for all specimens except titanium, which still exhibited a slight increase of approximately 7%. Tests performed on specimens after an elevated-temperature anneal ($> 80^{\circ}$ F) indicated that only insignificant changes in UTS had occurred for those specimens tested.

3.1.3.2 Tensile Yield Strength

Apparent changes in tensile yield strength, presented in tabular form in Table 3-25, were generally significant (> 10%). Materials tested at LN₂ temperature without warmup exhibited slight (5-10%) to significant changes in this property. Some recovery was apparent in the materials after warmup to room temperature, and only insignificant changes (< 5%) were noted for materials annealed at elevated temperatures.

For test condition Ai, increases of from 25% to 50% were evident for Inconel X-750, AISI 303-Se, and Al 2219 specimens. Except for beryllium, where a significant decrease of $\approx 70\%$

was noted, the remaining materials experienced increases of from 6% to 15% in this property. Tests performed after the specimens were allowed to warm up to room temperature (test conditions Ci and Bi) indicated that appreciable recovery had occurred; however, slight to significant increases in the property were still evident for Inconel 718, Inconel X-750, AISI 303-Se, and titanium. Specimens subjected to elevated annealing temperatures (>80°F) exhibited only insignificant changes in the property.

3.1.3.3 Notched Tensile Strength

Apparent changes in notched tensile strength, as presented in Table 3-26, indicate that slight (5-10%) to significant (>10%) changes occurred in this property for specimens tested at LN_2 temperature without warmup. Appreciable recovery was apparent in all materials after a room-temperature anneal, and only insignificant changes (<5%) were apparent after elevated-temperature (>80°) annealing.

Most of the materials subjected to test condition Ai experienced slight to significant increases; however, AISI 301-Cw and Al 2219-T6-transverse showed increases of less than 4%, and beryllium and titanium experienced decreases of 8% and 7%, respectively. Appreciable recovery was evident in specimens subjected to a room-temperature anneal; however, significant increases ($\approx 10\%$) were still apparent in this property for Inconel X-750, along with a slight ($\approx 8\%$) increase in the property for AISI 303-Se. Specimens tested after annealing treatments at elevated temperatures

(>80°F) experienced only insignificant changes in this property.

3.1.3.4 Notched-to-Unnotched Tensile Strength Ratio

Most materials tested experienced slight (5-10%) to significant (>10%) changes in this property, as evident in Table 3-27. No statistical analysis was performed on these data; therefore, significance probability is not included in the data. A recovery trend was established after specimen warmup to room temperature, and only significant (<5%) to slight changes in this property were experienced by the materials after annealing treatments at elevated temperatures (>80°F).

For test condition Ai, significant increases were noted for Inconel X-750 and AISI 303-Se. All other materials exhibited only slight changes except beryllium and titanium, which showed an increase of 205% and a decrease of 14.2%, respectively. The materials experienced appreciable recovery of this property after room-temperature anneal; however, Inconel X-750 specimens still exhibited increases of approximately 12%. Only insignificant to slight changes were apparent in this property for materials subjected to an elevated-temperature (>80°F) anneal.

3.1.3.5 Percent Elongation and Reduction in Area

As discussed previously, dimensional measurements were taken with a special test jig and micrometers. The two halves of each broken specimen were fitted together in the jig and elongation measurements made. Other measurements were made with vernier and dial micrometers. Since the specimens were radioactive (some as high as 20 r/hr) plexiglass body shielding was used. In addition, several operators were used to minimize the exposure to any one

operator. Measurements were made only once, and questionable data were not checked to the extent desired because of the personnel exposure required to separate particular specimens from the group.

Percent Elongation. Measured elongation values were checked against Instron chart elongation indication and the trends established by one were in general agreement with the other. Apparent changes in measured elongation, presented in Table 3-28, were generally significant (>10%). All materials tested at LN₂ temperature without warmup experienced slight (5-10%) to significant decreases in ductility with the exception of beryllium, where no change was discernable. Appreciable recovery was evident after a room-temperature anneal.

For test condition Ai, all materials with the exception of AISI 301-CW and beryllium exp. snced significant decreases in ductility of from 20% to 70%. The percent elongation of the beryllium specimens at LN₂ temperature was nil for both control and irradiated specimens. The AISI 301-CW specimens exhibited only insignificant decreases in the property. Appreciable recovery was evident after a room-temperature anneal; however, the titanium specimens still exhibited a significant decrease of 22.4% in this property. Although statistically significant, the 43.7% increase in percent elongation for Inconel 718-WS specimens is questionable because the percent elongation itself was small (<1.0%) and any error in the dimensional measurements appears quite large in the calculations for percent change from control to irradiated values.

Percent Reduction in Area. The percent changes in percent reduction in area are tabulated in Table 3-29. Slight errors made in the measurement of specimen diameters, widths, or thicknesses can result in large errors in the final calculations for percent change in the percent reduction in area of specimens. Because of the difficulties encountered in making dimensional measurements on irradiated specimens, the data are questionable and will not be discussed further.

3.1.4 Evaluation of Materials Tested

To supplement the general statements previously made, a summary of the results for each material is given below.

Inconel 718, Type 3

Test Conditions

Ai, Bi, Ci, D_{Ai}, D_{Ai}', D_{Bi}', D_{Ci}', D_{Ci}', D_{Di}', E_{Ci}
Ultimate Tensile Strength

Insignificant changes (< 5%).

Tensile Yield Strength

Insignificant changes for all test conditions except Ai, where an increase of 9.3% was evident.

Notched Tensile Strength

Insignificant changes for all test conditions except Ai, where an increase of 7.1% was noted.

Notched-to-Unnotched Tensile Strength Ratio

Insignificant changes for all test conditions except Ai, where an increase of 6.7% was noted.

Percent Elongation

Insignificant changes.

Inconel 718, Type 1

TOTAL SEPTEMBERS IN A

Test Conditions

A1, B1, C1, D_{C1}, E_{C1}

Ultimate Tensile Strength

Insignificant changes (< 5%).

Tensile Yield Strength

Increase of 13.8% for test condition Ai; increases of 7.7% and 5.0%, respectively, for test conditions Ci and Bi; and insignificant changes for test conditions D_{Ci} and E_{Ci} , indicating recovery.

Notched Tensile Strength

Increases of 9.4% and 8.2%, respectively, for test conditions Ai and Ci; insignificant changes for test conditions Bi, D_{Ci} , and E_{Ci} , indicating recovery.

Notched-to-Unnotched Tensile Strength Ratio

Increases of 8.6% and 6.0%, respectively, for test conditions Ai and Ci; insignificant changes for test conditions Bi, D_{C1} , and E_{C1} , indicating recovery.

Percent Elongation

Insignificant changes.

Inconel 718-WS, Type 3

Test Conditions

Ai, Bi, Ci

Ultimate Tensile Strength

Insignificant changes (< 5%).

Tensile Yield Strength

Increases of 7% and 5.2% for test conditions Ai and Ci, respectively.

Notched Tensile Strength

Insignificant increases.

Notched-to-Unnotched Tensile Strength Ratio

Insignificant changes.

Percent Elongation

Insignificant changes.

Inconel X-750, Type 1

Test Conditions

Ad, Bi, Ci, Dc1, Ec1

<u>Ultimate Tensile Strength</u>

Insignificant changes (< 5%).

Tensile Yield Strength

Increase of 39.1% for test condition Ai. Considerable recovery was evident for test conditions Ci and Bi; however, increases of 26.6% and 24.3%, respectively, were still evident. Insignificant changes for test conditions D_{Ci} and E_{Ci} .

Notched Tensile Strength

Increase of 22.5% for test condition Ai. Considerable recovery for test conditions Ci and Bi; however, increases of 14.7% and 11.1%, respectively, were still evident. Insignificant changes for test conditions D_{Ci} and E_{Ci} .

Notched-to-Unnotched Tensile Strength Ratio

Increase of 23.3% for test condition Ai. Considerable recovery for test conditions Ci and Bi; however, increases of 12.9% and 12.4%, respectively, were still evident. Insignificant changes for test conditions D Ci and E Ci.

Percent Elongation

Decrease of 23.3% for test condition Ai. Appreciable recovery for test condition Ci, although a decrease of 15.3% was still evident. Complete recovery for test condition Bi. Insignificant changes for test conditions D_{Ci} and E_{Ci} .

Inconel X-750, Type 3

Test Conditions

A1, B1, C1, D_{C1}, E_{C1}

Ultimate Tensile Strength

Insignificant changes (< 5%).

Tensile Yield Strength

Increase of 35.9% for test condition Ai. Some recovery evident for test conditions Ci and Bi; however, increases of 22.5% and 20.1%, respectively, were still evident. Insignificant changes for test conditions D_{Ci} and E_{Ci} .

Notched Tensile Strength

Increase of 19.2% for test condition Ai. Some recovery evident for test conditions Ci and Bi; however, increases of 12.3% and 10.2%, respectively, were still evident. Insignificant changes for test conditions D_{C1} and E_{C1} .

Notched-to-Unnotched Tensile Strength Ratio

Increase of 17.2% for test condition Ai. Appreciable recovery evident for test conditions Ci and Bi, although increases of 10.3% and 11.8%, respectively, were still evident. Insignificant changes for test conditions $D_{\hbox{\scriptsize Ci}}$ and $E_{\hbox{\scriptsize Ci}}$ $^\circ$

Percent Elongation

Decrease of 18.9% evident for test condition Ai. Considerable recovery for test condition Ci, where a decrease of 11.5% was noted. Complete recovery evident for test condition Bi. Insignificant changes for test condition D_{C1} and E_{C1} .

AISI 301-CW, Type 3

Test Conditions

A1, B1, C1, DA1, EA1

Ultimate Tensile Strength

Insignificant changes (< 5%).

Tensile Yield Strength

Increases of 6.6% and 5.1% for test conditions Ai and Ci, respectively. Insignificant increases for all other test conditions.

Notched Tensile Strength

Insignificant increases.

Notched-to-Unnotched Tensile Strength Ratio

Insignificant increases.

Percent Elongation

Insignificant decreases for all test conditions except Bi and E_{Ai} , where decreases of 9.4% and 7.0%, respectively, were noted.

AISI 303-Se, Type 1

Test Conditions

Ai, Bi, Ci, D_{A1} , E_{A1}

Ultimate Tensile Strength

Insignificant changes (< 5%).

Tensile Yield Strength

Increases of 34.7%, 22.2%, and 23.8%, respectively, for test conditions Ai, Ci, and Bi. Increase of 6.9% and insignificant increase of 2.6% for test conditions D_{Ai} and E_{Ai} , respectively, indicating recovery.

Notched Tensile Strength

Increase of 19.5% for test condition Ai; increases of 9.2% and 8.1% for test conditions Ci and Bi, respectively. Insignificant increases for test conditions D_{Ai} and E_{Ai} , indicating recovery.

Notched-to-Unnotched Tensile Strength Ratio

Increase of 14.8% for test condition Ai; increase of 5.2% for test condition Ci. Insignificant increases of 4.6%, 4.5%, and 4.6% for test conditions Bi, D_{Ai} , and E_{Ai} , respectively, indicating considerable recovery.

Percent Elongation

Decrease of 20.4% for test condition Ai. Considerable recovery evident after room-temperature anneal; only insignificant changes noted for test conditions Ci, Bi, D_{Ai} , and E_{Ai} .

Aluminum 2219-T6, Type 3

100mmのようなのであるからなった 100mm

Test Conditions

A1, B1, C1, DA1, EA1

Ultimate Tensile Strength

Insignificant changes (< 5%).

Tensile Yield Strength

Increase of 28.3% for test condition Ai. Increase of 5.3% for test condition Ci, indicating appreciable recovery. Insignificant changes for test conditions Bi, D_{Ai} , and E_{Ai} .

Notched Tensile Strength

Increase of 11.7% for test condition A1. Appreciable recovery evident for test conditions B1, C1, D_{A1} , and E_{A1} , with only a slight decrease of 5.3% noted in test condition B1; all other changes were insignificant.

Notched-to-Unnotched Tensile Strength Ratio

Increase of 8.7% for test condition Ai. Insignificant changes for all other test conditions.

Percent Elongation

Insignificant changes.

Aluminum 2219-T6-Transverse, Type 2

Test Conditions

A1, B1, C1

Ultimate Tensile Strength

Increase of 11.6% for test condition Ai. Insignificant changes (<5%) for test conditions Bi and Ci, indicating recovery.

Tensile Yield Strength

Increase of 47.4% for test condition Ai. Recovery evident for test conditions Ci and Bi, where changes of less than 2% were noted.

Notched Tensile Strength

Insignificant changes.

Notched-to-Unnotched Tensile Strength Ratio

Decrease of 7.4% for test condition Ai; insignificant changes for test conditions Ci and Bi.

Percent Elongation

Decrease of 70.1% noted for test condition Ai. Recovery evident for test conditionsCi and Bi, where only insignificant decreases were noted.

Aluminum 2219-T6-Radial, Type 2

Test Conditions

A1, B1, C1

Ultimate Tensile Strength

Increase of 9.4% for test condition Ai. Insignificant changes (<5%) for test conditions Ci and Bi.

Tensile Yield Strength

Increase of 51.5% for test condition Ai. Recovery is evident for test conditions Ci and Bi, where only slight increases of 3.5% and 1.1%, respectively, were noted.

Notched Tensile Strength

Increase of 13.3% for test condition Ai. Recovery is evident for test conditions Ci and Bi, where insignificant changes of less than 1% were noted.

Notched-to-Unnotched Tensile Strength Ratio

Insignificant changes.

Percent Elongation

Decrease of 56.7% for test condition Ai. Recovery is evident for test condition Ci and Bi, where only insignificant changes were noted.

Beryllium, Type 4

Test Conditions

Ai, Bi, Ci

Ultimate Tensile Strength

Decrease of 70.2% for test condition A1. Appreciable recovery for test conditions C1 and B1, where insignificant changes of less than 4% were noted.

Tensile Yield Strength

Decrease of 70.2% for test condition A1. Appreciable recovery for test conditions C1 and B1, which indicated insignificant increases of less than 4%.

Notched Tensile Strength

Large changes were noted in these data for test conditions A1 and C1; however, the statistical analysis of Section 3.1.2 established the significance probability of these changes to be less than 0.90.

Notched-to-Unnotched Tensile Strength Ratio

Large changes evident for test conditions A1 and C1; however, these would probably be statistically insignificant because of the significance probability assigned to changes in notched tensile strength.

Percent Elongation

The percent elongation of these specimens at LN₂ temperature was nil for both control and irradiated specimens. An increase of 59% was noted for test condition Bi; however, its significance probability was less than 0.90.

Titanium A-110-AT-Eli, Type 2

Test Conditions

A1, B1, C1

Ultimate Tensile Strength

Increases of 7.8% and 6.5% for test conditions Ai and Bi, respectively. Insignificant increase of 4.6% for test condition Ci.

Tensile Yield Strength

Increases of 9.8%, 6.3%, and 7.89% for test conditions Ai, Ci, and Bi, respectively.

Notched Tensile Strength

Decreases of 7.3% and 6.7% for test conditions Ai and Ci, respectively. Insignificant increase (<5%) for test condition Bi.

Notched-to-Unnotched Tensile Strength Ratio

Decrease of 14.2% and 11.2% for test conditions Ai and Ci, respectively. Insignificant decrease (< 5%) for test condition Bi.

Percent Elongation

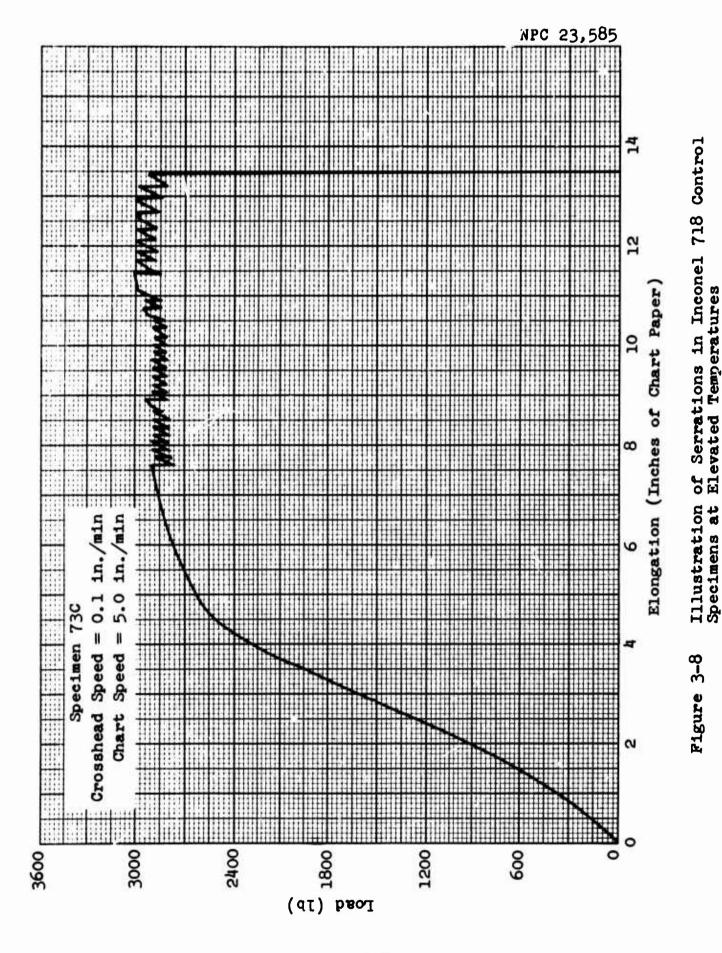
Decrease of 75% evident for test condition A1. Some recovery evident for test conditions C1 and B1, where decreases of 34.4% and 22.4%, respectively, were noted.

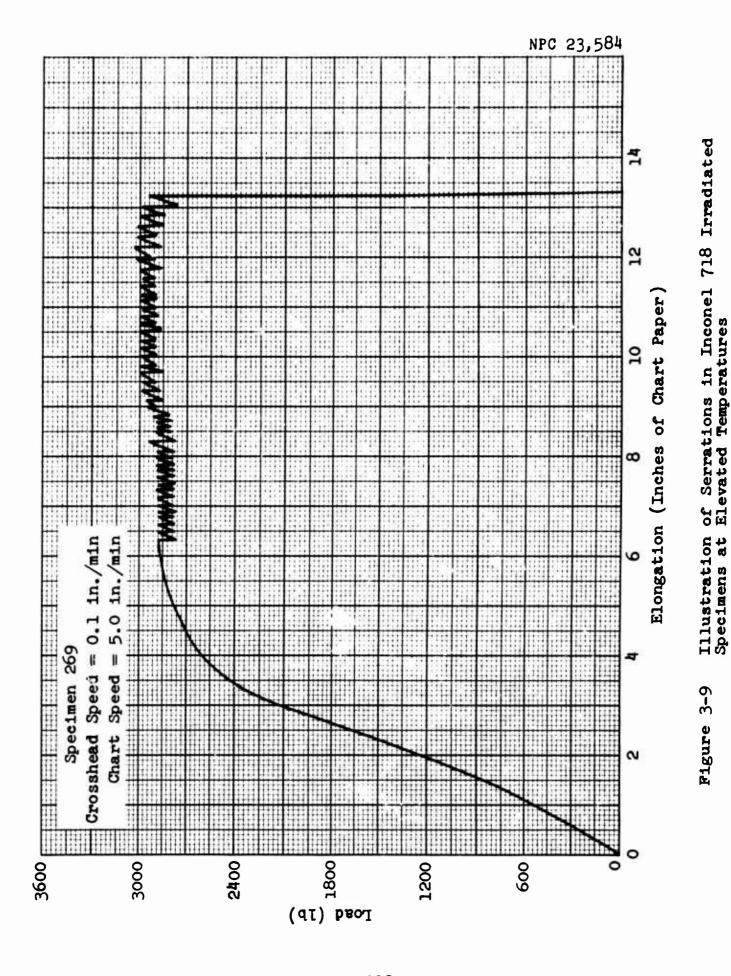
3.1.5 <u>High-Temperature Effects on Inconel Steel Specimens</u>

Inconel X-750 and Inconel 718 tensile specimens, both control and irradiated, exhibited marked serrations in their load-deflection curves (Figs. 3-8 and 3-9) when pulled in tension to break at temperatures above 540°F. All equipment was completely checked out and it was decided that these serrations were a specimen characteristic and not due to the test equipment. As a final check several specimens were returned to WANL for testing in their laboratory. These tests showed conclusively that the serrations were definitely a specimen characteristic for both Inconel X-750 and Inconel 718 specimens at temperatures above 540°F.

3.1.6 Conclusions

As a result of the radiation environment of this test, all materials tested at LN₂ temperatures without warmup experienced statistically significant changes in their tensile properties. Generally speaking, strength increased and ductility decreased for all materials except beryllium, which exhibited a highly significant decrease in strength. Subsequent recovery was apparent for all materials after annealing treatments. In general, the recovery was complete; however,





materials. From the results of this test alone, it would appear that these materials could be ranked, in order of their decreasing resistance to the effects of radiation, as follows; AISI 301-CW, Inconel 718, AISI 303-Se, Inconel X-750, Al 2219-T6, titanium, and beryllium, with AISI 301-CW being the most resistant and beryllium the least resistant.

3.2 Resistivity Tests

3.2.1 Data Presentation

The resistivity specimens received an average integrated neutron flux of 4.5 x 10^{17} n/cm² (E>1 MeV). Detailed dosimetry data are presented in Appendix B.

Data taken on the resistivity specimens during irradiation are presented in Table 3-30. Only resistivity specimen No. 1 was monitored during the irradiation. The bridge output voltage became erratic approximately 6 hr after termination of the first irradiation period and remained unstable until the experiment was removed from the area. The data recorded after the readings became erratic are not included in the table. A plot of representative data taken from Table 3-30 is presented in Figure 3-10. The postirradiation test data and results of measurements taken on the other three resistivity specimens are presented in Tables 3-31 and 3-32.

3.2.2 Discussion and Analysis of Results

The bridge output voltage increased by approximately 3150 μv as the reactor was brought to a power level of 3 Mw and traversed

Table 3-30

History of Resistivity Specimen During Irradiation

te (March 1965)	12	12	12	12	12	12	15	12
me	1400	1430	1500	1600	1720	1805	1838	1847
wer Level (Nw)	0	0	0	0	0	0	0	0
radiation Time	0	0	0	0	0	0	0	0
sistivity Input Current	0.100009	0.100061	0.099999	0.099999	0.099997	0.099989	0.099982	0.0999980
ltage Across 2.2525 g (mv)	225.271	225.388	225.248	555.549	225.245	225.226	225.210	225.205
LO	-1.6	+3.1	5.5	107.0	106.0	70.5	57.0	56.0
ble Current (amp)	0.100899	0.100882	0.100826	0.100745	0.100779	0.100781	0.100815	0.100806
ltage Across 2.2525 o (mv)	227.275	227.238	227.112	226.930	227.006	227.010	227.086	227.067
ble Voltage (v)	0.136514	0.136694	0.136710	0.136570	0.136560	0.136100	0.134620	0,133620
hle Resistance (ohms)	1.352976	1.354988	1.355900	1.35550	1.355044	1.350452	1.3353171	1.325516

Date (March 1965)	12	12	12	12	12	12	15	12
Time	1900	1916	1940	2010	2100	2200	2300	2347
Power Level (Mw)	0	3	3	3	3	3	3	Retracted
Irradiation Time	0	50.	1.25	2.75	5.55	8.25	11.25	13.,
Resistivity Input Current	0.099992	0.099394	0.099889	0.100003	0.100015	0.099997	1,666660.0	6.099999
Voltage Across 2.2525 g (mv)	225.232	225.238	225.00	225.258	225.285	225.244	225.237	225.248
Bridge Voltage (Eo) (µv)	40.05	1310.0	3150.0	3212.0	3242.0	3249.0	3315.0	327.0
Cable Current (amp)	0.100792	0.100792	0.100776	0.100810	0.100838	0.100802	0.100695	0.100807
Voltage Across 2.2525 p (mv)	227.034	227.034	227.00	227.075	227.139	227.058	226.817	227.070
Cable Voltage (v)	0.133960	0.134020	0.133940	0.134110	0.134500	0.134660	0.134620	0.134700
Cable Resistance (ohms)	1.329073	1.329669	1.329085	1,330324	1.333822	1.335886	1,3369081	1.336216

te (March 1965)	12	13	13	13	13	13	13	1.3
me	2354	2000	0019	0100	0200	0300	0440	0090
wer Level (Mw)	Retracted	3	3	3	3	0	0	3
radiation Time	13.4	13.65	14.25	16.3	19.3	21.1	21.1	21.55
sistivity Input Current	0.099999	0.099996	0.0999987	0.100005	0.099996	0.100003	0.099978	0.100007
Lage Across 2.2525 g (mv)	225.249	225.242	225.221	225.262	225.242	225.259	225.202	225.267
'dge Voltage (E.) (uv)	247.0	3240.0	3231.0	3335.9	3274.5	250.5	198.2	3162.5
ble Current (amp)	0.100763	0.100596	0.100825	0.100823	0.100788	0.100867	0.100828	0.10 766
ltage Across 2.2525 g (mv)	226.969	226.593	227.109	227.105	227.025	227.203	227.117	256.976
ble Voltage (v)	0.134660	0.134400	0.134150	0.134410	0.134225	0.133600	0.133465	0.133615
hie Resistance (ohms)	1.336403	1.336037	1 330523	1 333158	1.331755	1.324516	1.3236891	1.325992

Π		T		Π		Π]		1	Π	I	Π	I		
1500	18.55 0.100001 225.253 3398.0 0.100850 227.165	1.335656	0300	0.100018	225,292	0,100862	0.135102	1.33947	14	1900	132.55	0.099995	3861.3	0,10081	0.136570	1 35469
1400	45.55 0.130000 225.250 3355.6 0.100811 227.078	1.335072	0100	0.099993	225.235	0.100797	0.134990	1.339226	12	1700	126.55	0.099986	3938.2	0.100807	0.137802	SHOWS !
	3.55.25 0.100006 225.265 3.339.0 0.100868	1.331393	2300	0.099995	225.246	0.100874	0.134680	1.335130	14	1500	120.55	0.099990	3902.5	0,100805	0.137640	SECTION
1100	36.55 0.100000 225.252 3315.0 0.100820 227.099	1.330033	13 2100	0.099995	225,246	0.100901	0.134742	1.335388	14	1300	114.55	0.100002	3780.5	0,100820	0.137256	351365
1015	34.3 0.100007 225.266 3269.5 0.100761 226.966	1.329829	1900	60.55	225.270	0.100870	0.134860	1.336968	14	1100	108.55	0,0999999	3641.0	0.100841	0.135195	N. K. A.
0900	30.55 0.100000 225.251 3273.0 0.100755 226.951	1.329462	1800	57:55	225.270	0.101096	0.134964	1,335008	14	0060	102.55	0.099994	3571.5	0,100790	0 135,031	10000
13	3.27.55 0.100001 3.22.253 3.22.253 0.100850 0.227.188	0.133770	13	54.55	225.260	0.100851	0.134898	1,337597	2	0000	3 55	0.0999990	3523.0	0.100764	226.972	0.134005
0700	3 24.55 0.099988 225.225 3236.5 0.107859 227.186	0.13315	13 1608	48.95	225.253	3403.7	0.134765	1,334888	Ē	0500	30 55	0.10001.0	3531.1	6,101053	227.622	0.134001
Date (March 1965)	Power Level (Na) rridiation Time Resistivity Input Current Voltage Across 2.2525 g (mv) Bridge Voltage (Eg) (µv) Cable Current (amp)	111	Date (March 1965)			Bridge Voltage (Eo) (µv)	Voltage Across 2.2525 g (mv)	e Resistance (ohms)	220 CHO	Date (march 1902)	Power Level (Nw)	t Current	age Across 2.2525 g (mv)	e Current (amp)	Voltage Across 2.2525 g (mv)	e Voltage (v)

Table 3-30 (Cont'd)

пппппп	fillitim	CHILIDITA
1100 130,55 0.100009 225.272 375.71 0.10000 0.15051 1.354267	16 0300 288 59 0.099999 225 528 3797.2 0.100720 0.100188 0.1351648	16 2000 23,55 0.100013 41316 0.01045 0.10145 0.10145 0.175051 1.350435
15 0900 174,55 0.099999 225.247 3667.5 0.10084 0.135254 1.341250	16 0100 3 222-55 0.100002 225-55 3751-6 0.10008 227-206 0.17259 1.360776	1600 1800 23.55 0.099999 125.246 410.8 0.101164 0.101164 0.13652 1.346869
15 0700 183 1845 0.093987 225, 227 3201.0 0.1039 0.14990 0.14990 1.338622	15 2300 21 215.55 0.0999997 225.245 36.32.0 0.100792 0.137053 0.137053 1.359859	1600 1600 23.55 0.099996 25.220 404; 0.100863 0.120961 0.1347778
15 0500 18:55 0.100010 225:273 225:273 0.10002 0.15000 1.35500 1.35500	15 2100 3 3 3 210.55 0.099991 225.230 515.6 0.1185 0.137381 1.557721	16 24.55 0.100000 0.100000 178 0.10117 0.227.899 0.13550 1.346752
15 0300 13 0.1000.5 225.27 3628.6 0.100099 0.135140 1.339359	15 1900 204.55 0.100017 225.59 3.606.0 3.606.0 6.10919 0.1361290	1500 25 25 25 25 25 25 25 27 20 20 20 20 20 20 20 20 20 20
15 0100 3 13 0.100001 225.254 1675.0 0.100931 227.348 0.13550 1.341015	15 1700 1700 1388 55 0.1000006 225 265 3664.2 0.100325 227 334 0.110936 1.356809	16 0900 246.55 0.099999 383.6 0.101469 0.101469 0.101469 0.150320
14 2300 34.55 0.100004 225.260 3.755.260 0.100368 227.205 0.13757 1.345887	150 1500 13.00 1.00 1.00 1.00 1.00 1.00 1.00 1	16 0700 340.55 0.099989 2525 3844 30.100784 0.100784 0.13938
14 2100 3 13 55 0.099982 225 210 3799 4 0.100867 227 205 0.150699	15 1300 1300 136.55 0.099996 225.243 3716.0 277.422 0.136455 1.351521	16 0500 34.55 0.09991 3702.1 3702.1 0.100831 227.123 0.150556
Date (March 1955) Time Power Level (Ma) Fower Level (Ma) Irradiation Time Resistivity input Current Voltage Across 2.2552 g (mv) Gable Current (amp) Voltage Across 2.2525 g (mv) Cable Voltage (v) Cable Tesistance (ohms)	Date (March 1965) Time Fower Level (Mw) Irradiation Time Assistivity Input Current Voltagy Across 2.2525 g (mv) Cable Unrent (amp) Voltage Across 2.2525 g (mv) Cable Voltage (V) Cable Fesistance (ohms)	Date (March 1955) Time Power Lavel (Wm) Fordiation Time Assistivity Input Current Voltage Across 2.2555 g (mv) Bridge Voltage (E.A.) Voltage Across 2.2525 g (mv) Cable Voltage (V) Cable Voltage (V) Cable Voltage (V)

Table 3-30 (Cont'd)

THE CHARTCH TOOL	9	01]	JΤ	JT	7.7	7	7.7
	2200	2356	0500	0500	0040	0050	00/00	2060
Power Level (Nw)	_	0	0	0	m	3	3	m
Irradiation Time	285.55	29 .2	291.2	291.2	293.45	296.45	302.4	308.45
Resistivity Input Current	0.100000	0.099964	0.099989	0.100011 J	0.099979		0.10000	6.5995.5
Voltage Across 2.2525 g (mv)	225.250	225.170	225.226	225.277	225.204	225.252	225.253	225.203
Bridge Voltage (Eo) (µv)	3942.2	1770.0	574.2	561.5	4558.2	3750.0	1.606	3926.0
Cable Current (amp)	0.098095	0.100683	0.100722	0.100783	0.100839	0.100839	0.130843	0.130778
Voltage Across 2.2525 9 (av)	229.061	226.790	226.877	227.015	227.142	227.142	1 227 . 150]	227.004
Cable Voltage (V)	0.137388	0.137212	0.136377	0.135185	0.134363	0.135335	0.135640	0.136000
Cable Resistance (ohms)	1.400560	1.362811	1.353994	1.351269	1.332450	1.342089	1 1.345061	1.349500
							75 - 150 - 150	A
- 1								
mile (march 1902)	W.	1300	45,00	1700	1000	27.00	22,64	0010
Power Leve	2				200			
Τ.	312 45	30 055	12 45¢	1 6:1	448. 44	344 45	350 45	246 BE
Restativity Innit Current	0 10000	2 000001	500000	00000	AC.	0 10000	A 600004	0 10000
Voltame Across 7.2525 & (mv)	225.256	225,238	225,236	225, 240	225.290	225.250	256.246	٠ĸ
\and	10K B	do	1926.2	4030.5	4085.3	4163.6	4137.2	2.05.0
Cable Current (amp)	6,100815	6,100829	0.100815	0.100508	0.100619	0.100727	0.100795	0.100769
Voltage Across 2 2525 & (mv)	227,086	227.118	227.088	226.396	227.097	226.889	227.041	18 622
	0.136388	0.135825	0.136196	0.135267	0.135452	0.135247	0.135130	0.134787
Cable Resistance (ohms)	1.352854	1.347082	3500	1.345833	1.343913	1.342708	1.340641	1.337583
1								
r	8		8.5	X .	×	85	86	1
Date (March 1902)	707	TX.	KA KA		XX 5.	NAME OF TAXABLE PARKS		2
- 1	300	3/5	335	377	37	3	37	35,
		_	~	~		-	***	***
Time	102.45	373.8	379.8	365.0	391.8	26.0	403.0	400.0
out Current	0.100007	0.100005	0.100000	0.100000	0.100000	0.00	66660.0	0.130004
ΔĮ	225.200	225.263	225.250	225.250	225.250	225.240	625.23	225.250
Bridge Voltage (Eq.) (pv)	4107.6	4222,2	4256.8	4365.6	4313.3	4479.7	4509.5	4504.5
D	0.100729	0.100825	0.100852	0.100573	0.100799	0.100/64	$^{\circ}$	0.100915
Voltage Across 2,2525 g (mv)	226.893	227.109	227.171	226.541	N.	ė	227.019	227.313
	0.134420	0.133912	0.132568	0.134200	0.134004	0.134948	0.135001	0.132894
(Cable Resistance (ohms)	1.334471	1.328162	1.314480	1.324354	1.329417	1.339248	1.339494	1.336709

Table 3-30 (Cont'd)

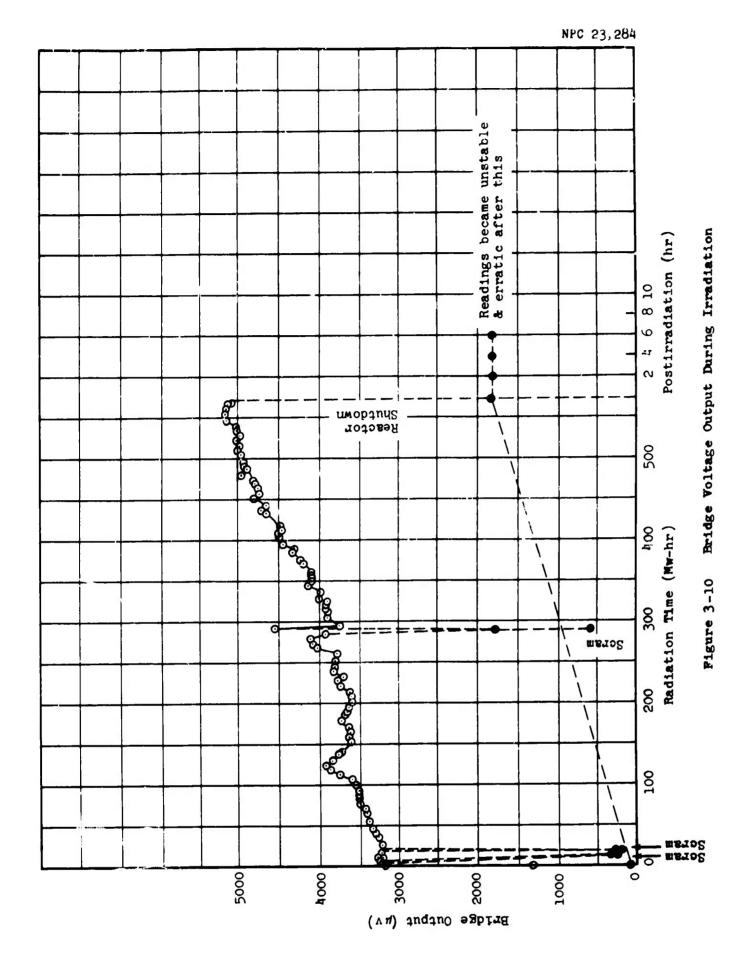
Date (March 1965)	18	18	19	19	19	19	19	19
Time	2100	2300	0100	0500	0300	0400	0090	0800
Power Level (Mw)	3	3	3	3	3	3	3	3
Irradiation Time	415.8	421.8	427.8	433.8	436.8	439.8	445.8	451.8
Resistivity Input Current	0.100004	0.099997	0.100001	0.100003	0.099996	0.099998	0.10001.0	0.100000
Voltage Across 2.2525 g (mv)	225.261	225.244	225.254	225.259	225.241	225.247	225.273	225.250
	4486.5	4522.2	1160.0	1488.2	4683.6	4734.3	4689.0	4814.5
Cable Current (amp)	0.100765	0.100751	0.100805	0.100782	0.100787	0.100735	0.100768	0.100668
Voltage Across 2.2525 2 (mv)	226.975	226.942	227.065	227.012	227.024	226.907	226.981	226.755
	0.134350	0.134687	0.132980	0.134260	0.134330	0.134225	0.134392	0.134302
Cable Resistance (ohms)	1.3333300	1.336830	1.319180	1.332182	1.332810	1.332456	1.333577	1.334108
Date (March 1965)	10	19	10	10	19	19	10 1	19
Time	1000	1200	1400	1600	1800	2000	2200	2400
Power Level (Mw)	3	3	3	3	3	3	3	3
Irradiation Time	457.8	463.8	469.8	475.8	481.8	487.8	493.8	4.69.8
Resistivity Input Current	966660.0	0.100004	0.100003	0.100007	0.099993	0.100000	0.0999988	0.100003
Voltage Across 2.2525 o (mv)	225.246	225.261	225.258	225.268	225.235	225.250	225.225	225.258
Bridge Voltage (Eo) (µv)	4753.2	4775.5	4768.5	4803.2	4970.5	4871.6	4943.6	4972.2
Cable Current (amp)	0.101021	0.100868	0.101405	0.100775	0.101365	0.100772	0.100895	0.100740
Voltage Across 2.2525 g (mv)	227.551	227.207	228.416	226.997	228.326	226.991	227.266	226.918
Cable Voltage (v)	0.133940	0.133810	0.134140	0.134524	0.134395	0.133230	0.133566	0.133461
Cable Resistance (ohms)	1.325862	1.326585	1.322814	1.334894	1.325852	1.322093	1.323811	1.324806
Date (March 1965)	20	20	20	20	20	20	20	20
Time	0504	0410	0000	0000	1000	1200	1400	1000
POWET LEVEL (MW)	× ×		,					
Irradiation Time	2000	2.4.3	517.0	243.0	559.3	232.3	241.3	5-1-5
rent	AAR AUA	***	NAME AND ADDRESS OF THE PERSON		N AXX	WX WXX	XX AXX	XXX XXX
Voltage Across 2.2525 g (mv)	562.503	663.66/	225.200	225.240	625.60	253.61	652.50	565.666
Bridge Voltage (En (pv)	4978.9	5039.1	5006.7	5048.1	5030.0	4903.2	5030.3	5140.9
1	ASA 488	200 200	AAL AFT	KILS KKK	AND SAN	000 500	256 00	519 455
VOITAGE ACTOSS C.COCO B (MV)	X 55 53 7	250:3(3	220.924	263.245	CC(: 35"	250.205	X 1333A7	251.543
Capte voltage Iv	0.133404	1.133171	0.13300	0.133575	0.133013	0.135007	10366110	0.13300

			***	XX			21	n
That Asses	20	20	22	2	77		123	
דמוב ושורוו דארטו	A GAN	2000	0100	2400	0115	0200	888	39
Time	No.T	2000	24.7		1	K		c
Down Town	*	•	'n		0	5		,
4	200	200	בעב א	571.3	573.25			
Irradiation Time	553.3	223.3	2000				1800 C C	00000
Destativity Innit Current								- KK KK
TO TO TO THE PARTY OF THE PARTY	1/30 300	555 557	525, 263	255.55	25.	225.230	C17.07	262.660
Voltage Across C.COCO M (mv)	563.633	1/11/11		1000	1000	ac.a.	X	0 1981
But den 170 to mo / 17 / 111	2177 4	1 5175.6	5100.3	>101C	1.00	1.0001	1000	
Dringe Voltage (Eo/ (PV)	7-11-						0.100795	0.1086.11
Cable Current (amp)		7.						L. L
X XCXE	XC1 700	227 031	727,004	227.046	227.009	267.722	77.	EC(- 122
Voltage Across 2.2525 g (mv)	251.150	15001		AXO E E X	A CANA	VICE V	X Contract	0.00
	1 1 2 2 X O C	13405	0.144/15	0.133005	0.13315	C. ISCOUL	1 01 1261 10 1	0.136.7
capie voltage (V)	0.133037	200					1 4 25000	1.115289
Anna Bastatanon Chma								1
TOTAL TOTAL DES								

ſ	**	16	10	21	21	21	22	77
Date (march 1905)	72	177		22.7	XXX	Occes	× 24 C	
	0815	1050	11110	1400	1050	0000	3	200
ľ		c	0	0	0	0	0	5
POWer Level (MW)	,							
Tresdant on Mane		11						× × × × ×
	B00000	A 000005	0 100195	0.099973	0.099987	0.099979	NO CON	カロスへんつ
Resistivity Input Current	0.099999	0.02222	77,77	*******	XXX	200 200	300	100 200
11 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1	076 566	070 070	225.090	227.130	777.666	402.622	262.622	1-2-/-52
LABE ACTUSS C.C.)C. M	21.7.1	2000	2 422X		1 202 2	0 0008	- C 6711#	1777 C
Bridge Voltage (F.) (UV)	726.6	+2982 · 0	6.)002		1323.3	2000		XXXX
10000	C.(DXX	7/000/17	10000	00000	0.100790	0.10000	05/501.0	10/331.0
Cable Current (am	0.100045	0.10094	0.1000		XXX XXX	200	100	020 000
1	227 150	727 385	227.100	CC2. /22	22(.031	50.022	1460.12	250.315
Lage Across C.C.C.J.	251 32	- XXXXX	X X X X X	330035	Deliter C	0.201.5	14180	
(4) Page (1)	0.132578	0.131320	0.130220	0.130317	0.131730	002767.0		2772
5		XOUXX	500100	1 202501	1 20200K	7 700770	17.17.1	いった。
Cahle Registance (chas)	1.314697	7.30000	L.CYLOYJ	1.691001	1.000001	21/24		111111111111111111111111111111111111111
,								

(March 1965)	55	57	1 5 (
	1645	1730	1045	1200	
wer Level (Mv)	0	0	0	0	
rradiation Time		‡ -	-		
stivity Input Current	0.100153		0.099932	0.099931	
age Across 2.2525 g (mv)	225.596		225.098	225.092	
ge Voltage (En) (pv)	*3510.0	*	**-0.3735	**-0.2°	
Cable Current (amp)	0.1007.0		0.100699	0.100217	
Age Across 2.2525 0 (mv)	226.851		226.826	220.415	
e Voltage (v)	0.131300		0.131510	0.134750	
Peststance (ohms)	1 1 303743		1.305971	1.340509	

*Unstable and intermittent. **Reading drifts. Readout in volts.



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Table 3-31
Postirradiation Resistivity Data

	nnealing Condition	Incone		Incon	e No. 3 el X-750	Incone	No. 4 21 X-750
Time (min)	Temperature (°F)	Bridge Voltage (10 ⁻⁵ v)	Resistance Change (10-3 Ω)	Bridge Voltage (10-5 v)	Resistance Change (10-3 Ω)	Bridge Voltage (10-0 v)	Resistance Change (10-3 Ω)
	-320			o .	0	o	0
10	-270			624	6.79	751	8.17
10	-220			629	6.84	803	8.74
10	-180			702	7.64	859	9.35
10	-130			790	8.60	854	9.29
10	- 80			932	10.1	847	9.22
10	- 30			1176	12.8	846	9.20
10	+ 70					854	9.29
30	+ 78					861	9.37
	-320	o	0				
5	-320 to -215	43	0.47				
5	-320 to -215	36	0.39				
10	-320 to -125	99	1.08				
30	-320 to + 37	765	8.32				
60	-320 to + 67	935	10.2				
60	-320 to + 67	932	10.1				

Table 3-32

Percent Recovery of Resistivity Samples

Annea:	ling Condition		Percent Recovery	
Time (min)	Temperature (°F)	Sample No. 2 Inconel 718	Sample No. 3 Inconel X-750	Sample No. 4 Inconel X-750
-	-320	U	0	0
10	-270	-	53	87
10	-220	-	53	93
10	-180	-	60	99
10	-130	-	67	99
10	- 80	-	79	98
10	- 30	-	170	98
10	+ 70	-	-	99
30	+ 78	-	-	100
5	-320 to -215	4	-	-
5	-320 to -215	4	-	_
10	-320 to -125	11	-	-
30	-320 to + 37	82	-	-
60	-320 to 67	100	-	-
б 0	-320 to 67	100	-	

into the closet. This transient effect was also apparent during reactor scrams, when the output voltage decreased by approximately 3000 µv as the reactor power fell off. The output voltage increased by 2000 µv during the irradiation to a value of approximately 5150 µv. This 2000-µv increase represents an increase in resistance of approximately 0.020 ohm. At the termination of the first irradiation period, the bridge voltage decreased to approximately 1850 µv as the power level decreased to zero. The voltage remained at this value for approximately 6 hr, at which time the readings became erratic. The 1850-µv output is of the same magnitude as the 2000-µv increase during irradiation, indicating an apparent permanent change in the resistance in the order of 0.018 to 0.020 ohm. Details of the resistance bridge circuit and a sample calculation are presented in Appendix C.

Postirradiation data taken several weeks after the irradiation still indicated erratic readings for specimen No. 1. The other specimens experienced apparent decreases in resistance of approximately 0.010 ohm after the annealing treatments discussed in Section 2.1.2. These data indicate that the resistivity specimens increased in resistance by some amount greater than 0.010 ohm during the irradiation.

3.3 Steel-Spring Specimen Tests

3.3.1 Data Presentation

The spring specimens received an average integrated neutron flux of $4.5 \times 10^{17} \text{ n/cm}^2$ (E>1 MeV). Detailed dosimetry data on all specimens may be found in Appendix B.

The results of the tests on the spring specimens are tabulated in Table 3-33.

3.3.2 Discussion and Analysis of Results

Any changes in the specimens as a result of radiation are not discernable from the data. There appears to be a slight increase in the amount of permanent set (after load removal) in the irradiated specimens as compared to the control specimens. The control specimens and the irradiated specimens were kept under load for the same period of time. The control specimens were stored in LN₂ during the storage of irradiated specimens for radioactivity decay. Upon removal of the loads, the measurements indicated that the free length of the irradiated specimens had decreased by an average of 3.89%, while the free length of the control specimens had decreased by 1.59%. This apparent difference of 2.3% is so small as to possibly be a result of measuring technique; however, it could be an effect of radiation. In any case the change is of such small magnitude as to probably be insignificant.

3.4 O-Ring Seal Tests

3.4.1 Data Presentation

The 0-Ring specimens received an average integrated neutron flux of 4.0 x 10^{16} n/cm² (E>2.9 MeV) and an average gamma dose of 2.6 x 10^{10} ergs/gm(C).

The temperature, during the irradiation period, of the fixture mounted on the outside of the LH_2 dewar is tabulated in Table 3-34.

Table 3-33

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Test	
Feet .	
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Spring	
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V2	
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Results	
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100	

		9	œ	3	<u>س</u>	æ	m	5	0	O.
	88	Change	+1.18	+1.83	+1.03	+2.38	+1.83	+2.15	+3.79	40.39
Spring Rate	After	(lb/in.)	128.5	128.0	127.2	128.9	128.0	128.4	131.4	128.0
Spri	Bafore	(lb/in.)	127.0	125.7	125.9	125.9	125.7	125.7	126.6	127.5
	After Defor- mation & Anneal	چ Onange			-1.30	-1.45			-2.48	-2.51
	After Demation 8	Length (in.)			2.5710 -0.93 2.5510 -1.30	2.6480			2.6410 -2.32 2.5355	2.5115
	After Anneal**	್ಲ್ Change			-0.93	-1.09			-2.32	-2.21
sth	After Anneal	Length (in.)		•	2.5710	2.6575 -1.09			2.6410	2.6195 -2.21
Pree Length	After Deformation*	% Change	-1.75	-0.37			-3.00	-4.30		
P	AF Defor	Length (in.)	2.5990 2.5545 -1.65 2.6525 -1.75	2.0725 -0.37			2.5185 -3.00	2.5830 -4.30		
	After Irradiation	ಸ್ Change	-1.65	-0.39	-2.08	-2.23	-3.05	-4.29	-4.22	-3.98
	After Irradiat	Length (in.)	2.6545	2.5824 2.6720 -0.39	2.6960 2.6400 -2.08	2.5870 2.5270 -2.23	2.5994 2.5170 -3.05	2.0992 2.5835 -4.29	2.7037 2.5895 -4.22	2.6787 2.5720 -3.98
	Before	(in.)	2.5990	2.5824	2.6960	2,5870	2.5994	2.6992	2.7037	2.6787
ring	Identification	Irra- diated				Section 2000	R1	R _{tt}	ВĞ	В.
Sp	Identi	Con- trol	×	B1	B2	ď,				

*Deformation to 130 lb at 1.0 in./min.

Table 3-34

Representative Temperatures for the Orifice Cement and the O-Ring Fixture

Date	T1me	Power Level (Mw)	Exposure	Temperature (°F)	
			(Mw-hr)	Orifice	O-Ring
3-12-65	1805	O	0	- 50	+ 48
	1904	3	1.25	+ 39	+ 64
	2300	3	11.25 13.4	+151 +124	+ 88 + 68
	2347	3 0 0 3 3 3 3 3 3 3 3 3 3	13.4	+110	+ 60
3-13-65	0100	ĭ	16.3	+145	+ 96
J- 1 J-0 <i>)</i>	0300	ď	21.1	+ 97	+ 58
	0440	O	21.1	+ 28	+ 40
	0000	3	21.55	+ 51	+ 52
	0/00	3	24.55	+128	+ 82
	1100	3	30.55	+150	+ 81
	1700 2300	. 3	54.55	+102	+ 88 + 88
3-14-65	0700	3	72.55 96.55	+161 +162	+ 97
	1100	3	108.55	+105	+ 92
	1700	1	126.55	+167	+100
	2300	3	144.55	+164	+ 98
3-15-65	0700	3	168.55	+164	+ 99
	1100	3	180.55	+163	+ 95
	1700	3	198.55	+168	+ 98
36 17	2300	3	216.55	+168	+ 98
-16-65	0700	3333333330003333333333333333333333000003333	240.55	+171	+ 99 + 99 + 98
	1100 1800	3	252.55	+176	+ 99
	2356	3	2/3.55 291.2	+174 +155	+ 98 + 80
-17-65	0200	Ö	291.2	+ 32	+ 37
-1	0700	3	302.45	+162	+ 97
	1100	3	314.45	+166	+ 97
	1700	$\tilde{3}$	332.45	+171	+ 91
	2300	3	350.45	+166	+ 91 + 84
-18-65	0700	3	373.8	+148	+ 95
	1100	3	385.8	+158	+ 96
	1700	3	403.8	+160	+ 91
-19-65	2300 0600	3	421.8 442.8	+158 +157	+ 85 + 82
-19-05	1200	3	460.8	+158	+ 82
	1800	3	478.8	+158	+ 82
	2400	, š	496.8	+154	+ 80
-20-65	0600	<u>3</u>	514.8	+157	+ 78
	1200	3	532.55	+156	+ 84
	1800	3	550.55	+156	+ 84
	2400	3	568.55	+152	+ 83
-21-65	0200	0	571.1	+ 28	+ 18
00 65	1400	O	571.1	+ 11	+ 31
-22-65	0400 1645	0	571.1 571.1	- 9 - 10	+ 28 + 28
-26-65	1200	0	571.1	-100	¥ 20
-27-65	0007	ž	571.45	- 40	
J 21-47	0100	3	574.1	+ 40	
	0400	ž	583.1	+146	
	1200	3	607.1	+150	
0 (-	5500	3	637.1	+155	
-28-65	0900	3	670.1	+155	
00 65	2300	3	710.8	+142	
3-29-65	0300	3	722.8	+151	
3-30-65	1800 1200	3	767.8 821.8	+151 +163	
J-30-07	2300	2	854.8	+166	
3-31-65 4-1-65	0900	3	884.8	+154	
	2300	٦	926.8	+159	
	0930	ž	955.3	+150	
4-2-65	2115	ž	994.3	+164	
	0900	3	1028.8	+163	
	2000	3	1060.43		
4-3-65 4-4-65	0500	3	1087.43	+268	
	1100	Ö	.104.18	+ 72	
	1700	o		+ 20	
	0730	0		- 12	
	2100	0		- 21	

3.4.2 Discussion and Analysis of Results

The data presented in Table 3-34 were picked at random to illustrate the temperature excursions inside the container during the irradiation and shutdown periods. All testing of these specimens was performed at WANL by Westinghouse personnel and no results are available for this report.

3.5 Cemented-Orifice Tests

3.5.1 Data Presentation

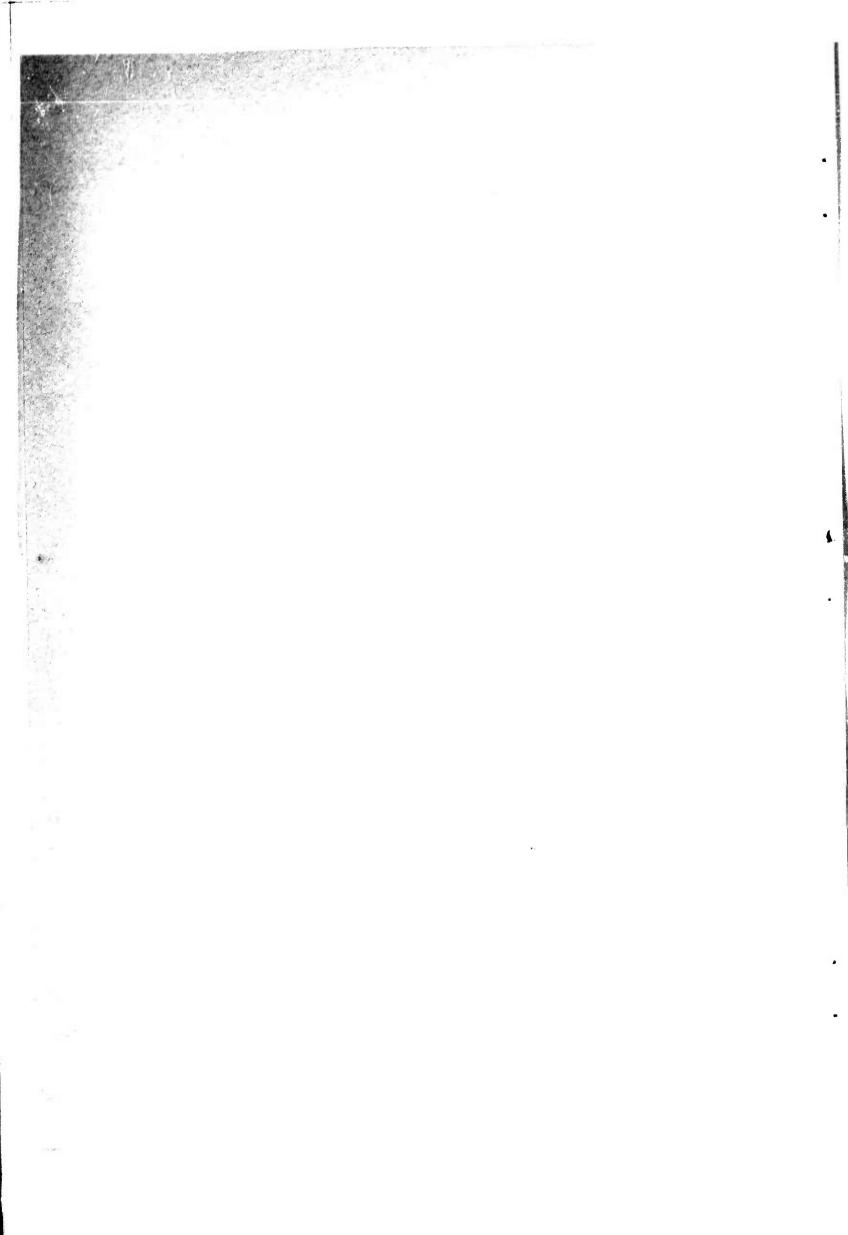
The cemented orifice specimens inside the north dewar received an average integrated neutron flux of 2.5 x 10^{17} n/cm² (E>1 MeV) and an average gamma dose of 1.0 x 10^{11} ergs/gm(C). The specimens mounted outside the north dewar received an average integrated neutron flux of 4.0 x 10^{17} n/cm² (E>1 MeV) and an average gamma dose of 2.0 x 10^{11} ergs/gm(C).

The temperature, during irradiation, of the container mounted outside the LN_2 dewar is tabulated in Table 3-34.

3.5.2 Discussion and Analysis of Results

The data in Table 3-34 were picked at random to illustrate temperature excursions during the irradiation period. All testing of these specimens was conducted by Westinghouse personnel at their facilities and the results are not available for this report.

APPENDIX A GTR RADIATION EFFECTS TESTING SYSTEM



APPENDIX A

GTR RADIATION EFFECTS TESTING SYSTEM

The GTR Radiation Effects Testing System is located in the Reactor Operations Area at the north end of the NARF complex. Figure A-l is a plan view and Figure A-2 is a cutaway view of the system. A closeup of the irradiation test cell and the reactor tank is pictured in Figure A-3. During operation, the reactor is moved into the closet-like structure built into the north wall of the GTR tank. Items to be irradiated can be located on the north, east, or west sides of the closet, as indicated in the figures.

The reactor closet is constructed of 1-in. aluminum plate and is partially covered by 1/4-in.-thick boral to attenuate thermal neutrons. The boral extends 36 in. east and west from the closet along the tank wall and 36 in: up and down from the horizontal centerline of the reactor core. The centerline is 57 in. above the test-cell floor.

The Ground Test Reactor (GTR) is a heterogeneous, highly enriched, thermal reactor that utilizes water as neutron moderator and reflector, as radiation shielding, and as coolant.

Maximum power generation is 3 Mw. The GTR, in an aluminum enclosure to facilitate cooling-water flow, is suspended by an open framework that is carried on a horizontal positioning mechanism at the top of the reactor tank. This mechanism permits the reactor to be positioned at distances ranging from

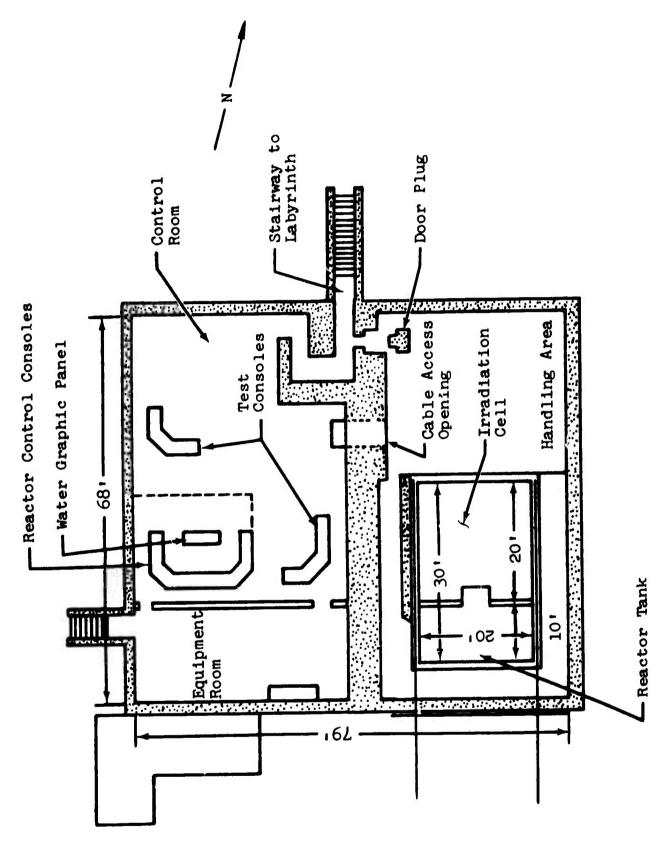


Figure A-1 Operations Building and GTR Facility

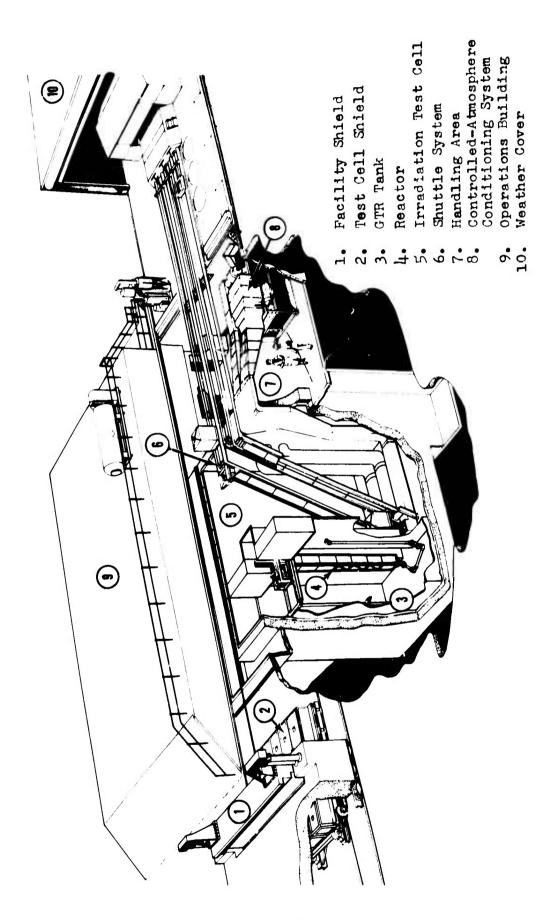
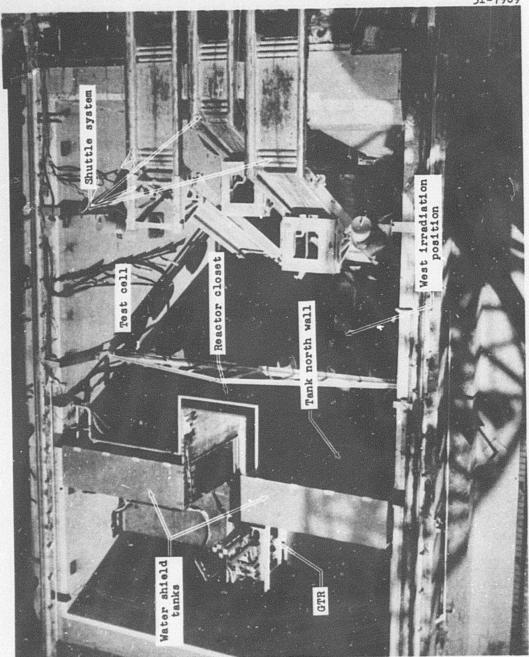


Figure A-2 Cutaway View of GTR Radiation Effects System



Pigure A-3 Irradiation Test Cell and Reactor Tank

2 to 87 in. from the north face of the closet.

Adjacent to the north wall of the irradiation cell is the handling area. In this area, various connections are made for cryogenic, hydraulic, and pneumatic equipment.

An integral part of the GTR testing facility is the shuttle system, which is used to move test assemblies into irradiation position. This system consists of cable-driven dollies mounted on three sets of parallel tracks. The tracks extend from the irradiation positions adjacent to the reactor closet, up an incline to the north wall of the irradiation cell, and to a loading area on the ramp just north of the handling area. The system can be operated from either the control room or the dolly motor-drive shed on the north ramp. Full-coverage televiewing of the entire shuttle system is provided by means of a closed-circuit television in the control room.

The control room (Fig. A-1) is a below-grade, reinforced concrete structure adjacent to the GTR system. The control room provides a shielded area for reactor instrumentation, control consoles, and test systems as well as special test equipment needed to conduct radiation experiments.

APPENDIX B
DOSIMETRY

APPENDIX B

DOSIMETRY

Extensive nuclear measurements, performed prior to and during GTR Test 16, were required to provide data sufficient for a reliable characterization of the radiation incident on the test materials. The purpose of this discussion is to detail the procedures used for obtaining the incident-fast-neutron fluxes and the incident-gamma dose rates. All gamma dose values are based on the results obtained in the two mapping runs described in Section B.2.

B.1 GTR-16 Irradiations

B.1.1 Tensile Tests

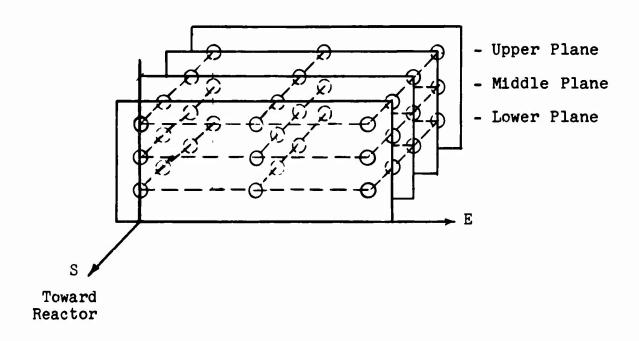
Measurements of the neutron flux were made with standard dosimetry packets attached to each rack of material specimens. Each packet contained a nickel foil for measurement of the fast-neutron flux (E>2.9 Mev) and two phosphorous pellets (one bare and one cadmium-shielded) for measurement of the thermal-neutron flux. Standard foil techniques were used specifying the neutron flux field. The activated foils were counted in the GD/FW Nuclear Radiation Effects Foil Counting Facility, and the data reduced using an IBM 7090 digital computer program.

From neutron spectral measurements (Ref. 2) made previously in the north position of the GTR irradiation facility, the following neutron flux ratios were obtained:

$$\frac{\Phi(E>1.0 \text{ Mev})}{\Phi(E>2.9 \text{ Mev})} = 2.82 \qquad \frac{\Phi(E>0.1 \text{ Mev})}{\Phi(E>2.9 \text{ Mev})} = 4.9$$

Preliminary analysis of the mapping experiment made prior to GTR-16 indicates no significant variation in the shape of the neutron spectrum between 0.1 and 2.9 Mev, regardless of position inside the dewar. Further, the neutron flux (E>2.9 Mev) measured during GTR-16 with nickel foils was virtually identical to that measured during the mapping experiment with sulfur pellets. The neutron flux for E>1 Mev was obtained by multiplying the neutron flux for E>2.9 Mev (measured with nickel foils) by the factor 2.82.

Figure B-1 is a sketch showing the position of dosimetry packets within the north dewar. Basically, the dosimetry packets were located on three horizontal planes - Upper, Middle, and Lower - corresponding to imaginary planes through the center of the upper, middle, and lower tensile specimen trays. The packets were, in general, in the same locations as those shown in the photographs in Section B.2.2. On each of these three planes, three packets were located longitudinally and four packets transversely, as shown below.



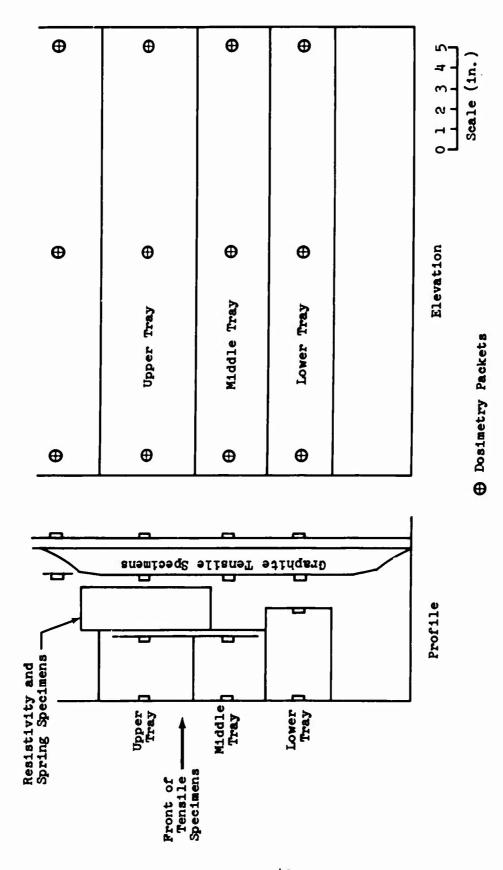


Figure B-1 Dosimetry Locations in North Irradiation Dewar

Plots of the integrated neutron flux (E>1 Mev) vs distance from front to rear through the specimen assembly are shown in Figures B-2, B-3, and B-4 for the Upper, Middle, and Lower planes, respectively. Similarly, Figures B-5, B-6, and B-7 are plots of the gamma dose for these planes. Figures B-8, B-9, B-10, and B-11 are plots of integrated neutron flux (E>1 Mev) in the planes in front of and behind the tensile and the graphite specimens; Figures B-12, B-13, and B-14 are plots of the gamma dose in front of and behind the tensile specimens and behind the graphite specimens. Table B-1 gives the integrated thermal-neutron flux (E<0.48 ev) for the Upper, Middle, and Lower planes.

Table B-1

Integrated Thermal-Neutron Flux in North Specimen Assembly (n/cm²)

	0.1				
Plane	West	Column Center	East		
UPPER Row 1	1.35(16)*	1.86(16)	1.02(16)		
Row 2 Row 3 Row 4	9.30(15) 1.12(16)	1.86(16) 1.47(16) 1.88(16)	1.02(16) 1.38(16) 1.42(16) 1.66(16)		
MIDDLE Row 1 Row 2 Row 3 Row 4	1.38(16) 5.0 (15)	2.20(16) 2.20(16) 1.25(16)	1,12(16) 1,38(16) 8,70(15)		
LOWER ROW 1 ROW 2 ROW 3 ROW 4	7.90(15) 6.60(15) 3.35(15) 6.85(15)	1.31(16) 2.21(15) 7.56(15)	5.50(15) 2.45(15) 5.50(15) 1.32(16)		

^{*}Read 1.35(16) as 1.35 x 10¹⁶

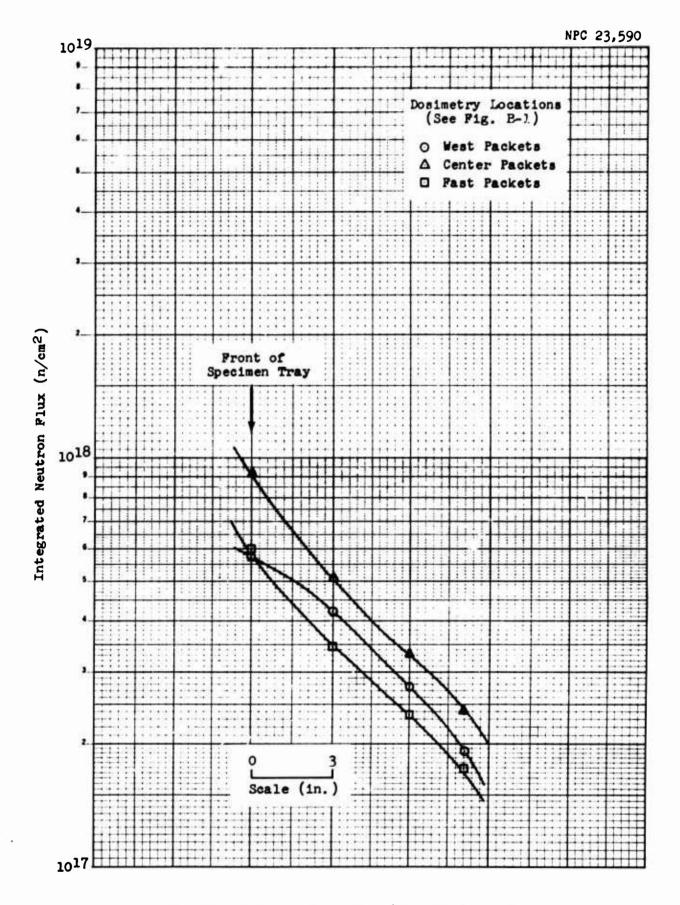


Figure B-2 Integrated Neutron Flux (E > 1 Mev) Profile - North Dewar, Upper Plane

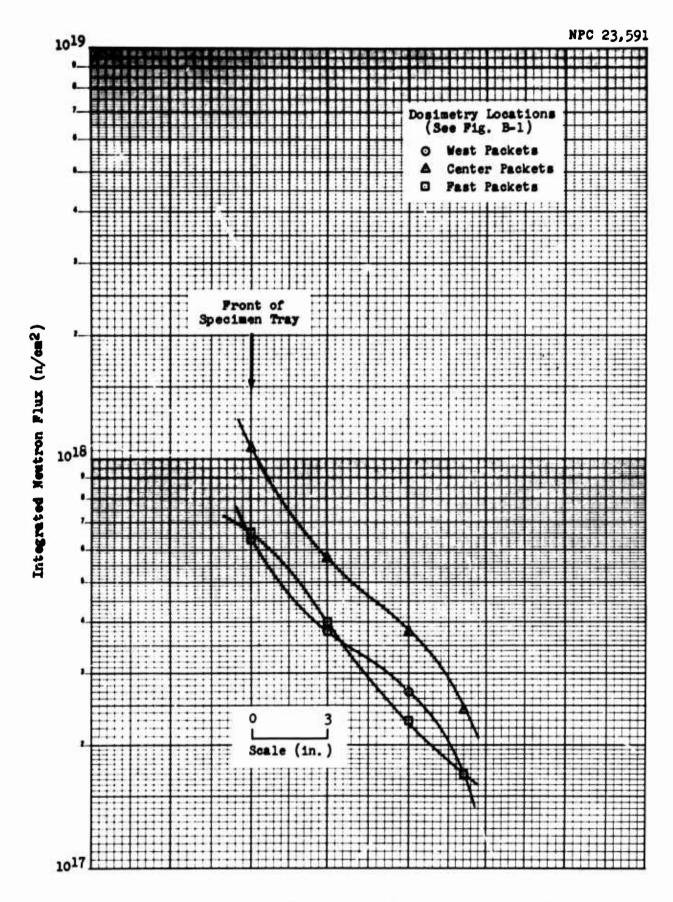


Figure B-3 Integrated Neutron Flux (E>1 Mev) Profile - North Dewar, Middle Plane

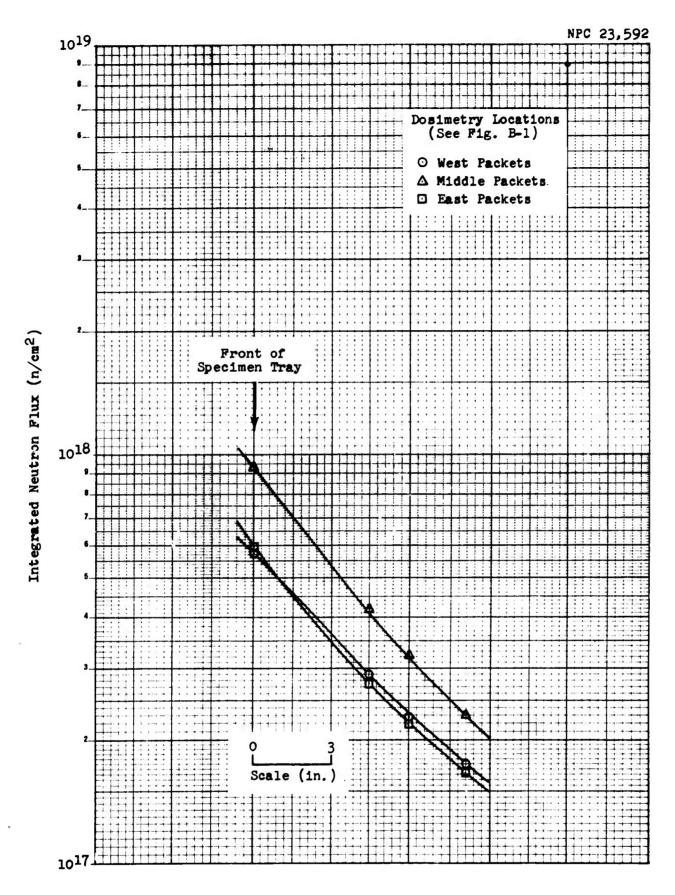


Figure B-4 Integrated Neutron Flux (E > 1 Mev) Profile - North Dewar, Lower Plane

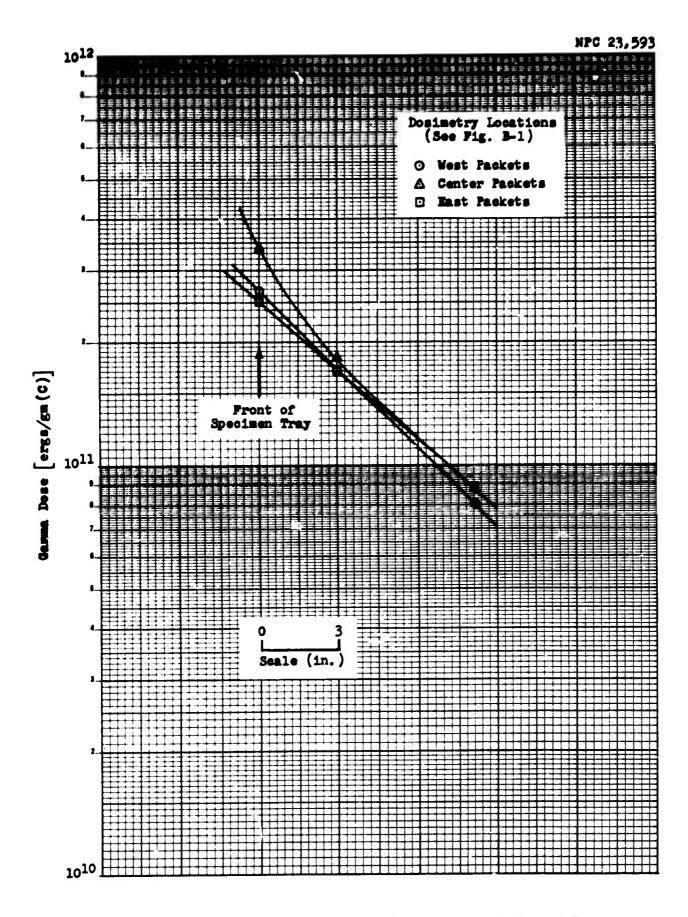


Figure B-5 Gamma Dose Profile - North Dewar, Upper Plane

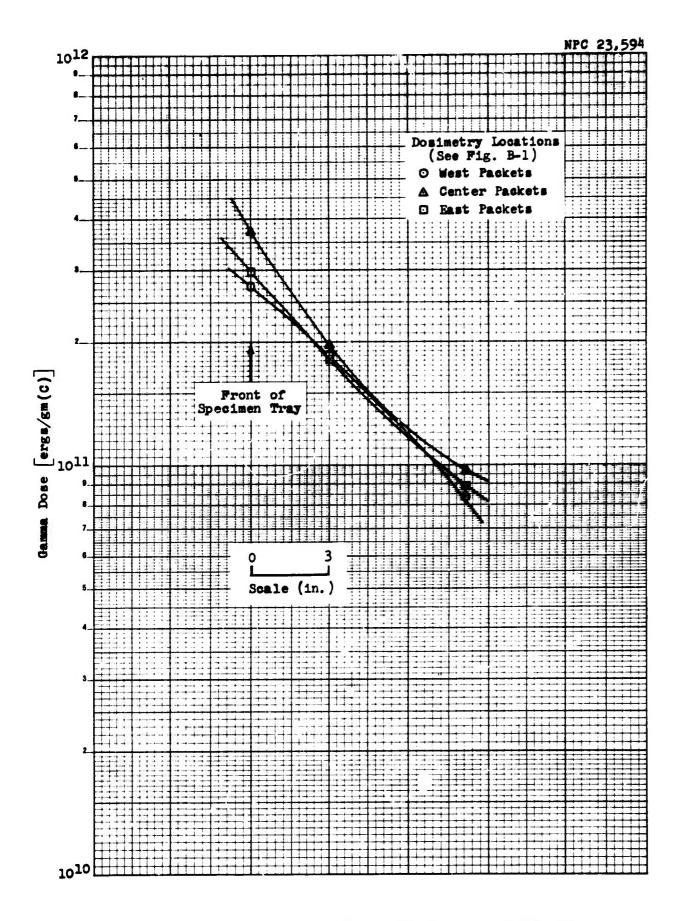


Figure B-6 Gamma Dose Profile - North Dewar, Middle Plane

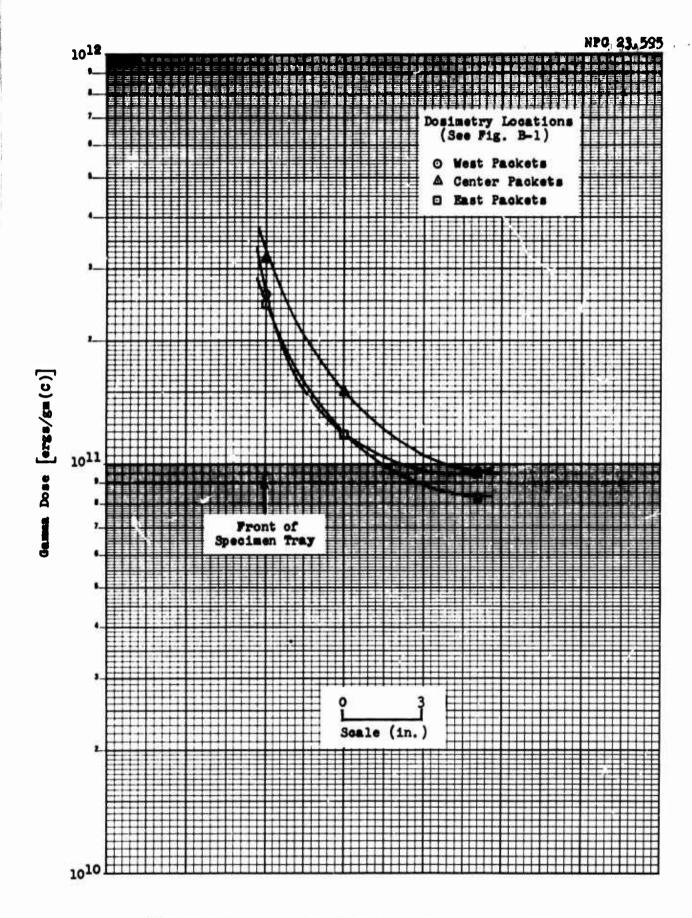


Figure B-7 Gamma Dose Profile - North Dewar, Lower Plane

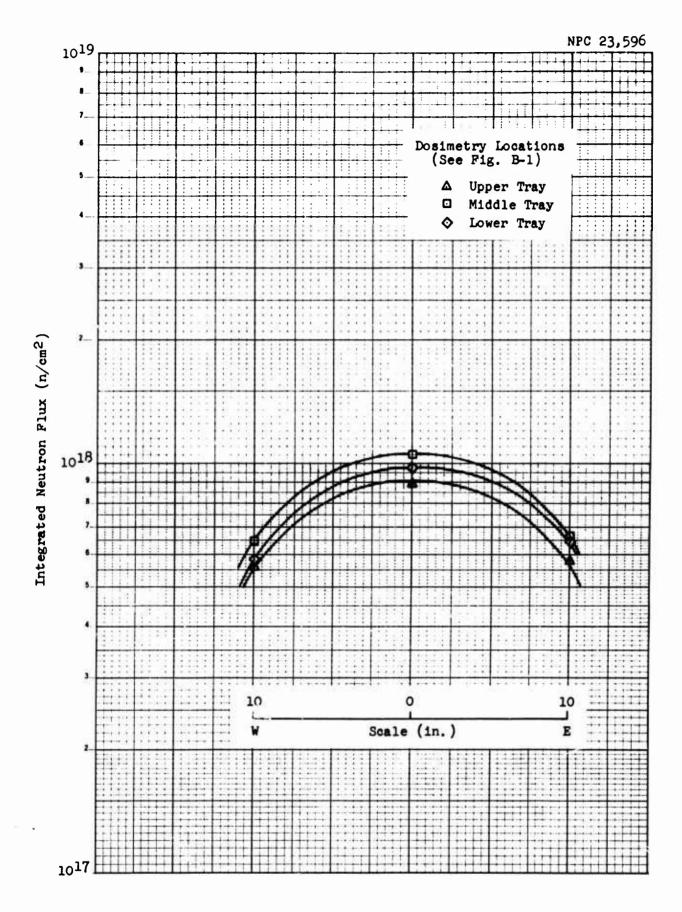


Figure B-8 Integrated Neutron Flux (E > 1 Mev) Profile - North Dewar, Plane in Front of Tensile Specimens

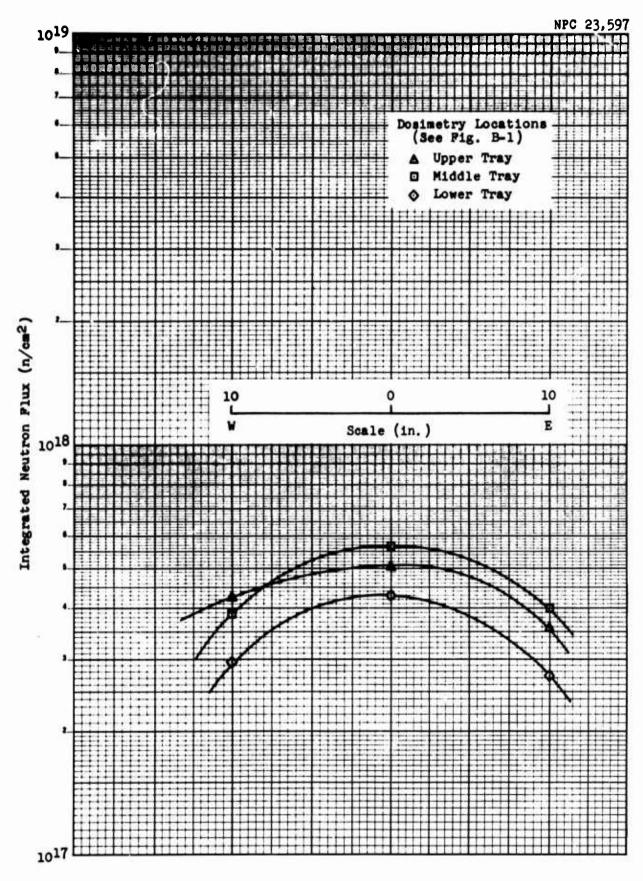


Figure B-9 Integrated Neutron Flux (E>1 Mev) Profile - North Dewar, Plane Behind Tensile Specimens

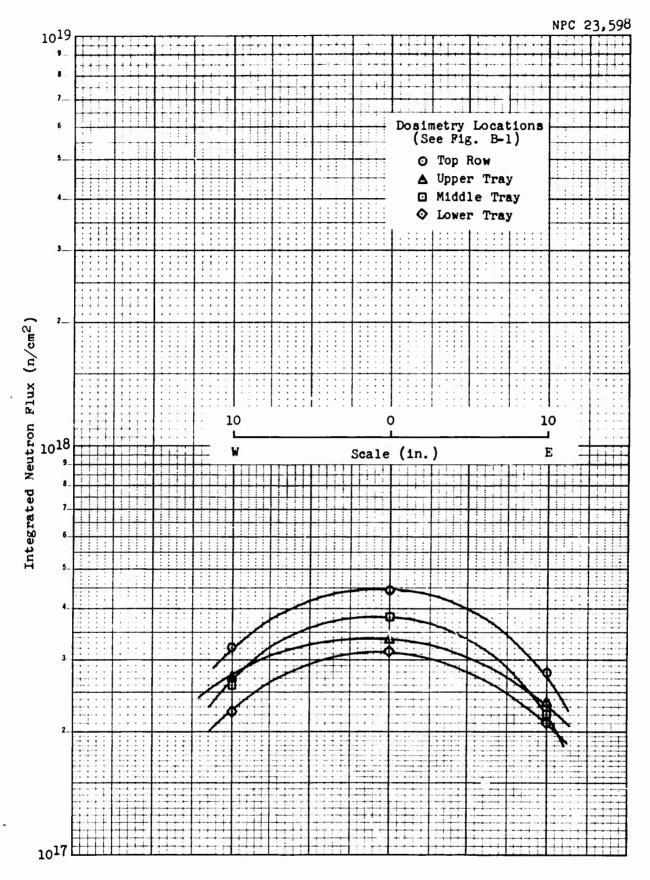


Figure B-10 Integrated Neutron Flux (E>1 Mev) Profile - North Dewar, Plane in Front of Graphite Specimens

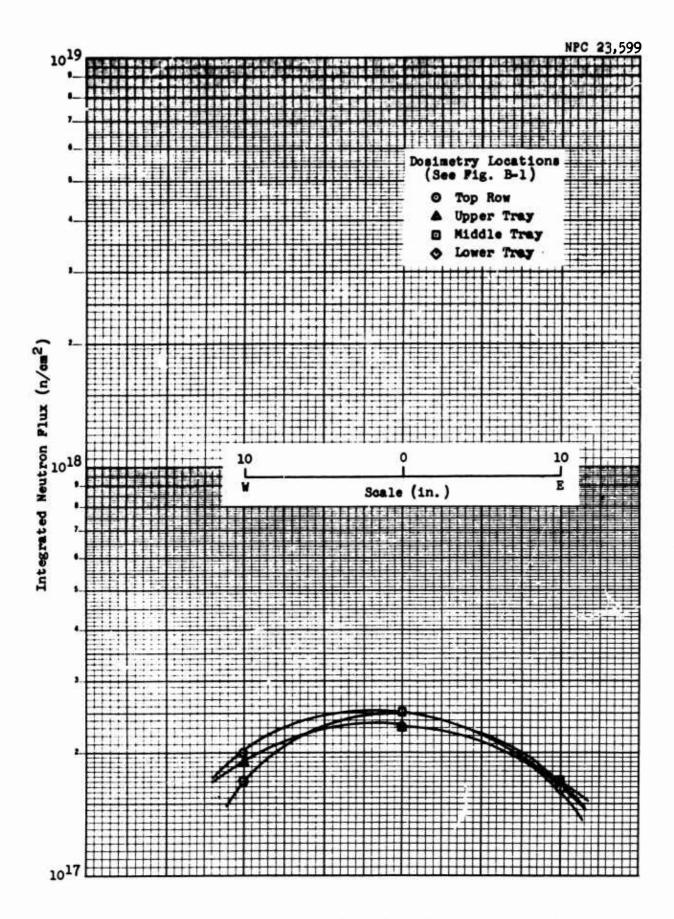


Figure B-11 Integrated Neutron Flux (E > 1 Mev) Profile - North Dewar, Plane Behind Graphite Specimens

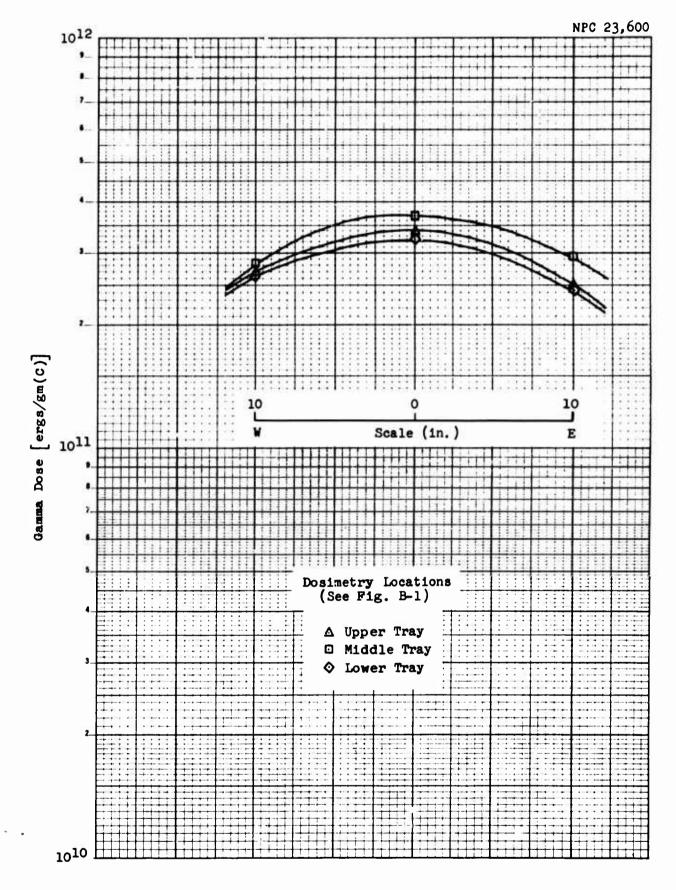


Figure B-12 Gamma Dose Profile - North Dewar, Plane in Front of Tensile Specimens

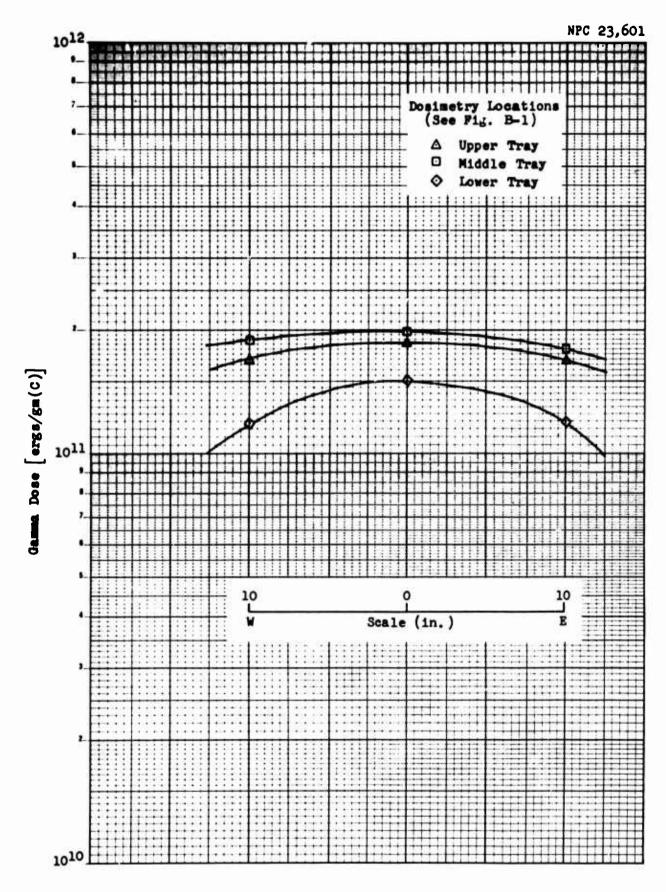


Figure B-13 Gamma Dose Profile - North Dewar, Plane Behind Tensile Specimens

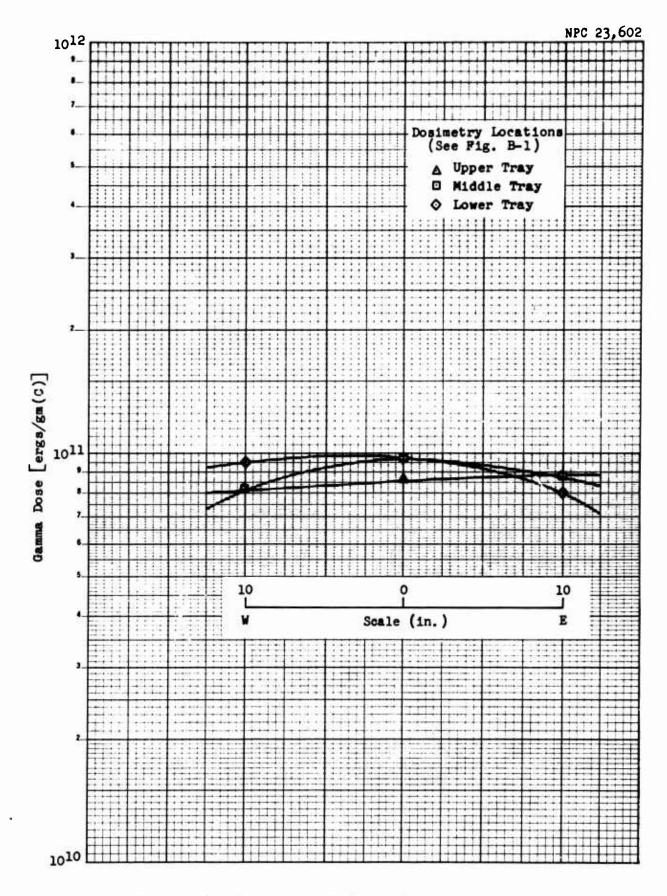


Figure B-14 Gamma Dose Profile - North Dewar, Plane Behind Graphite Specimens

B.1.2 Resistivity Test

These specimens, irradiated in the north position along with the tensile specimens, received an average integrated neutron flux of $4.5 \times 10^{17} \text{ n/cm}^2$ (E>1 MeV).

B.1.3 Steel Spring Test

These specimens, irradiated in the north position along with the tensile specimens, received an average integrated neutron flux of 9.5×10^{17} n/cm² (E>1 MeV).

B.1.4 C-Ring Seal Test

The two 0-ring fixtures were irradiated in the east position. One dosimetry packet containing one nickel foil and two phosphorous foils was attached to each fixture. The test fixtures received an integrated neutron flux of $4 \times 10^{16} \text{ n/cm}^2$ (E>2.9 MeV) and a gamma dose, based on the results of the mapping runs (see Sec. B.2), of $2.6 \times 10^{10} \text{ ergs/gm}(C)$.

B.1.5 Cemented Orifice Test

The dosimetry for these specimens is the same as that described in Section B.1.1. Specimens mounted inside the north dewar received an average integrated neutron flux of 2.5 x 10^{17} n/cm² (E>1 MeV) and an average gamma dose of 1.0 x 10^{11} ergs/gm(C). Those mounted outside the north dewar received an average integrated neutron flux of 4.0 x 10^{17} n/cm² (E>1 MeV) and an average gamma dose of 2.0 x 10^{11} ergs/gm(C).

B.2 GTR 16 Mapping Runs - North Dewar

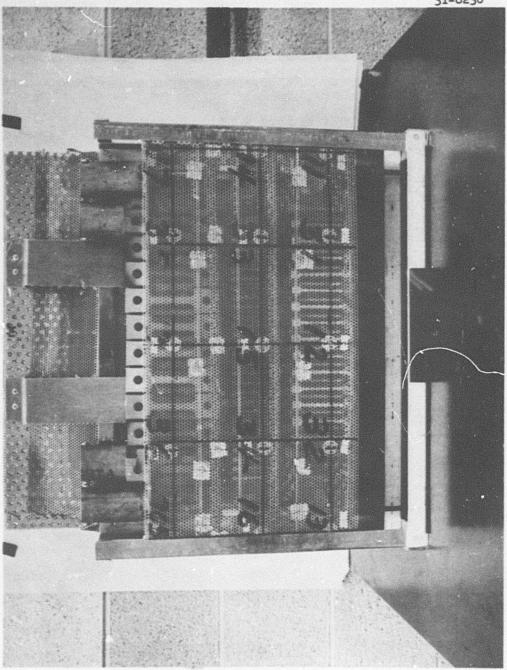
Passive nuclear measuring systems, i.e., neutron-detecting foils and gamma-dose-rate integrators, represent the most practical means for obtaining the desired radiation information. The neutron

exposure and gamma-dose range anticipated for GTR-16 exceeded the measuring capability of all neutron detectors except nickel (E>2.9 MeV) and all practical gamma detectors. Therefore, mapping tests at the desired dose levels were made before GTR-16 with foils and cobalt glass. Since the reactor core was relatively unpoisoned prior to GTR-16, and since the 200- and 400-hr planned irradiations at 3 Mw constituted a poisoned condition, two mapping irradiations were performed.

The first mapping run, with unpoisoned core, was made to verify predicted exposure conditions for foils and gamma detectors. The second, with the retracted core poisoned by a 45-Mw-hr exposure immediately preceding the mapping run, was made to establish neutron fluxes and spectral dependence, as well as gamma-dose rates, under simulated GTR-16 conditions. Both mapping runs were made with AGC and WANL cryogen test assemblies in irradiation position and with each containing actual test specimens (or facsimilies) and pertinent cryogen, namely, LH₂ in the AGC dewars and LN₂ in the WANL dewar. The AGC dewar, with pulling assembly and tensile test specimens, was located at the east irradiation cell position; the thermal-conductivity simulator with an AGC dewar was located at the west position; the WANL dewar was located at the north position.

Figures B-15 through B-18 show nuclear detector packet locations within the cryogen volume of the north dewar. In addition to these locations, detectors, placed in vertical planes in a symmetric pattern, were exposed in front and back of the cryogen fluid container to obtain supplementary dose-extrapolation data.

NPC 22,761 31-8230



NPC 22,762 31-8231

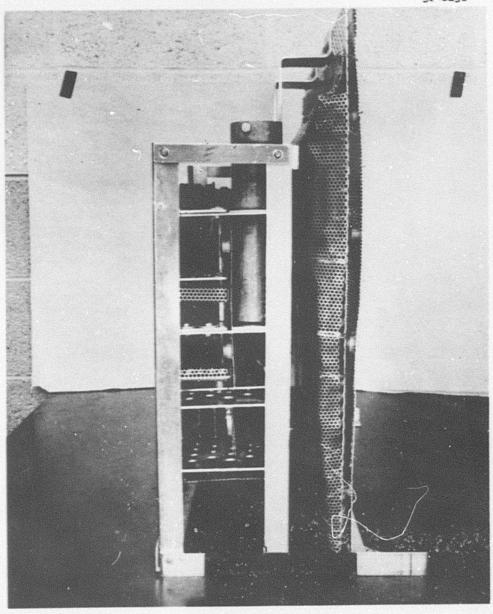


Figure B-16 Side View of Mapping-Run Dosimetry

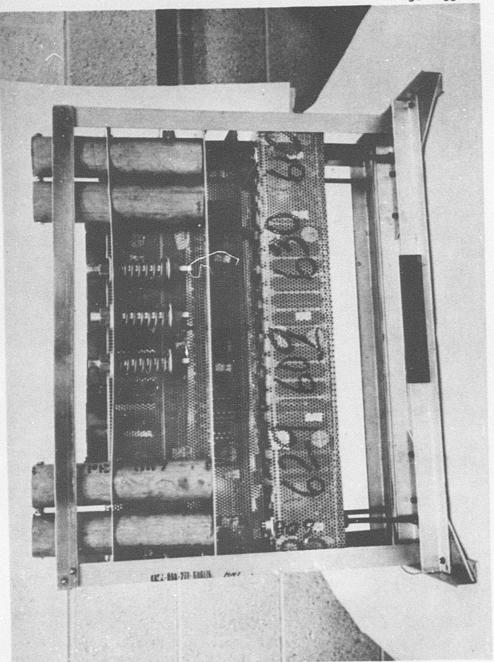


Figure B-17 Internal View of Mapping-Run Dosimetry

NPC 22,764 31-8232

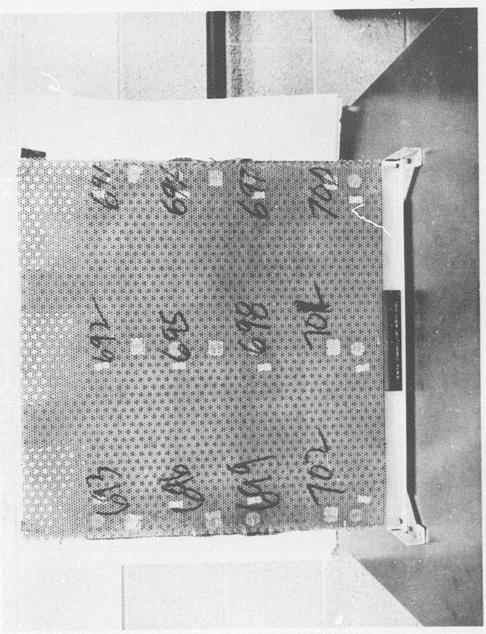


Figure B-19 illustrates schematically the overall detector arrangement for the mapping runs.

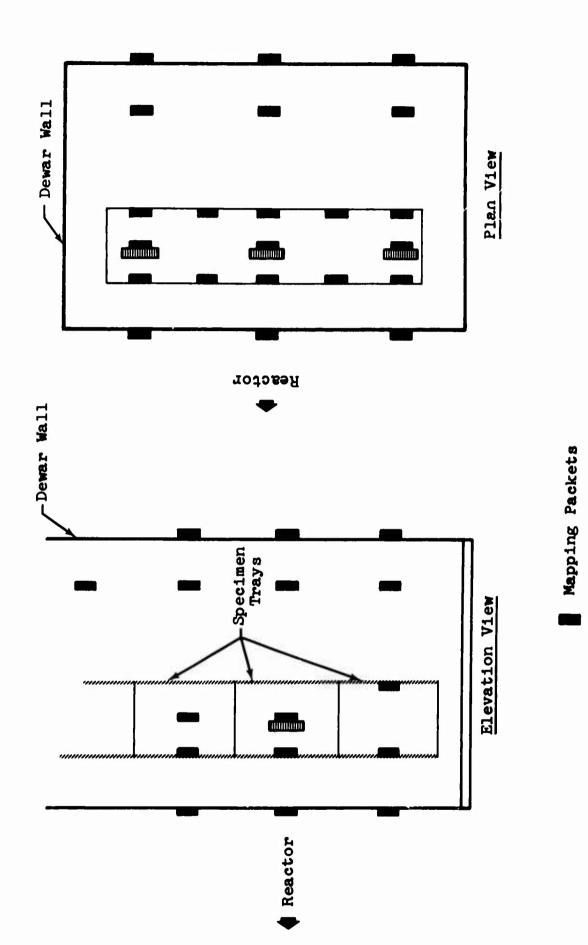
B.2.1 Detector Packets

Neutron and gamma detector packets for the described mapping locations consisted of indium, sulfur, and aluminum foils (one each per packet) for determining the fast flux of nominal energies greater than 0.85 Mev, 2.9 Mev, and 8.1 Mev, respectively; a pair of bare and cadmium-covered copper foils per packet for thermal-flux monitoring; and an enriched-boron-encapsulated cobalt glass per packet for gamma monitoring. These detectors were mounted on perforated aluminum sheets which, in turn, were wired to the test assembly at the given locations.

In addition to these detector packets, nine special neutron spectral packets were included in each mapping run (3 packets per test assembly) to provide additional data from which confidence in the extrapolation of the fast-neutron fluxes to 0.1 MeV would result. These packets contained resonance detectors for estimating the spectral dependence of the neutron flux between thermal energy and several kilovolts. The detectors, sensitive to (n,γ) reactions, included the common elements indium, gold, tungsten, cadmium, manganese, copper, and phosphorus, as well as the rare earths of lutetium, europium, samarium, and lanthenum. These detectors, in the form of thin, non-flux-perturbing foils, are being developed in the NARF program for intermediate-energy spectral studies of various neutron environments. Activation data are treated differently in that fluxes responsible for the activation are not calculated

Special Spectral Packets

NPC 23,603



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per se; rather, the data are compared with similar data obtained in more or less known spectral environments, with the result that confidence is gained in the accuracy of fast-flux spectral measurements.

The special spectral packets were mounted, together with additional fast-neutron detectors of indium, sulfur, and aluminum, at selected locations within the cryogen volumes indicated in Figure B-19.

B.2.2 Irradiation Procedure, Foil-Counting, and Data Processing

After all systems had been checked out, the three cryogen chambers were positioned in their respective locations north, east, and west of the reactor closet while the GTR was at zero power in the full south position. The east and west cryogen chambers were then filled with LH₂ and the north with LN₂. The systems were stabilized and the reactor, still in the retracted position, was brought to power, moved into irradiation position within the closet (2 in. of water on north face), and maintained at power for 30 min. The power level was 40 kw for the first mapping run and 120 kw for the second.

Since the purpose of the second mapping run (which was made several days after the first) was to simulate the poisoned core conditions encountered in long, high-power irradiations such as GTR Test 16, the 120-kw irradiation was immediately preceded by a 45-Mw-hr exposure followed by a 12-hr delay.

All foils were retrieved about 4 hr after each exposure and prepared for count-data accumulation. The foils were counted on

the GD/FW end-window flow counters at various time intervals until sufficient data were obtained for adequate IBM analysis with a statistical accuracy of 1-2%. All foil data were accumulated on tapes and processed by the K-26 IBM procedure for activity levels and flux information.

Cobalt-glass gamma detectors were also retrieved following the mapping irradiations and processed routinely for gamma exposure levels.

B.2.3 Data Analysis and Results

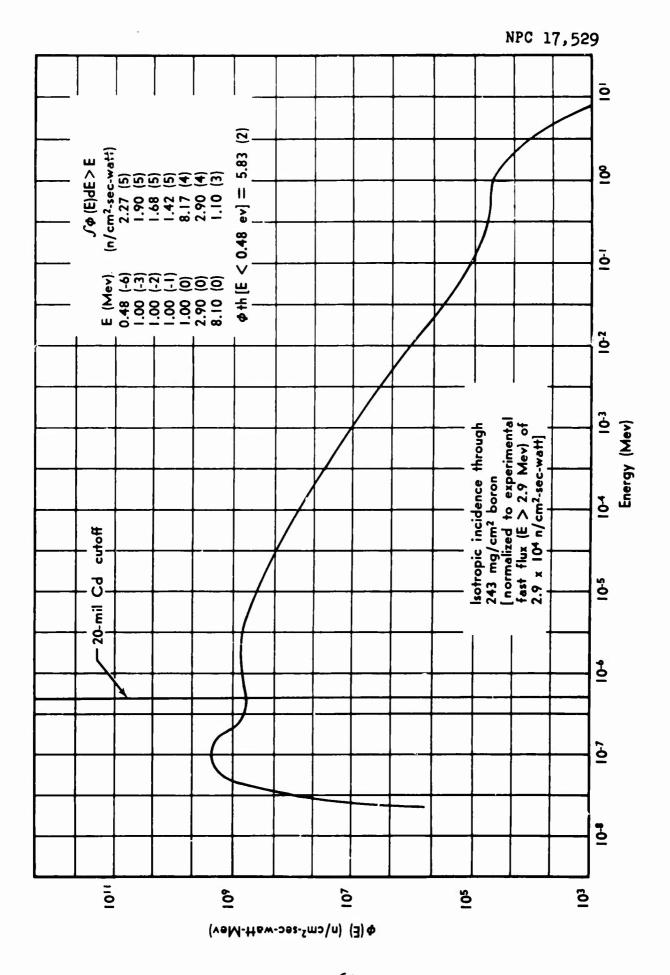
The data analysis for the two extensive mapping tests required the analysis of over 1500 flux and/or activation data points. The flux and spectral perturbations in all cryogen chambers were analyzed. It is anticipated that an independent report will be published on these data at some later date.

The neutron flux ratios calculated on the basis of all flux data from both runs include $\Phi_{\rm In}/\Phi_{\rm S}$ and $\Phi_{\rm S}/\Phi_{\rm Al}$, where $\Phi_{\rm In}$ is the fast flux for E>0.85 MeV, $\Phi_{\rm S}$ is that for E>2.3 MeV, and $\Phi_{\rm Al}$ is that for E>8.1 MeV. Activation ratios relative to sulfur activation were computed for the foils exposed in the special spectral packets. All flux and activation ratios for the cryogen chambers were compared with similar air data obtained in the normal boral-attenuated GTR irradiation cell. A brief description of the results for the north cryogen chamber follows.

The ratios $\Phi_{\rm In}/\Phi_{\rm S}$ and $\Phi_{\rm S}/\Phi_{\rm Al}$ calculated for the north chamber ranged from 3.11 to 3.69 and from 28.4 to 35.1, respectively. The averages of these values are within 5-10% of those anticipated

for the north irradiation volume when the GTR is operating with 2 in. of water reflector. No correlation of variance with location or between poisoned- and unpoisoned-core maps was obtained in the analysis.

The resonance-activation data indicate a significant depression within the cryogen chamber for the thermal and low-energy (<20 eV) epithermal fluxes. However, because of the small difference in the fast-flux ratios to those anticipated, the detailed analysis did not reveal any surprising results. Therefore, use of the GTR neutron spectrum, shown in Figure B-20, with flux monitoring data for the chamber should yield reliable values for the neutron fluxes in excess of 0.1 MeV.



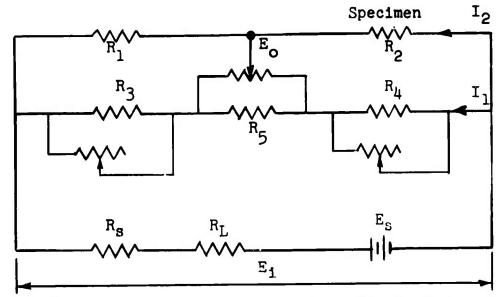
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APPENDIX C RESISTANCE BRIDGE ANALYSIS

APPENDIX C

RESISTANCE BRIDGE ANALYSIS

The following analysis is presented to show the relationship between specimen resistance change (ΔR_2) and bridge unbalance voltage (E_0^1) , bridge input voltage (E_1) , and dummy resistance plus specimen resistance $(R_1 + R_2)$.



When the bridge is balanced, the following relationships exist:

$$I_1R_4^1 = I_2R_2$$
 and $I_1R_3^1 = I_2R_1$

or

$$\frac{I_1 R_4^1}{I_1 R_3^1} = \frac{I_2 R_2}{I_2 R_1}$$

After cancelling and rearranging terms, we have

$$R_2 = \frac{R_4^1 R_1}{R_3^1}$$

where

 R_3^1 = resistance R_3 plus that portion of R_5 required for halance

 R_{μ}^{1} = resistance R_{μ} plus that portion of R_{5} necessary for balance

Also in the balanced system, we have

$$I_1 = \frac{E_1}{R_4^1 + R_3^1}$$
 and $I_2 = \frac{E_1}{R_1 + R_2}$

$$E_0 = I_1 R_4^1 - I_2 R_2$$
 and

$$E_0 = \frac{E_1 R_{\downarrow \downarrow}^1}{R_{\downarrow \downarrow}^1 + R_3^1} - \frac{E_1 R_2}{R_1 + R_2} = E_1 \left(\frac{R_{\downarrow \downarrow}^1}{R_{\downarrow \downarrow}^1 + R_3^1} - \frac{R_2}{R_1 + R_2} \right)$$

E 0 at bridge balance

At bridge unbalance (E_0^1) due to change in specimen resistance, R_2 changes to $(R_2 + \Delta R_2)$, so that

$$E_0^1 = E_1 \left(\frac{R_{4}^1}{R_{3}^1 + R_{4}^1} - \frac{R_2 + \Delta R_2}{R_1 + R_2 + \Delta R_2} \right)$$

and since ΔR_2 (approximately 0.01 ohm) is very small compared to R_1 + R_2 (approximately 32 ohms), it can be neglected in the denominator above, making

$$E_0^1 = E_1 \left(\frac{R_4^1}{R_3^1 + R_4^1} - \frac{R_2}{R_1 + R_2} - \frac{\Delta R_2}{R_1 + R_2} \right)$$

Since
$$R_2 = \frac{R_4^1 R_1}{R_3^1}$$
,

$$\frac{R_4^1}{R_3^1 + R_4^1} - \frac{R_2}{R_1 + R_2} = 0$$

Therefore,
$$E_0^1 = E_1 \left(-\frac{\Delta R_2}{R_1 + R_2} \right)$$

or
$$\Delta R_2 = -\frac{E_0^1 (R_1 + R_2)}{E_1}$$

Sample Calculations

During irradiation the bridge output voltage changed by approximately -2000 $\mu\nu$. This corresponds to a resistance increase of approximately 0.02 ohm, as shown below:

$$\Delta R_2 = -\frac{E_0^1 (R_1 + R_2)}{E_1} = \frac{2000 \times 10^{-6} (16 + 16)}{3.15}$$

$$\Delta R_2 = 0.0203 \text{ ohm}$$

where

 $R_1 = 16 \text{ ohms}$

 $R_2 = 16 \text{ ohms}$

 $E_1 = 3.15 \text{ volts}$

A typical maximum bridge voltage unbalance during postirradiation annealing treatments was 935 μv . This corresponds to a

resistance decrease of approximately 0.01 ohm, as shown below:

$$\Delta R_2 = -\frac{E_0^1 (R_1 + R_2)}{E_1} = -\frac{935 \times 10^{-6} (16 + 16)}{2.94}$$

$$\Delta R_2 = -0.0102 \text{ ohm}$$

REFERENCES

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- 2. NARF Facilities Handbook, General Dynamics/Fort Worth Report FZK-185A, March 1964.